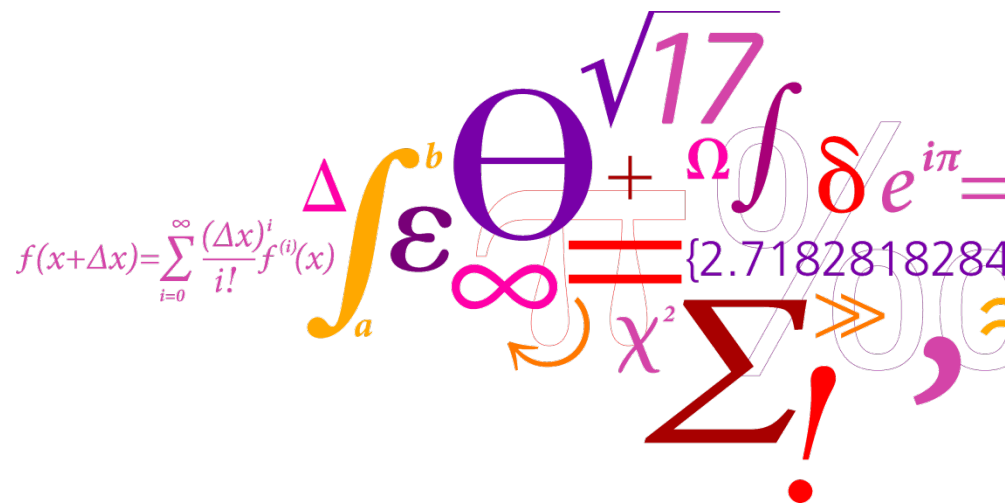


# A new type of white light-emitting diode using fluorescent silicon carbide (LEDSiC)

Acknowledgement:

Innovation Fund Denmark (No. 4106-00018B)



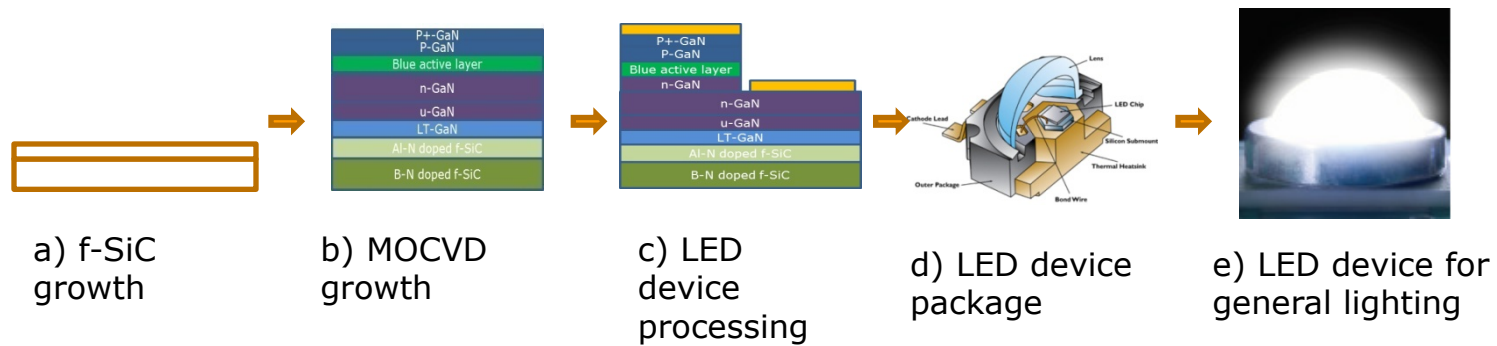
# Agenda

- Participants: 12 Danish partners (Leif Jensen; Flemming Jensen; Berit Herstrøm; Carsten Dam-Hansen; Paul Michael Petersen; Yiyu Ou; Yi Wei; Li Lin; Weifang Lu; Jiehui Li; Ahmed Fadil; Haiyan Ou)
- Location: **S04 in building 101**

## Agenda:

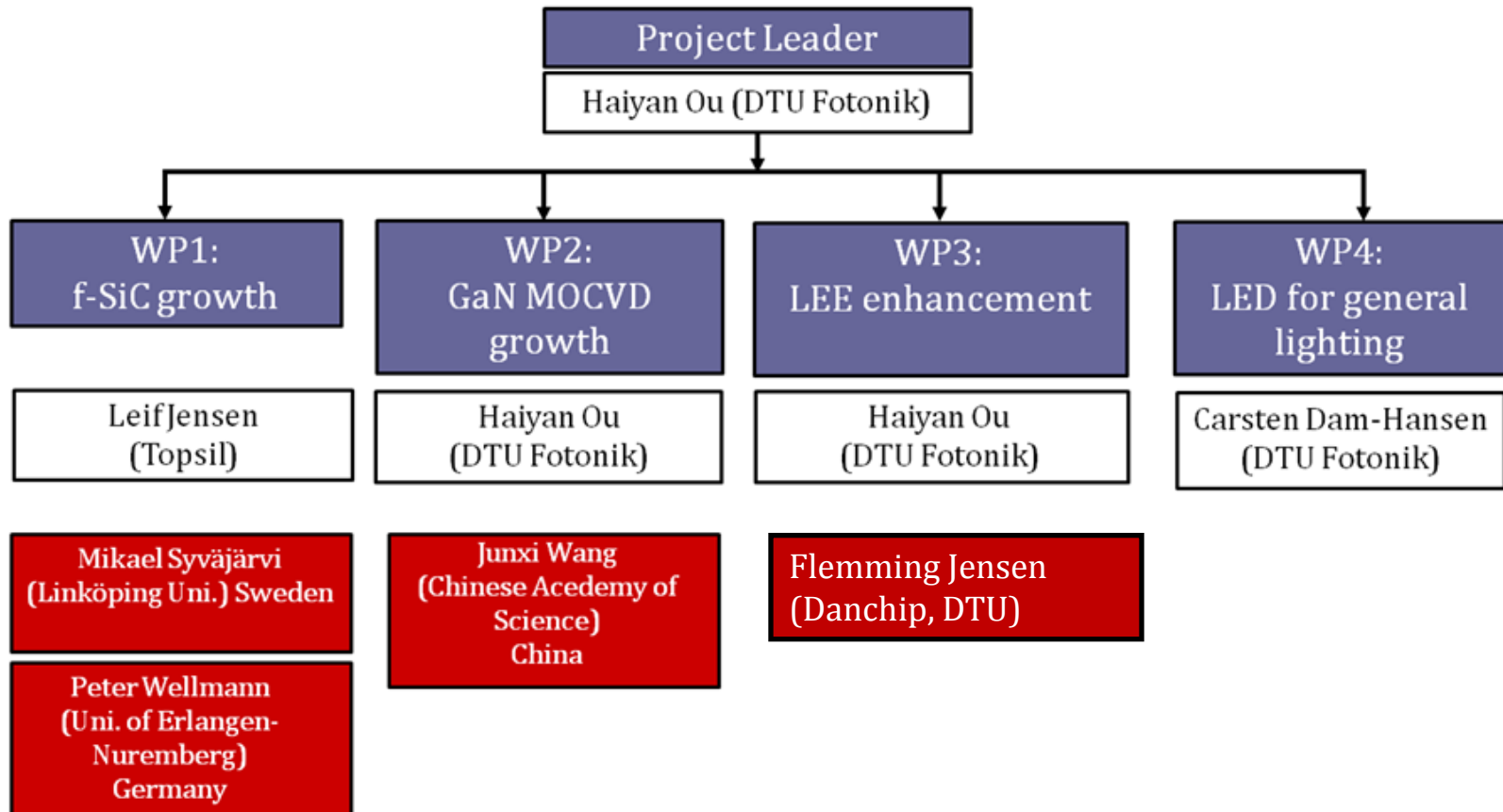
- 10:00-10:05 Status of the project by Haiyan
- 10:05-10:25 Main results on 'Hybrid structures for luminescence enhancement of fluorescent SiC' by Yiyu
- 10:25-10:45 Main results on 'PL enhancement of <math>\langle 6H \rangle</math> bulk SiC by localized surface plasmon' by Yi
- 10:45-11:00 Coffee/Tea break and discussion
- 11:00-11:20 Main results on 'Porous SiC' by Weifang
- 11:20-11:40 Main results on 'Fabrication and characterization of GaN LEDs' by Li
- 11:40-11:55 Main results on 'Package of LEDs' by Jiehui
- 11:55-12:00 Concluding remark by Haiyan
- 12:00-13:00 Sandwich and discussion

# Work packages

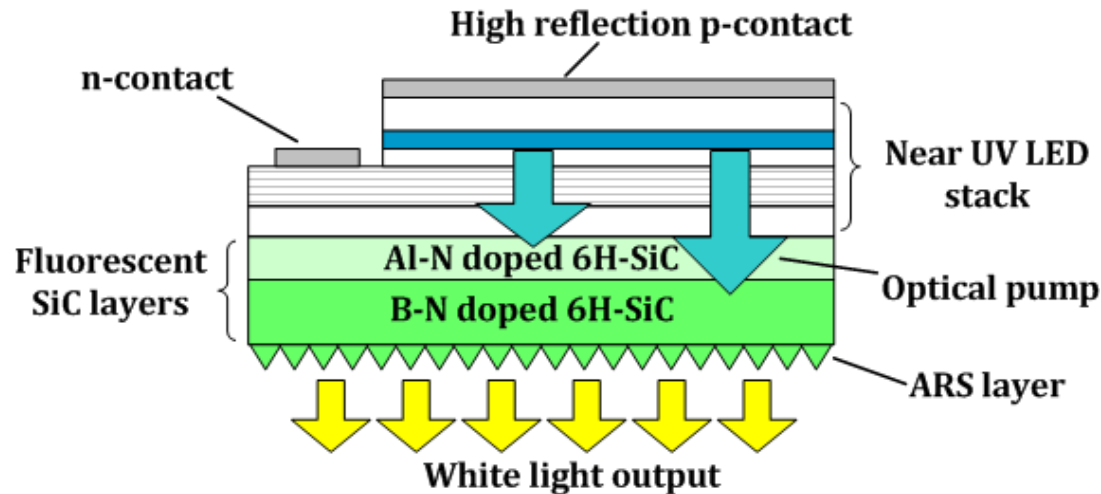


- **WP#1** f-SiC growth and optimization
- **WP#2** MOCVD growth of GaN based LED on f-SiC
- **WP#3** Efficiency enhancement of LED in term of light extraction
- **WP#4** Processing and optical characterization of white LED device for general lighting

# Project structure and division into work packages



# Status






- Epitaxial growth: **boron-nitrogen (B-N)** co-doped f-SiC
- MOCVD growth: GaN blue LEDs on SiC substrates




At DTU:

- **Surface nanostructuring** on the f-SiC surface to enhance the light extraction efficiency
- **Surface plasmon** for emission efficiency enhancement
- **Porous SiC** and passivation for an alternative light source
- A post-growth **LED processing flow** (photolithography, ICP etch, n and p contact, etc.) is being developed in the cleanroom of DTU Danchip
- **LED package** for system application

# Main manpower:

Main participants	WP involved	Focus	Main activities
 <p>Ph. D student <b>Yi Wei</b></p>	WP1	f-SiC material growth	<ol style="list-style-type: none"> <li>Growth of source material using physical vapor deposition (PVD) method</li> <li>Growth of epitaxial layer using fast sublimation growth process (FSGP)</li> <li>Material characterization (SIMS, X-ray diffraction, carrier lifetime, photoluminescence etc.) for optimization of the material growth</li> </ol>
 <p>Ph. D student <b>Li Lin</b></p>	WP2, WP3, WP4	The fabrication and optimization of LEDs for general lighting	<ol style="list-style-type: none"> <li>Post processing of the LED devices including mesa etching, electrode deposition, etc.</li> <li>Surface nanostructuring and passivation;</li> <li>On-chip LED tests (IV curve, IP curve, efficiency, CRI, etc.)</li> <li>Package of the LED devices</li> <li>LED test and evaluation for general lighting</li> </ol>
 <p>Postdoc <b>Yiyu Ou</b></p>	WP1, WP2, WP3, WP4,	MOCVD growth of GaN on top of f-SiC for a complete LED device The fabrication and optimization of LEDs for general lighting	<ol style="list-style-type: none"> <li>High efficiency GaN LED growth on f-SiC by using MOCVD</li> <li>Material characterization of the grown GaN LED by using SEM, TEM, X-ray diffraction, etc.</li> <li>Optical characterization of the complete LED device by using electroluminescence for efficiency and CRI, etc.</li> </ol>

# Main manpower+:

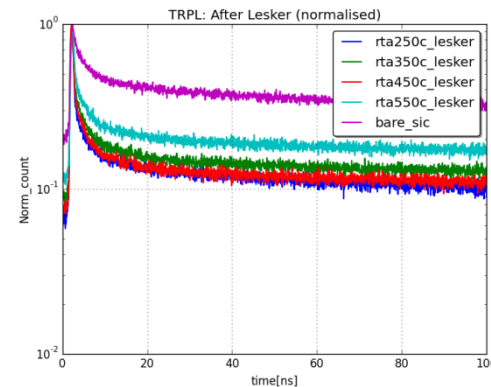
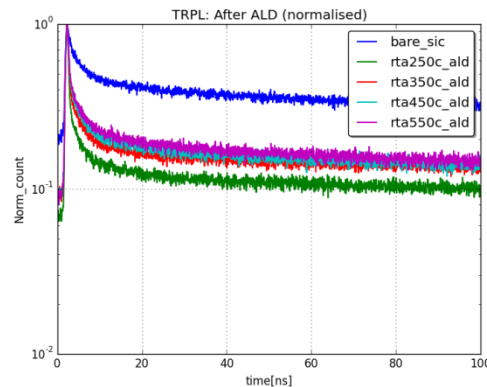
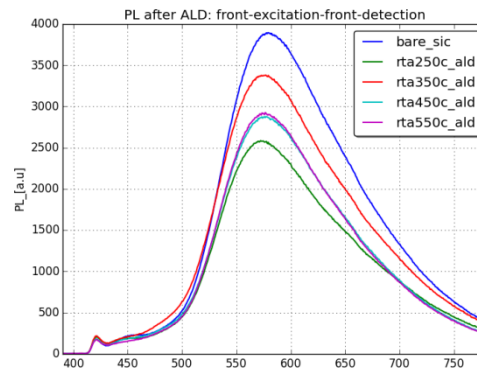
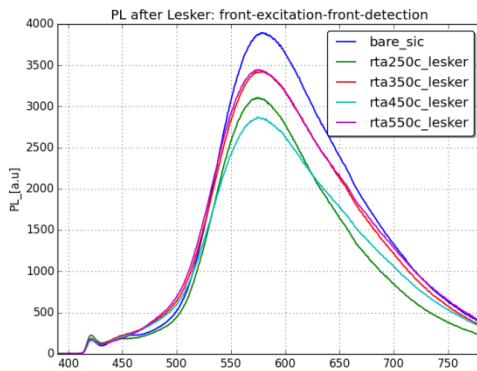
Main participants	WP involved	Focus	Main activities
 <p>Ph. D student <b>Weifang Lu</b></p>	WP1	f-SiC material growth	<ol style="list-style-type: none"> <li>1. Passivation of surface textured f-SiC</li> <li>2. Fabrication and passivation of porous SiC</li> </ol>
 <p>Visiting Ph. D student <b>Jiehui Li</b></p>	WP3, WP4	The fabrication and package of LEDs for general lighting	<ol style="list-style-type: none"> <li>1. On-chip LED tests (IV curve, IP curve, efficiency, CRI, etc.)</li> <li>2. Package of the LED devices</li> <li>3. LED test and evaluation for general lighting and visible light communication</li> </ol>
 <p>Postdoc <b>Ahmed Fadil</b></p>	WP1,	Surface plasmon enhancement of f-SiC	<ol style="list-style-type: none"> <li>1. Surface plasmon enhanced NUV LED</li> <li>2. Surface plasmon enhanced f- SiC</li> </ol>

• Motivation:

- N-doped <6H> SiC: Luminescent material with active layer ~ 100nm.
- Ag NPs: Inducing localized surface plasmon for PL enhancement.
- Al<sub>2</sub>O<sub>3</sub> thin film: Tuning  $\lambda_{peak}$  of LSP & Oxidation prevention.

• Experimental Investigation:

- Temperature of rapid thermal annealing (RTA) of Ag thin film.
- Method of Al<sub>2</sub>O<sub>3</sub> deposition.
- Thickness of Ag thin film (resulting in different size of Ag NPs).



Top: PL / Bottom: TRPL

Left: different RTA temperature, Al<sub>2</sub>O<sub>3</sub> grown by ALD method;

Right: different RTA temperature, Al<sub>2</sub>O<sub>3</sub> grown by RF sputtering method;

**Conclusion:**

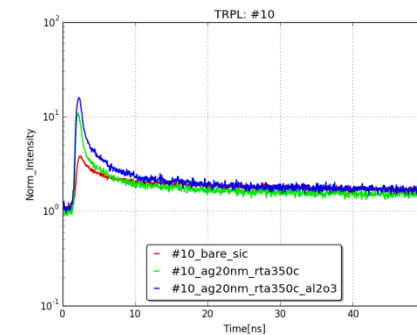
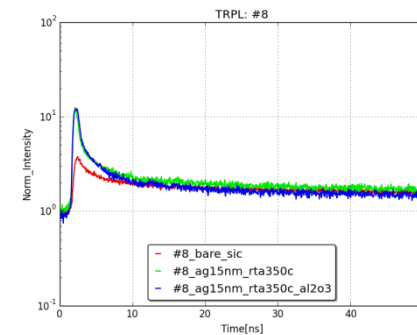
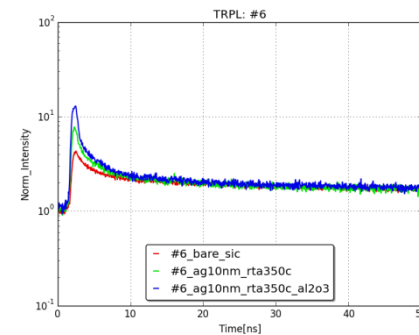
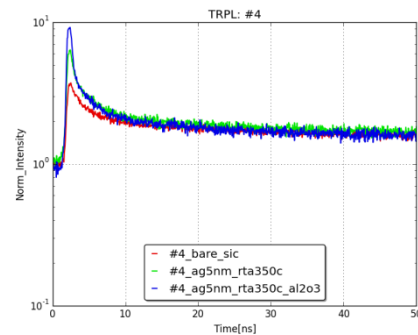
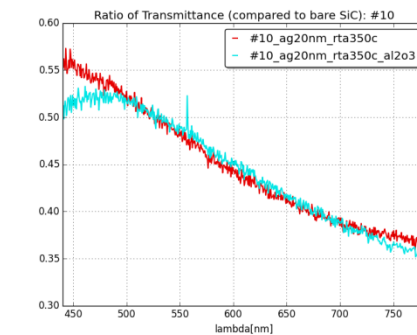
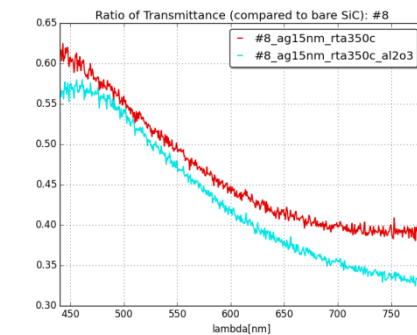
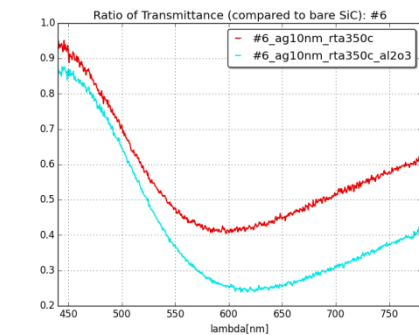
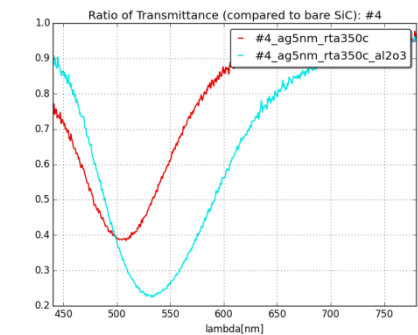
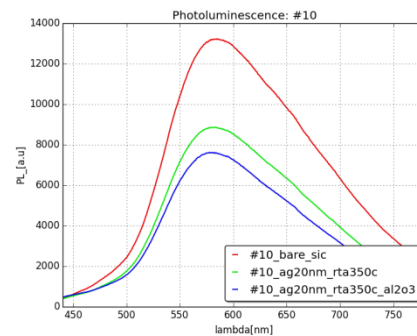
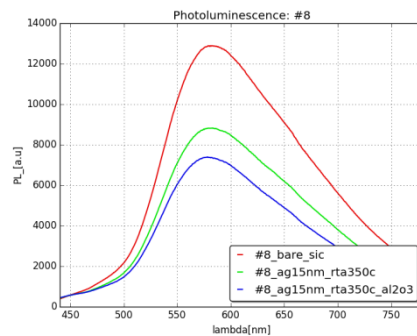
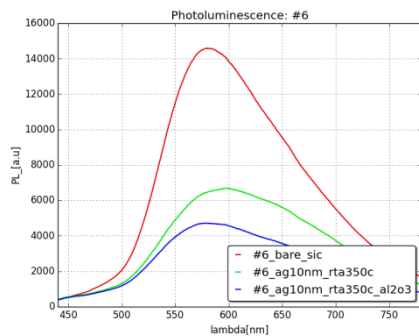
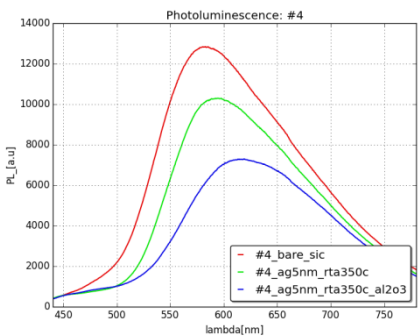
-For Al<sub>2</sub>O<sub>3</sub> deposition at ~20nm range, ALD and RF sputtering makes no apparent difference;

-RTA at 350 degree is preferable;



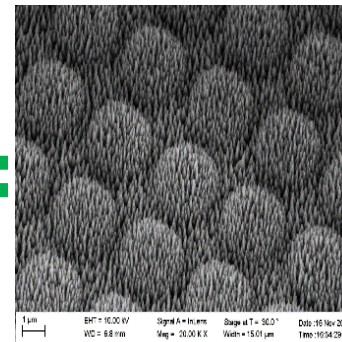
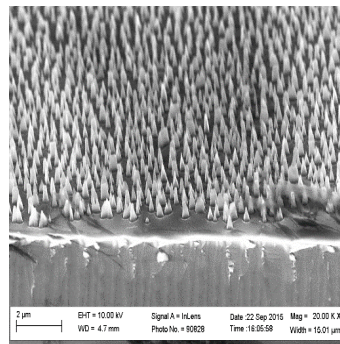
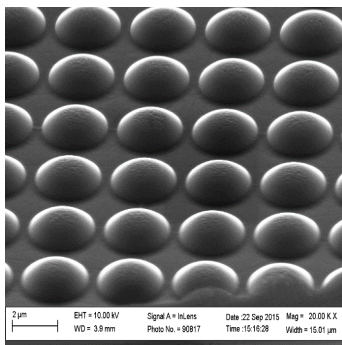
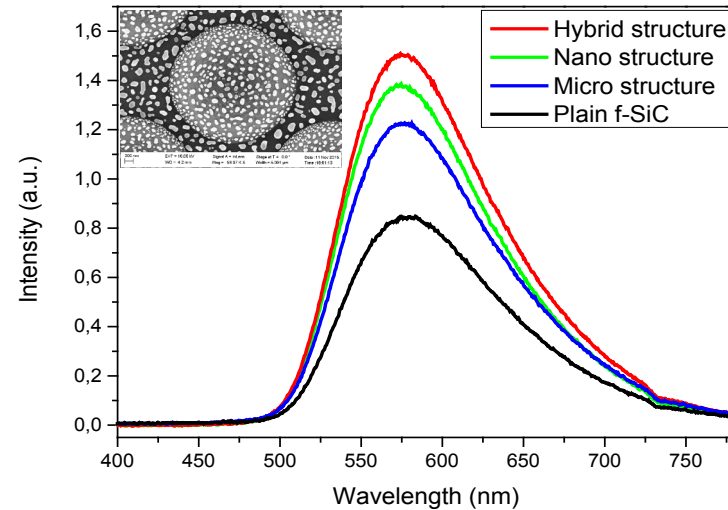
**Conclusion:**

- Al2O3 thin film will further reduce PL but cause  $\lambda_{peak}$  shift;
- LSP is observed, in which the range depends on Ag NPs size;
- The photon lifetime will be decreased by Ag NPs, in which can be further decreased when coated with Al2O3.



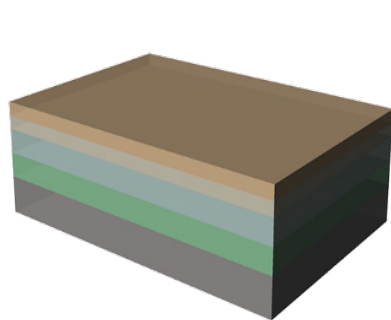
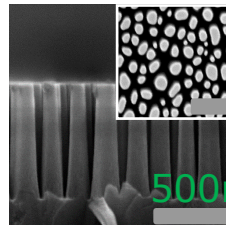
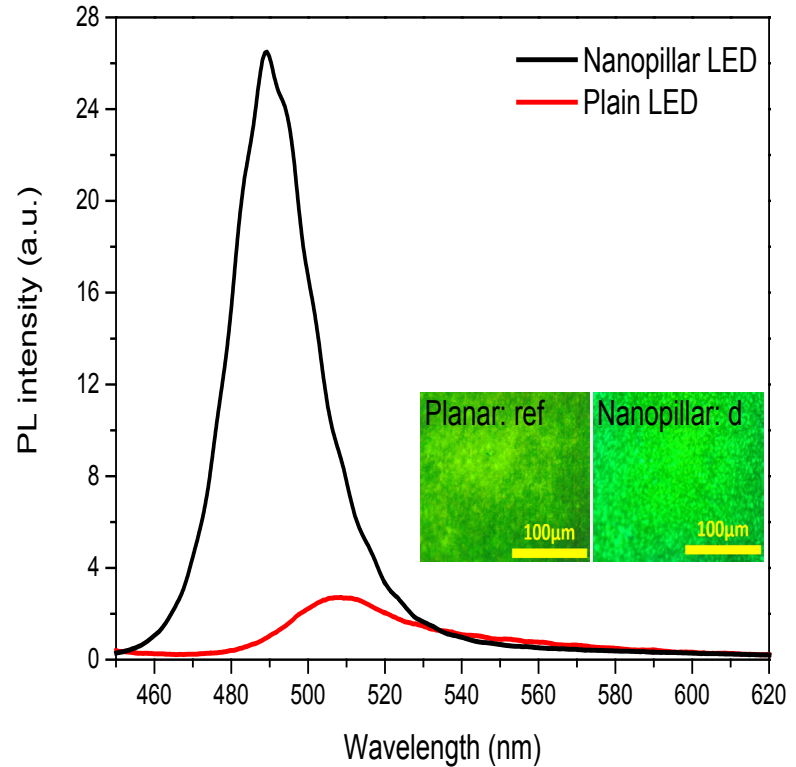
# Fabrication of hybrid structures on SiC

- A combination of micro-structure ( $\sim 3\mu\text{m}$ ) and nano-structure (100-200nm)
- Method: speical photolithography with nanopatterning
- Larger luminescence enhancement than pure nano- or micro-structure

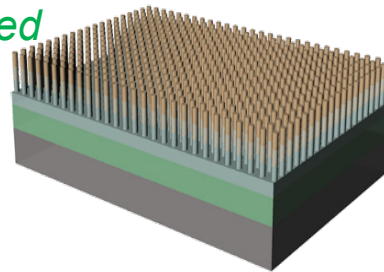


# Fabrication of nanopillar structure LED

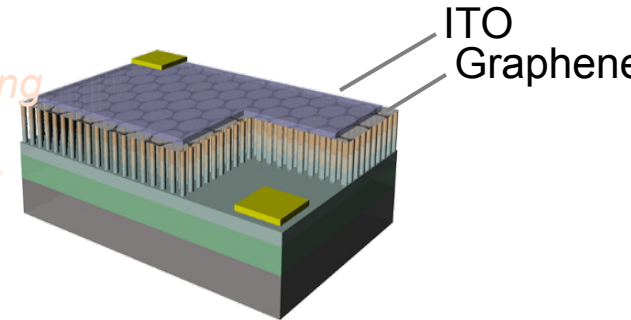
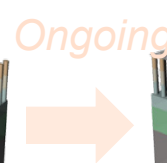
- To reduce the QCSE in QWs by releasing the internal strain
- Method: nanopatterning with dry etching
- Significant luminescence enhancement: more than 4 times



LED stack



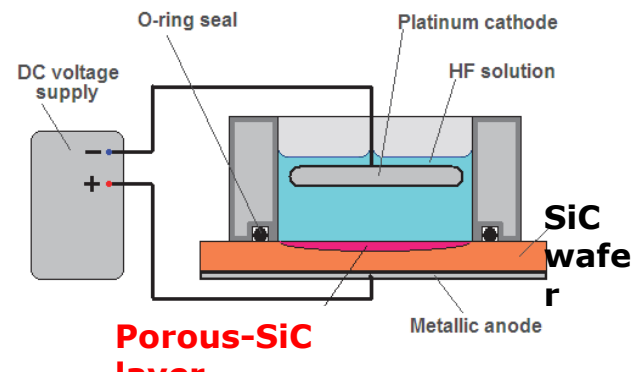
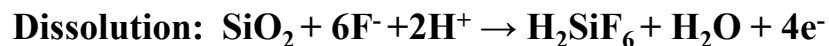
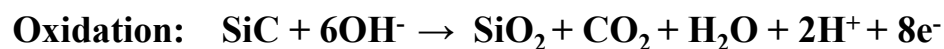
Nanopillar LED



Nanopillar LED device

# 1 Fabrication of porous SiC by anodic oxidation method

The SiC anodic etching in HF solution could be described with two steps, oxidation of SiC and dissolution of the formed SiO<sub>2</sub>:



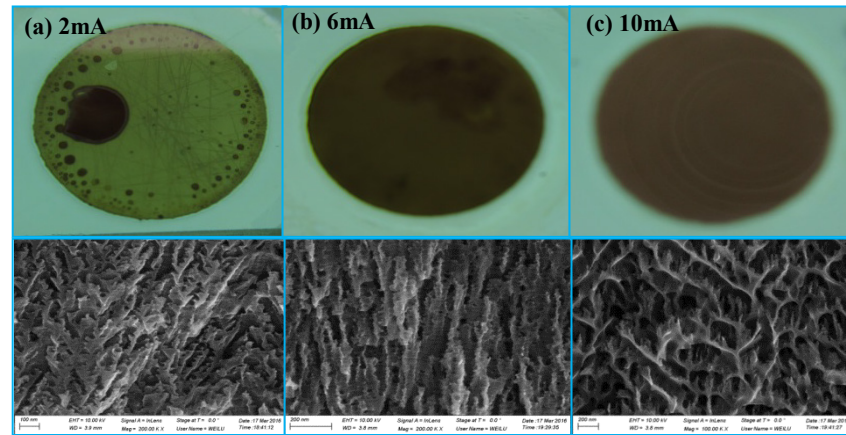
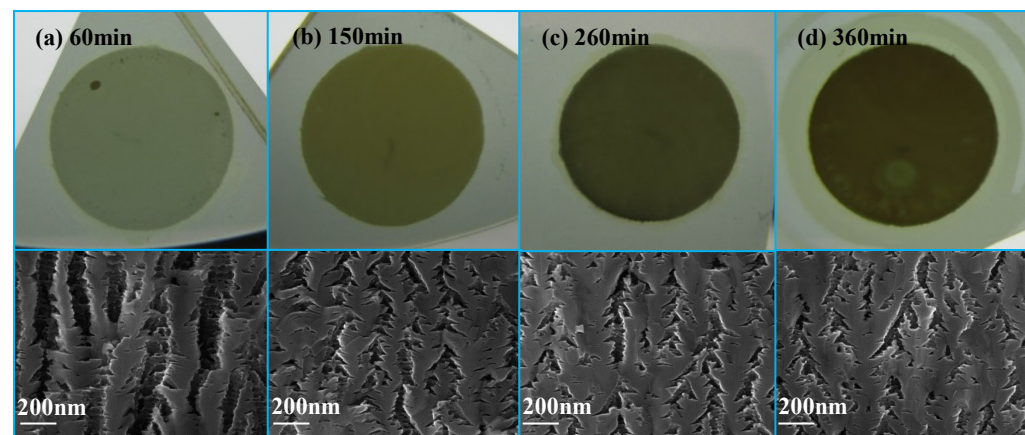
Schematic diagram of the experimental setup for anodic oxidation

## ① SiCrystal 6H-SiC

Thickness: 250 μm, N-doped, on-axis, Si-Face polished, MPD < 100/cm<sup>2</sup>

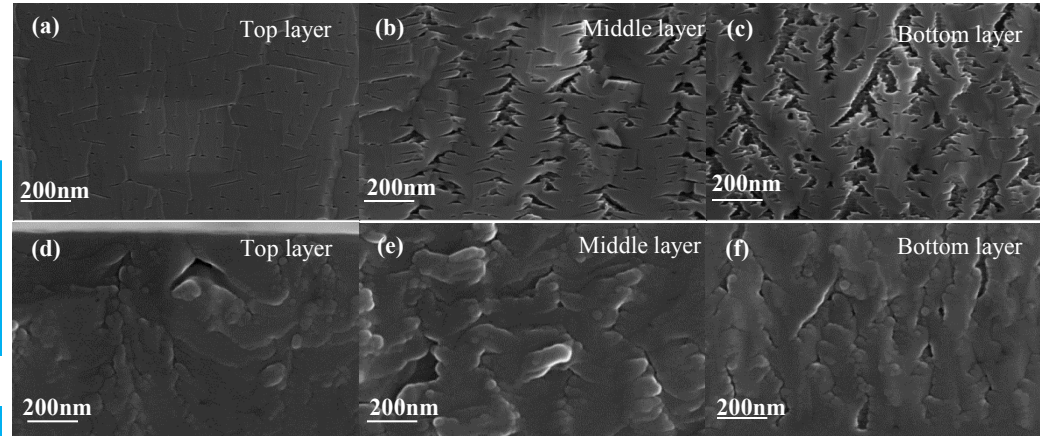
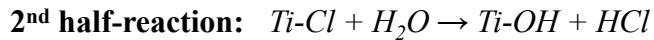
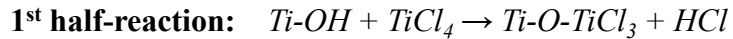
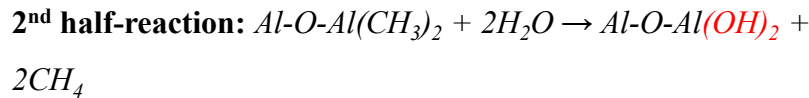
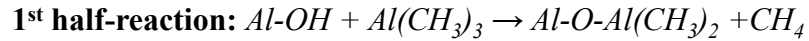
## ② Tankeblue 6H-SiC

Thickness: 430 μm, B-N co-doped  
Double sides polished with Si face CMP



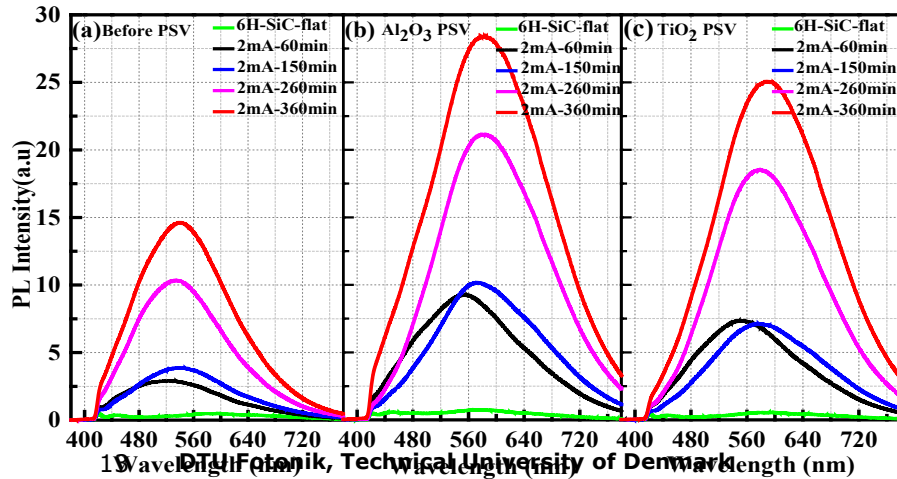
# 2 Passivation by atomic layer deposited $\text{Al}_2\text{O}_3$ and $\text{TiO}_2$ film

The surface chemistry reaction during  $\text{Al}_2\text{O}_3$  and  $\text{TiO}_2$  ALD (atomic layer deposition) deposition:

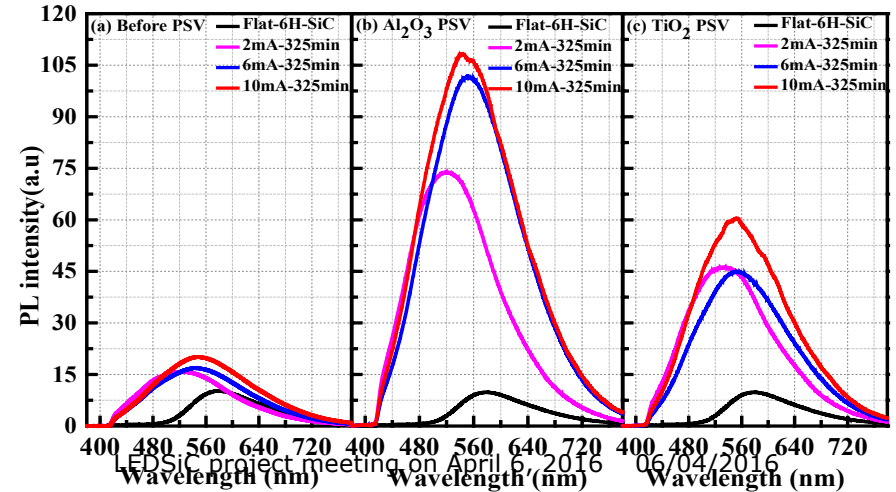


Cross-sectional SEM images of SiCrystal sample (360 min): (a) top layer, (b) middle layer and (c) bottom layer, and covered with 20 nm thick  $\text{TiO}_2$ .

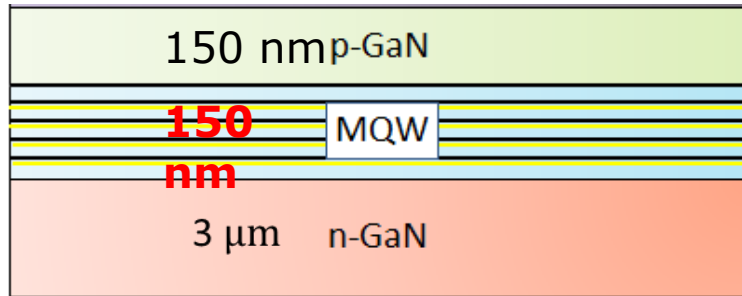
## SiCrystal samples



## Tankeblue samples

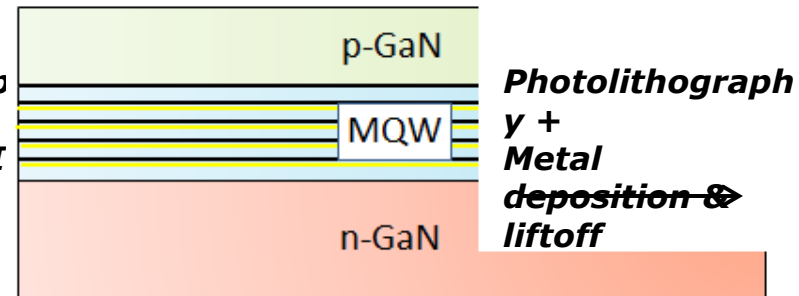


## Fabrication of LED devices

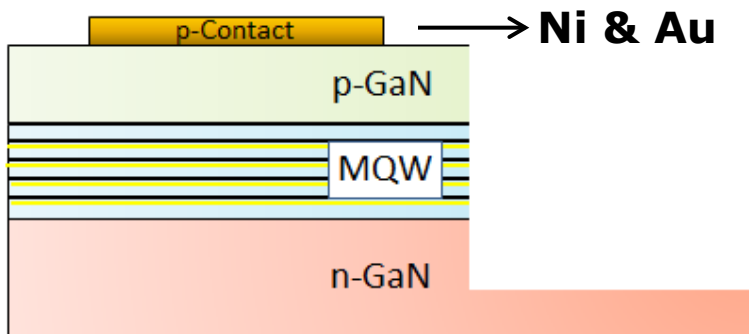


a) The GaN-based green LED epitaxial wafer

*Photolithography + Mesa etch by I*

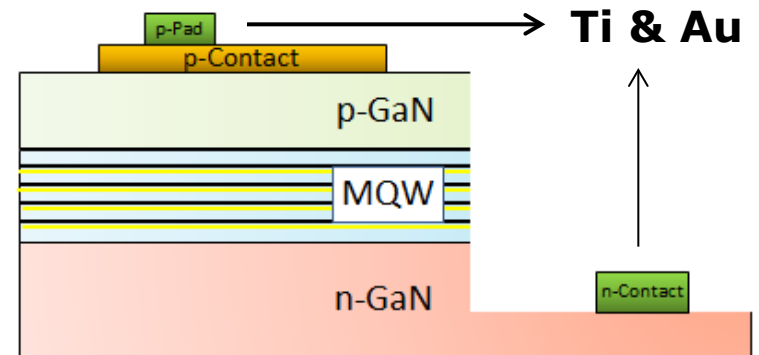


b) Mesa Formation to expose the n-GaN (etch depth of 500-600 nm)



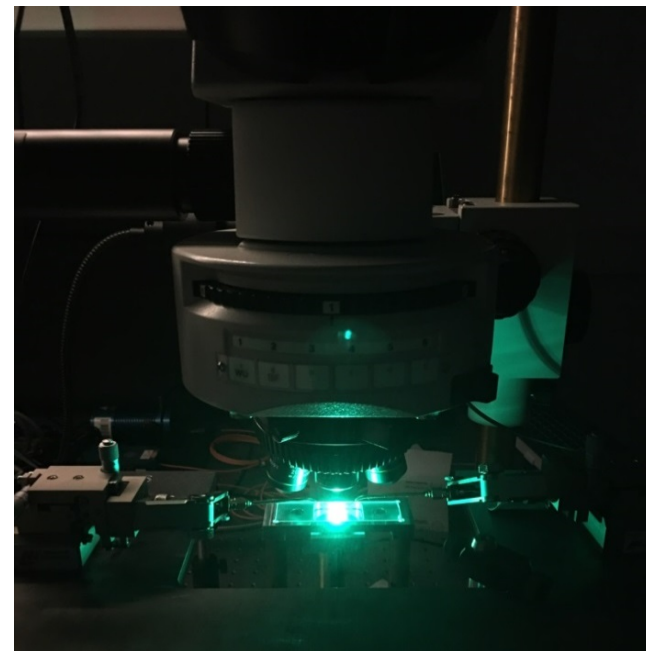
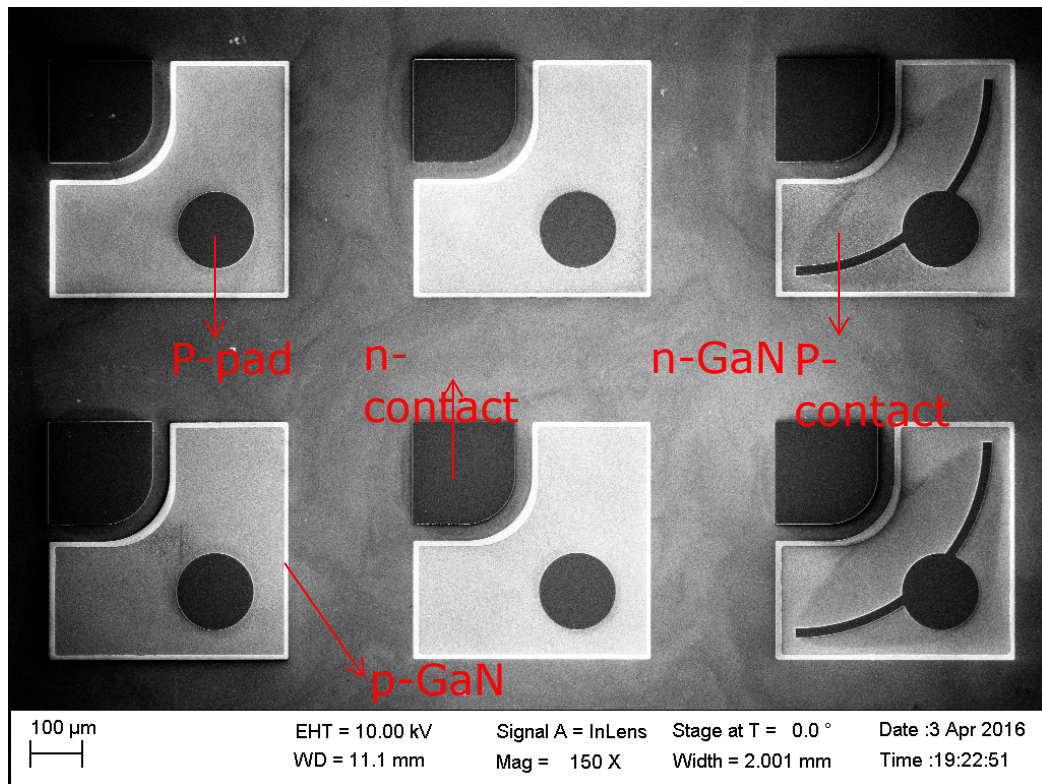
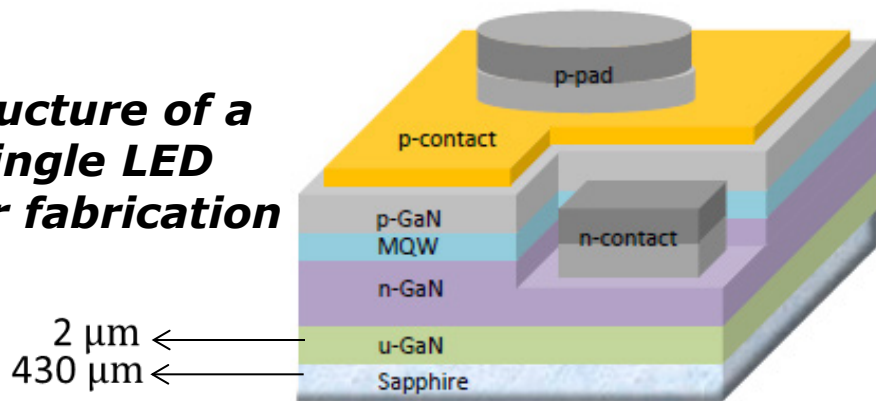
c) Formation of p-contact (current spreading layer) to p-GaN

*Photolithography Metal deposition liftoff*



d) Formation of p-pad to p-contact & n-contact to n-GaN (metal deposition at the same time)

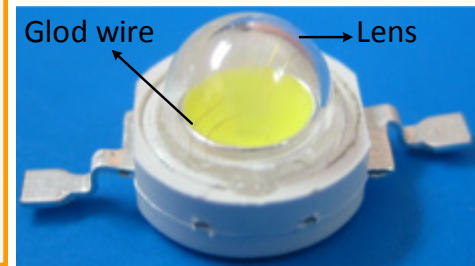
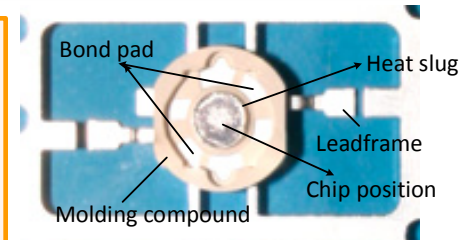
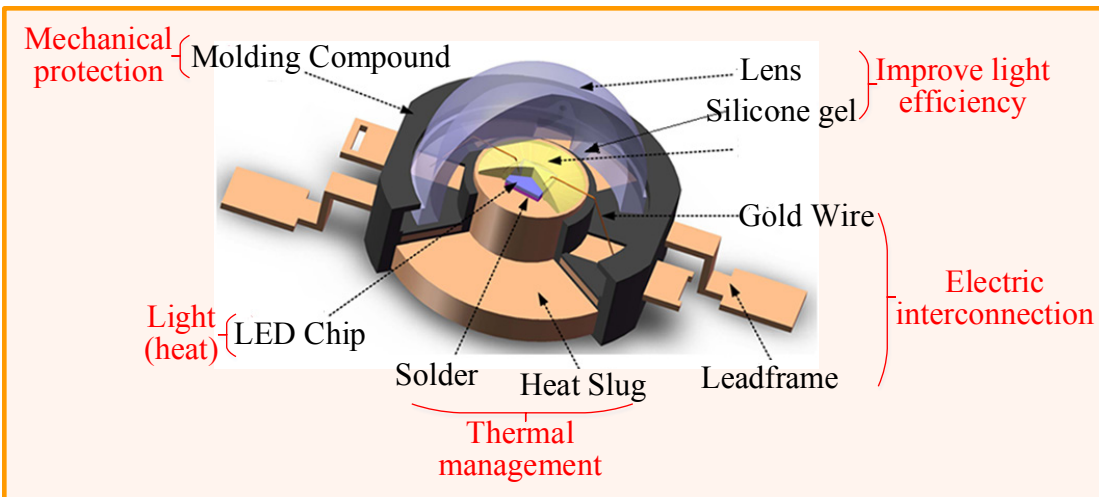
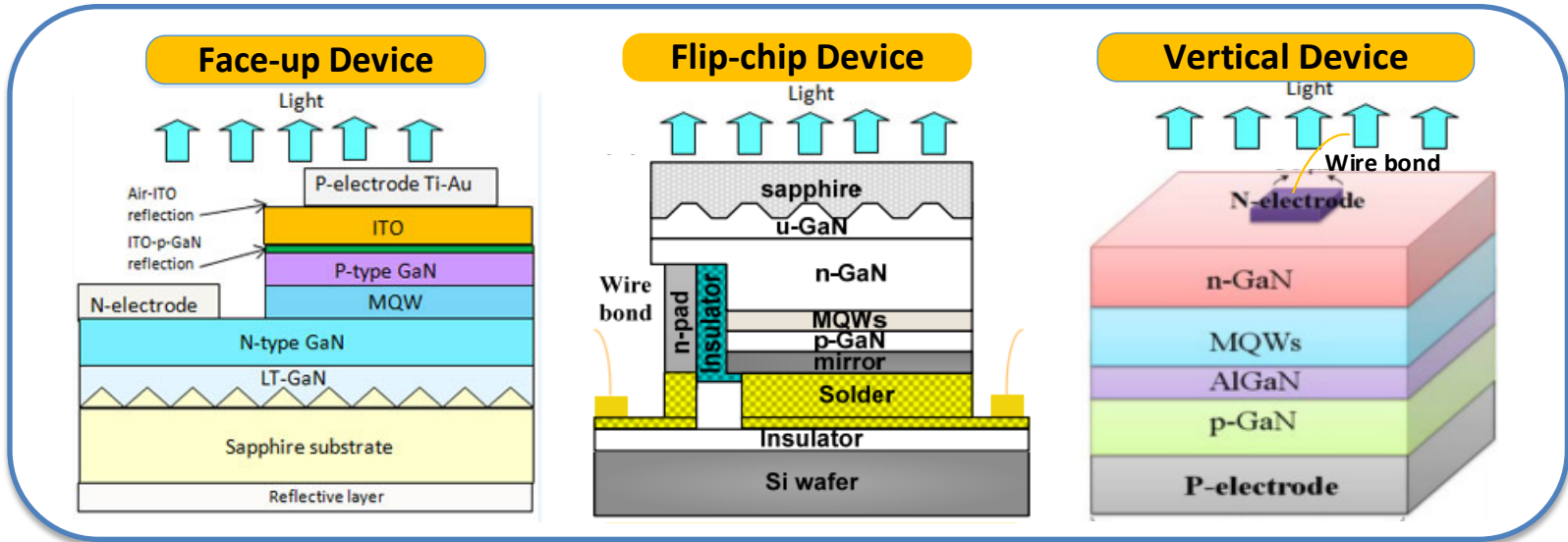
**Structure of a single LED after fabrication**



**Electro-luminescence of the LED devices**

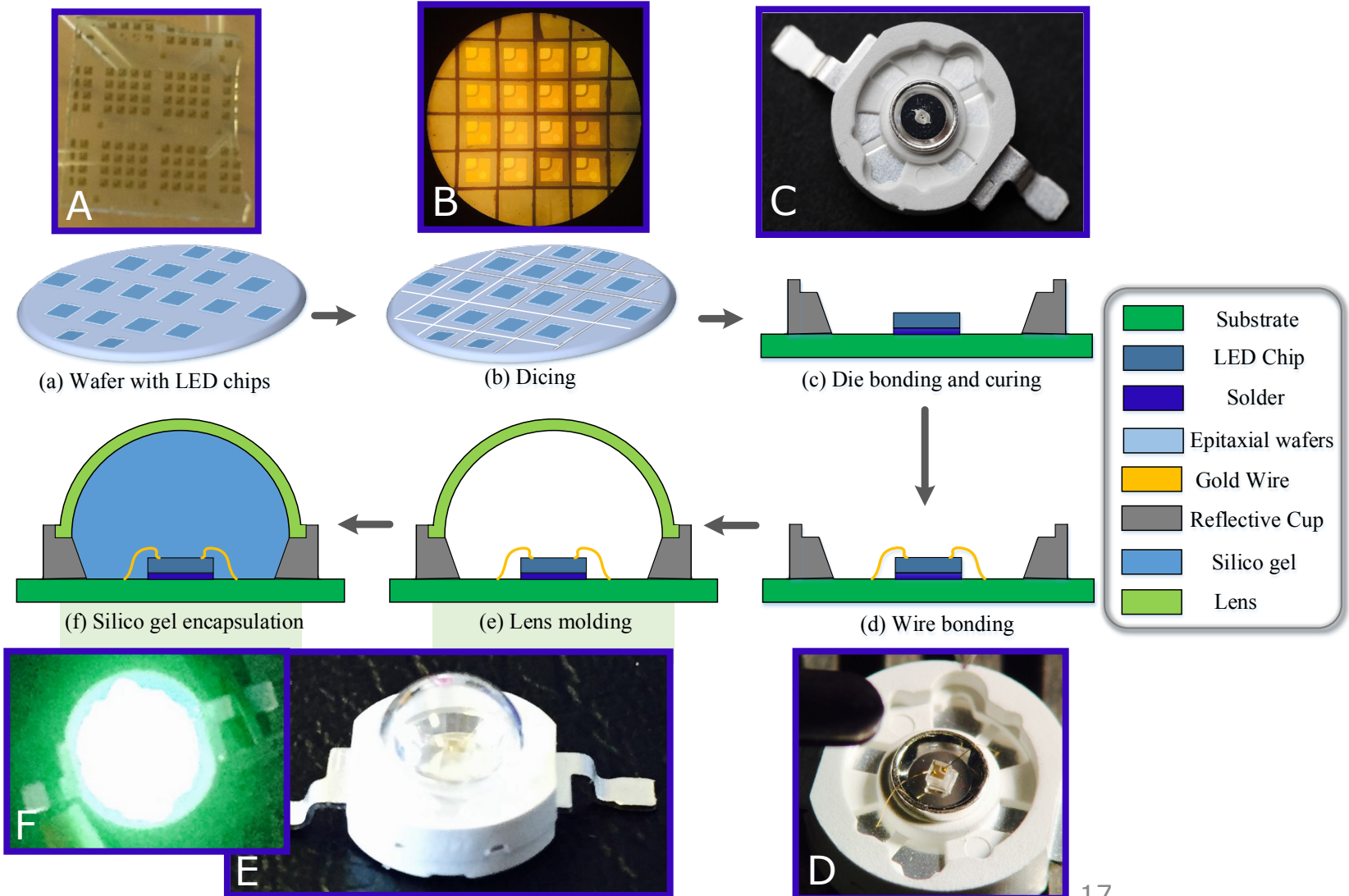
**← Top view of the LEDs by SEM**

# LED Structures:





# LED package processes:



# Current Challenges for discussion

- Metal surface adhesion for wire bonding
- Blue emission from porous SiC

# Advisory board

- Satoshi Kamiyama. Meijo University, Japan.
- ???

## Plan for the next 3 months:

- To conclude on the enhancement effect of surface plasmon: does it work? How much is the potential?
- Fabrication of NUV LED and improve the efficiency of the device by surface plasmons, nanopillars, and nanostructuring.
- To make blue emission out of porous SiC
- Make packaged device for demonstration

Funding application:

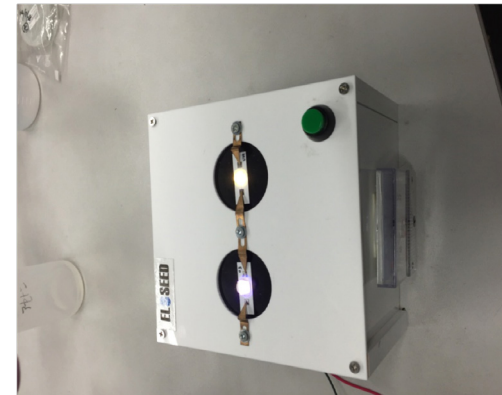
To secure a longer period for Jiehui and Ahmed

# Publications

- Yiyu Ou , Meng Xiong, Weifang Lu, Ahmed Fadil, Valdas Jokubavicius, Mikael Syväjärvi, Paul Michael Petersen, and Haiyan Ou, Efficiency enhancement of fluorescent SiC for white LED application, The 4th international Workshop on LEDs and Solar Applications, March 30-31, 2016, Nagoya, Japan.
- Haiyan Ou, Weifang Lu, Yiyu Ou, Valdas Jokubavicius, Mikael Syväjärvi, Philipp Schuh, Peter Wellmann, Yoshimi Iwasa, Satoshi Kamiyama, Passivation of surface-nanostructured f-SiC and porous SiC, The 4th international Workshop on LEDs and Solar Applications, March 30-31, 2016, Nagoya, Japan.
- 
- Submitted:
- Conference paper:
- 3. Weifang Lu, Yoshimi Iwasa, Yiyu Ou, Satoshi Kamiyama, Paul Michael Petersen, and Haiyan Ou. "Photoluminescence enhancement in porous SiC passivated by atomic layer deposited Al<sub>2</sub>O<sub>3</sub> films" (submitted at CLEO:2016- Laser Science to Photonic Applications)
- Journal papers:
- 4. Weifang Lu, Yiyu Ou, Valdas Jokubavicius, Ahmed Fadil, Mikael Syväjärvi, Paul Michael Petersen, and Haiyan Ou "Morphology and optical properties in nano-textured fluorescent 6H-SiC covered by atomic layer deposited titanium oxide" (submitted to Physica Scripta).
- 
- To be submitted:
- 5. Weifang Lu, Yiyu Ou, Paul Michael Petersen, and Haiyan "Passivation effect of atomic layer deposited Al<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> on photoluminescence of anodically etched porous 6H-SiC" (will be submitted at Optical Material Express Feature Issue: Two-Dimensional Materials for Photonics and Optoelectronics).

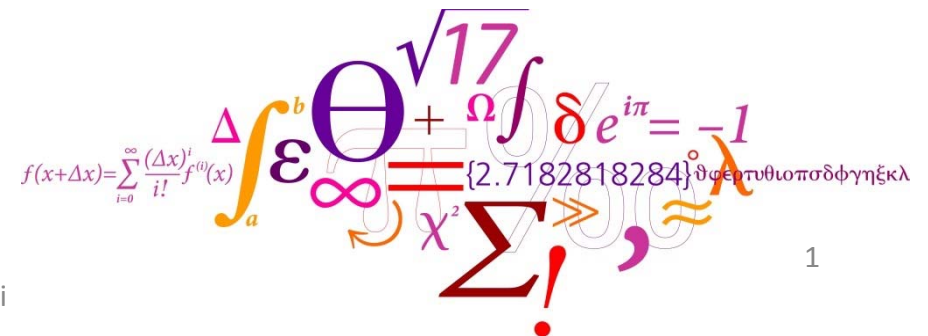
# Project management

Order	Meeting date and place	Participants	Program	Follow-up
1.	Kick-off, Sept. 2, 2015 DTU	All partners, Satoshi Kamiyama	refer	
2.	Dec. 11, 2015, DTU	Internal (2 PhD students + 1 postdoc + Haiyan)		<ol style="list-style-type: none"> <li>1. Report hand-in before March 1, 2016</li> <li>2. Li, Commercial LED epiwafer on SiC substrate</li> </ol>
3.	March 30, 2016, Meijo	Workpackage leader		
4.	April 6th, 2016, DTU	Danish partners		
5.	September xx, 2016, DTU	All partners		
6.	Dec. xx, 2016, DTU	Danish partners		






# LED package

Jiehui Li  
6<sup>th</sup> April 2016

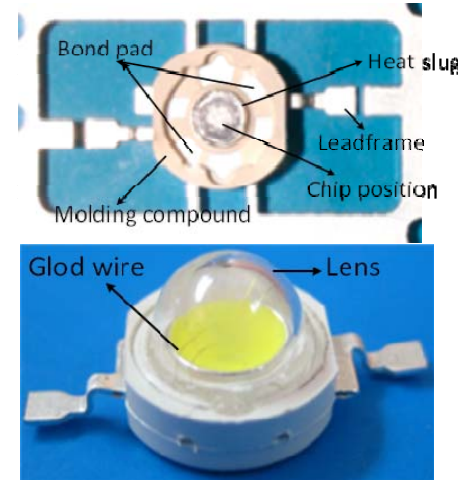
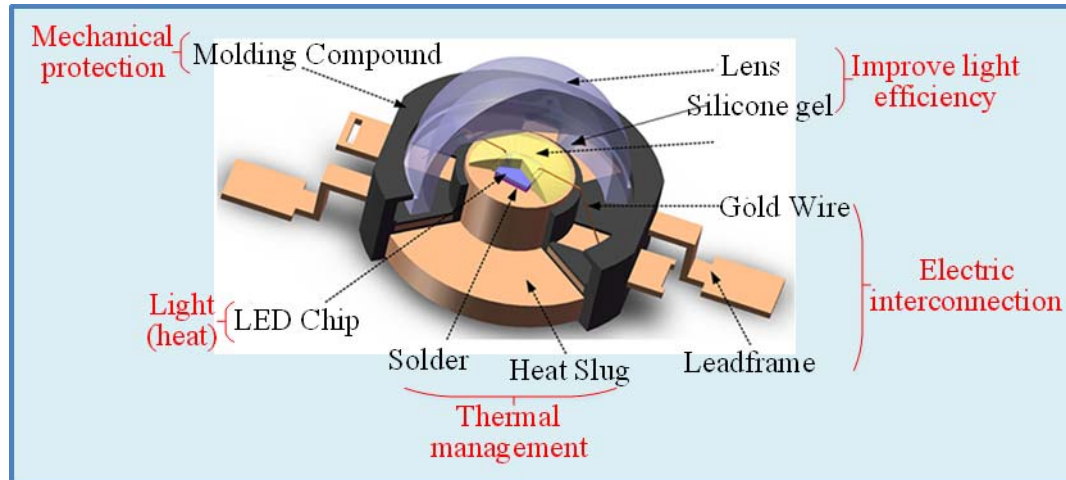
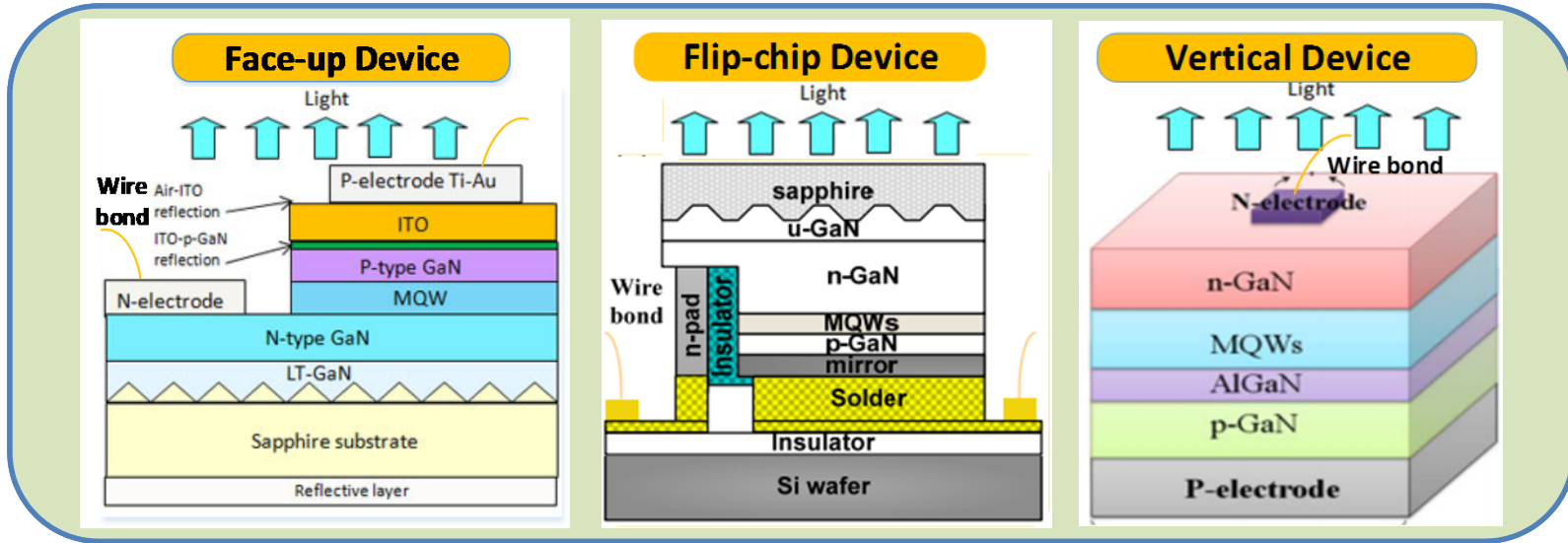


# Outline

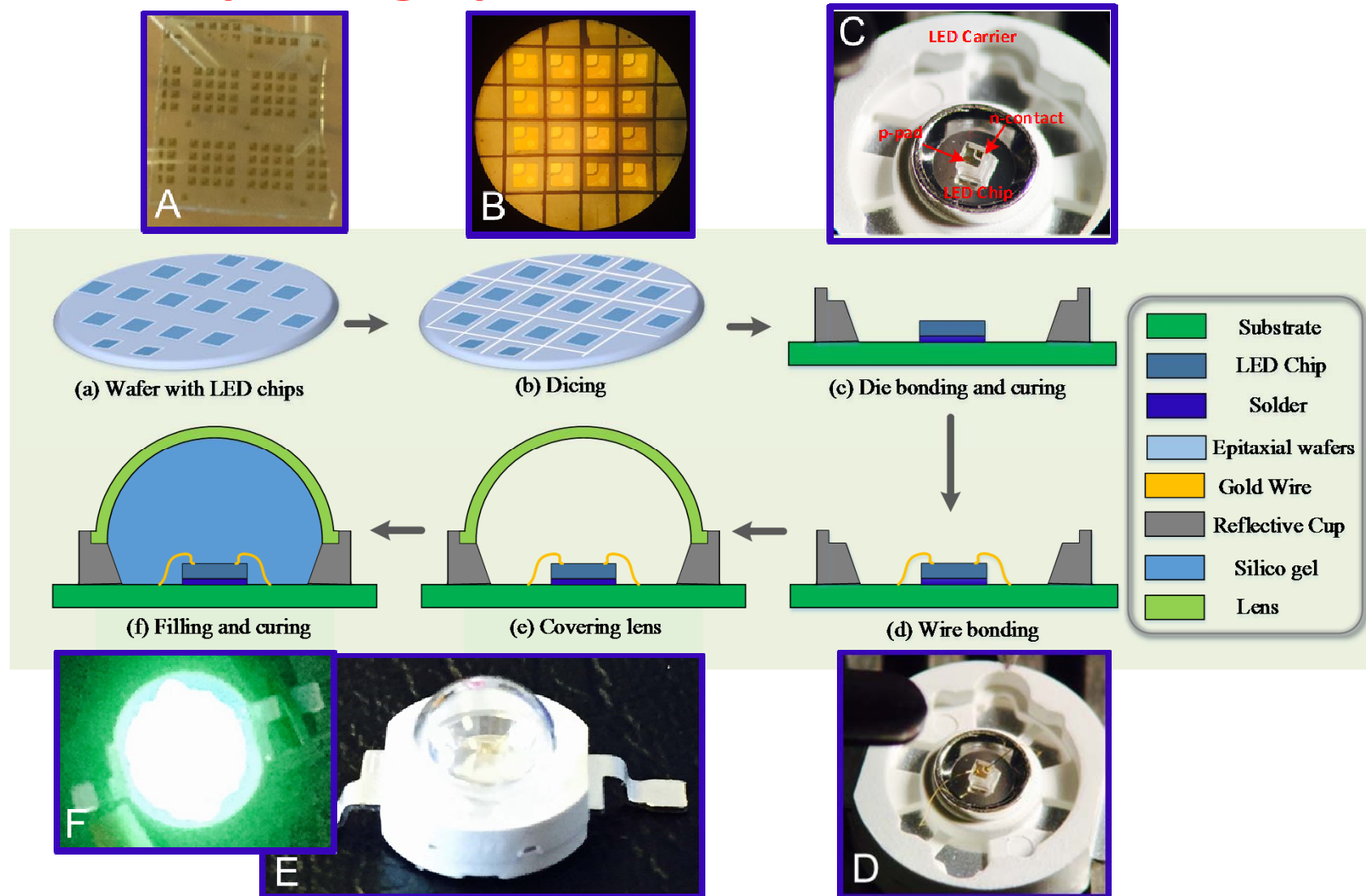
-  1 LED package structures
-  2 LED package processes
-  3 Summary



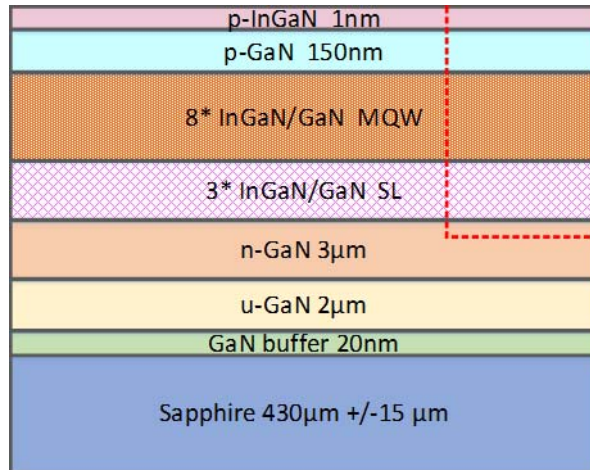
# LED package structures



# LED package processes:

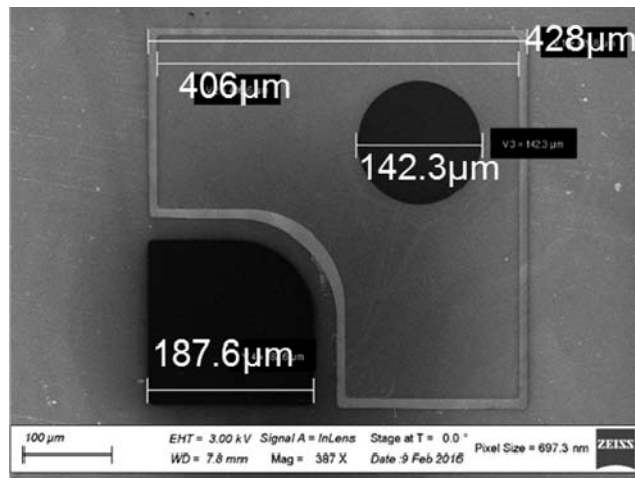


## The characterizations of GaN-LED

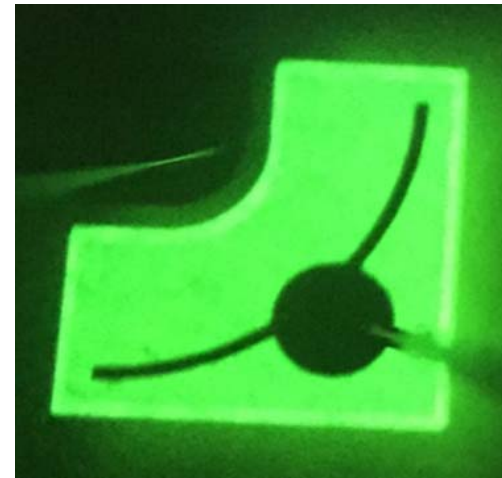


LED epitaxial structure

- Epitaxial: Commercial green GaN-LED
- Large LED chip size: 428 μm × 428 μm  
p-pad diameter: 142.3μm  
n-contact: 187.6μm
- P-contact: Ni/Au
- p-pad & n-contact: Ti/Au
- Turn-on voltage: about 2V @ 100mA

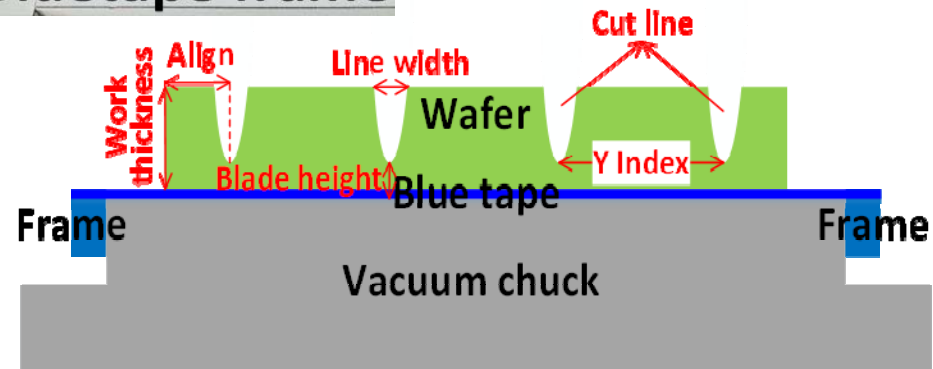
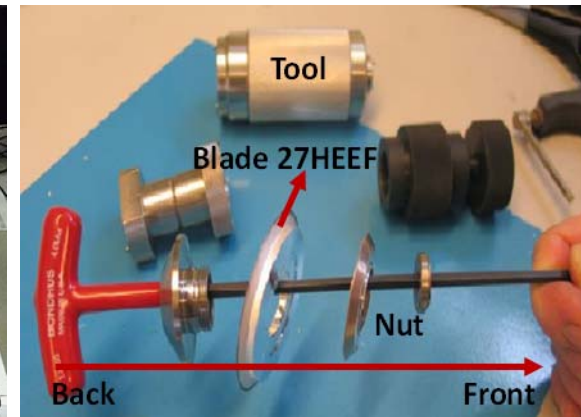
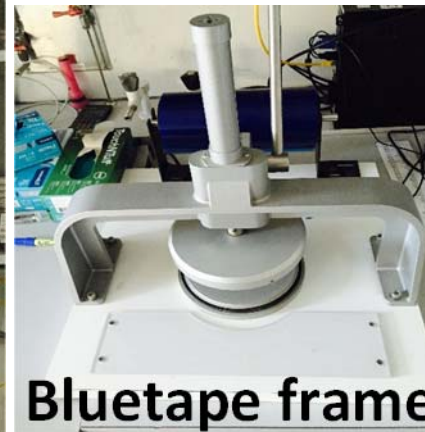
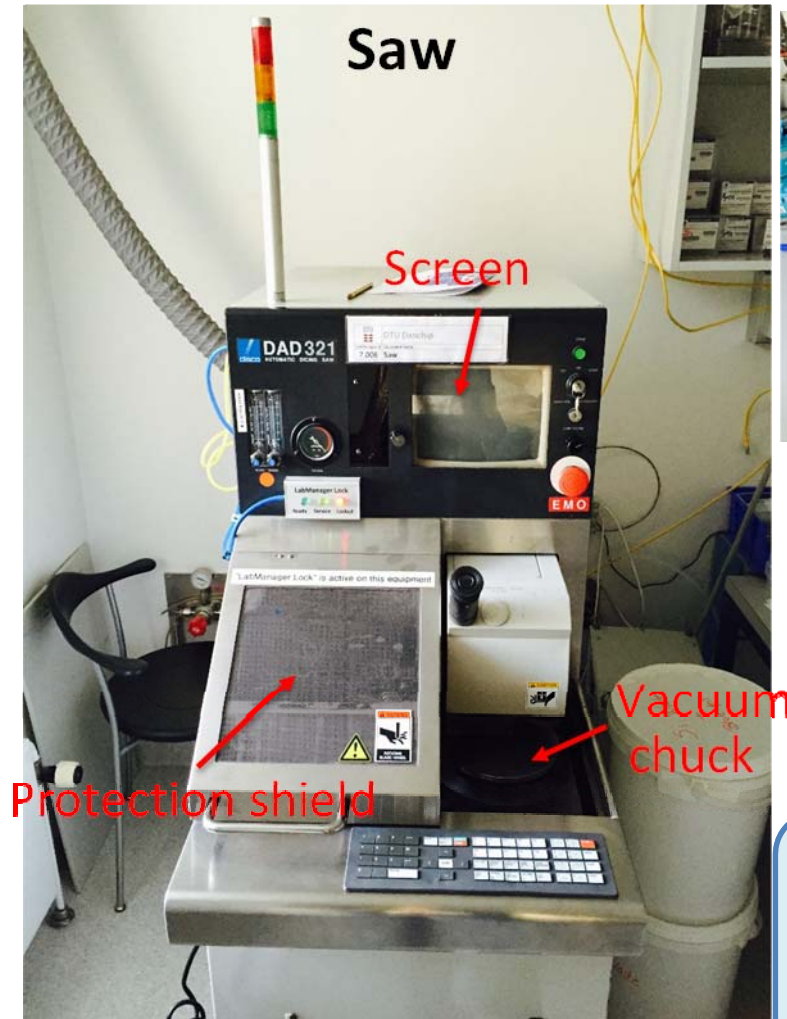


SEM image of Large-LED



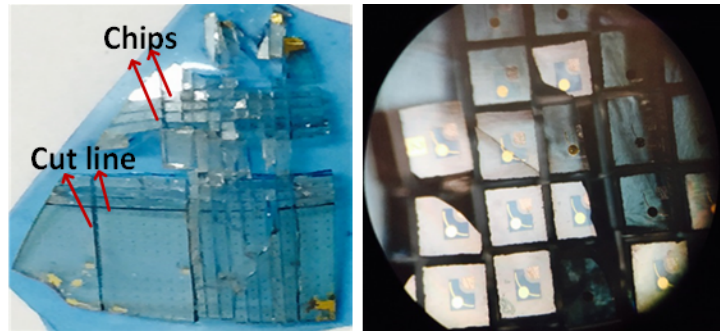
Normal LED (2V, 100mA)

## LED chip dicing by using Saw:



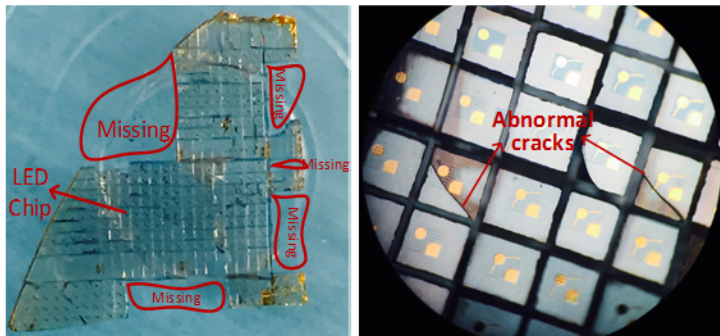
- Mechanical singulation
- Main parameters: **Blade height**, **Feed speed**, **Y-Index**, **Work thickness**, **Align**

## LED chip dicing:



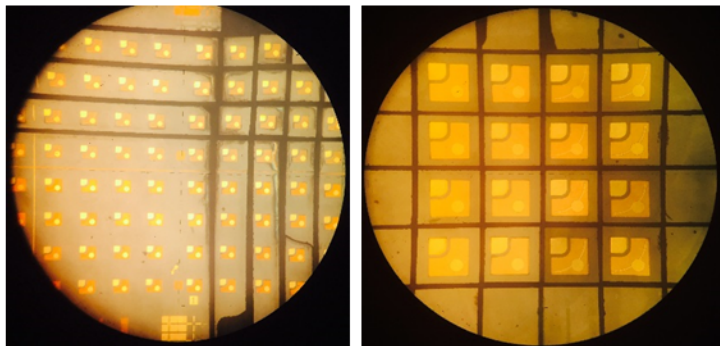
Blade height: 0.20mm  
Feed speed: 0.20mm/s

- ❖ Yield: about 30%-40%
- ❖ Unexpected line and dislocation occurred



Blade height: 0.25mm  
Feed speed: 0.15mm/s

- ❖ Better cutting effect has been achieved
- ❖ Yield: More than 90%

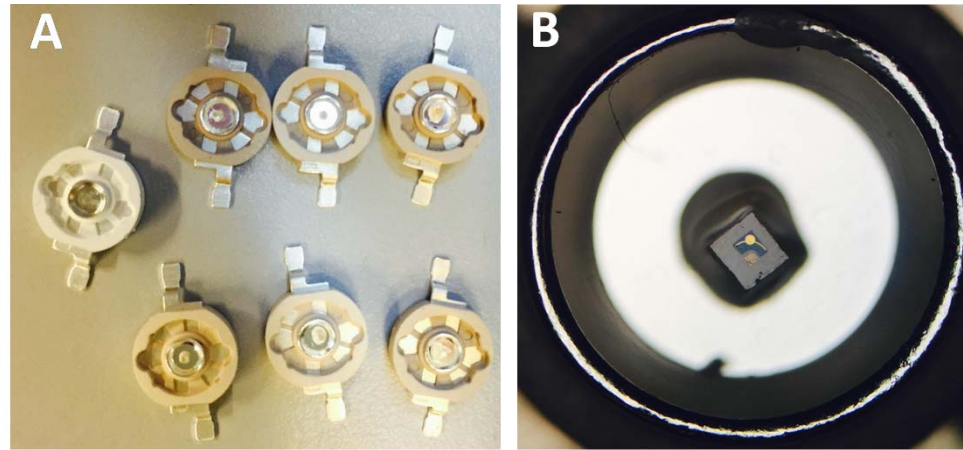


Blade height: 0.23mm  
Feed speed: 0.01mm/s (0.07mm/s)

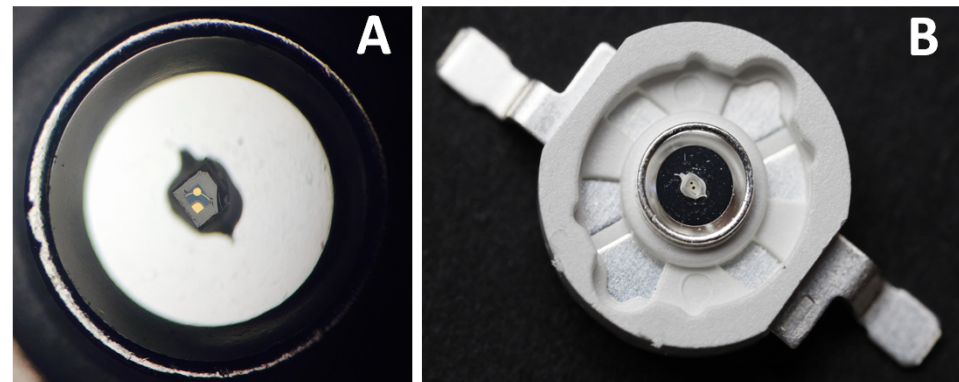
- ❖ Yield: near by 100%
- ❖ Much better than last time
- ❖ **Best parameters for dicing**

## Die bonding and curing:

- Clear epoxy die attach adhesive GCM-3100 was used to solder the GaN-LED chip on the carrier.
- Dispensing LED chip die bonding material on the hot plug.
- Take the LED chip from blue tape, and put it in the LED die bonding material on the hot plug.
- Curing on the hot plate.

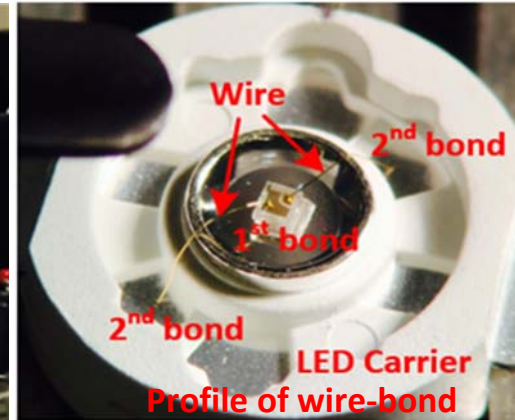
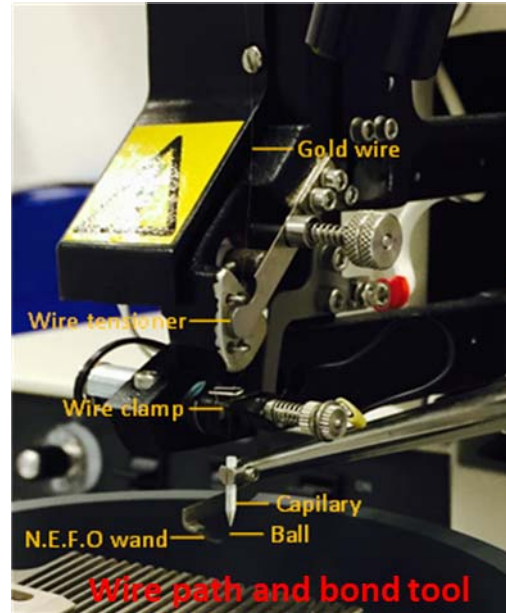
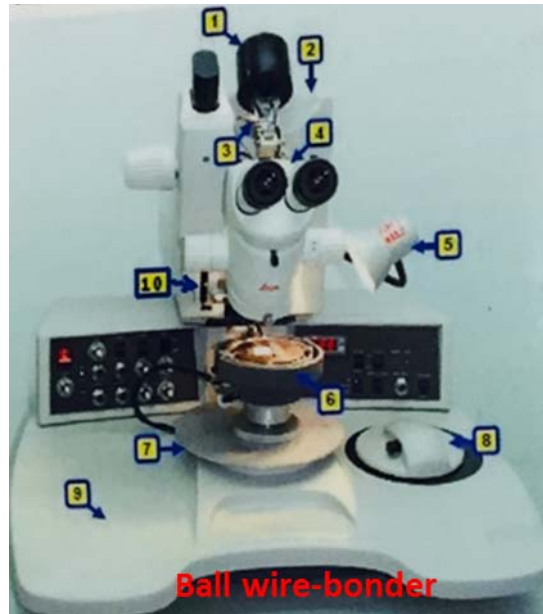


(Curing: 90min @ 170°C )



(Curing: 30min @ 120°C )

## Ball Wire-bonding:

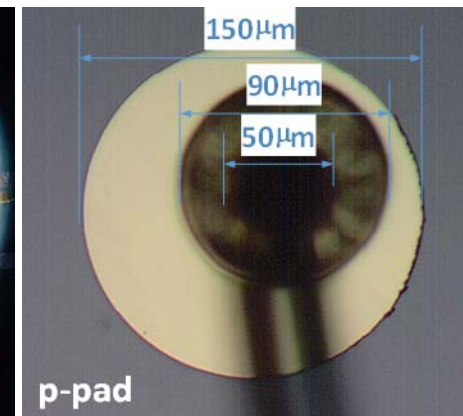
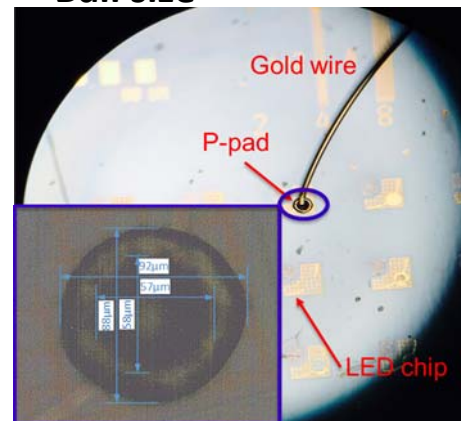


- Au wire diameter: 25 $\mu$ m
- Ball diameter: about 90 $\mu$ m

### Parameters setting:

1 <sup>st</sup> power	1	2 <sup>nd</sup> power	1.5
1 <sup>st</sup> Time	3	2 <sup>nd</sup> Time	3
1 <sup>st</sup> Force	2	2 <sup>nd</sup> Force	2
Loop	6	Tail	2.5
Ball size	4.4	Temperature	120°C

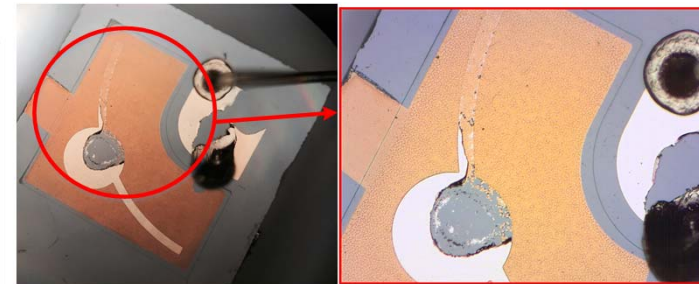
### Ball size



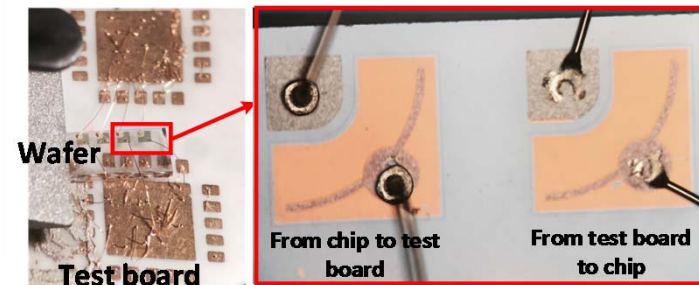
## Ball Wire-bonding:

**Metal adhesion** is the key to success for wire-bonding.

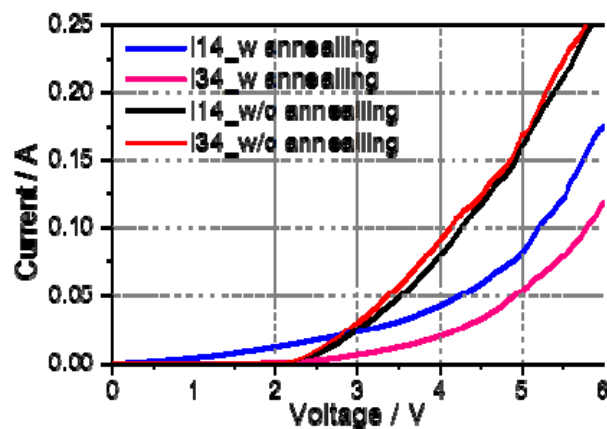
- The bonding was done by a combination of temperature, pressure and ultrasonic power applied to the wire.
- Before annealing, the p-pad is very easy to fall off from the p-contact.
- Annealing: 5min @ 500°C in N<sub>2</sub>
- The wires are fine when bonding from test board to chip or from chip to test board.



Before annealing (30nm Ti / 400nm Au)



After annealing (5min @ 500°C in N<sub>2</sub>)

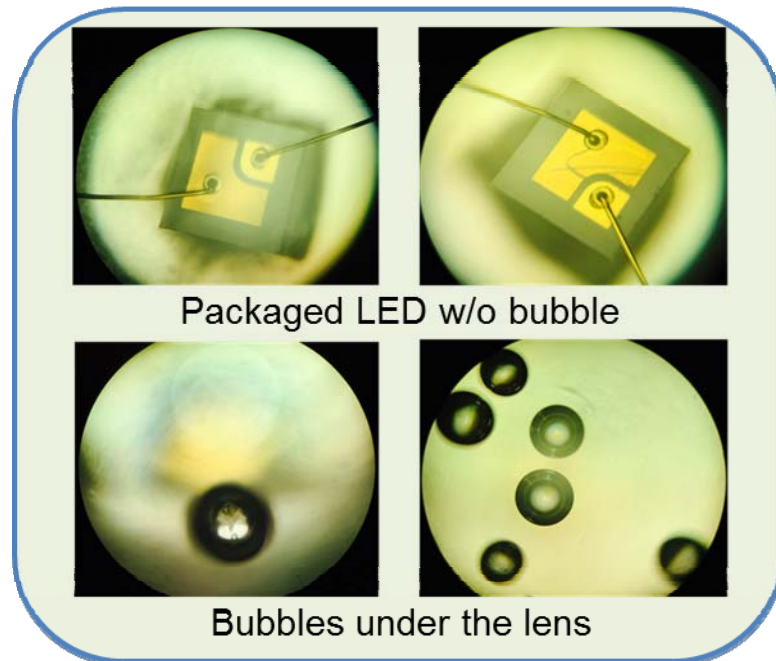


- **Annealing** is one of the main methods to enhance the metal adhesion.
- But the performance of LED will become weakened after annealing.



## Filling and curing:

- The lens was covered on the LED chip after wire-bonding.
- The silicone gel was filled into the slit slowly between the chip and lens from the small hole in the edge of lens.
- Curing: 1h @ 80°C in the oven.



### Hazard bubbles:

- Hinder electron transport
- Damage the internal structure
- Paralysis of lighting system, Leakage current and dead lights

### Avoid bubbles:

- Before filled, remove all the bubbles of silica gel via placed in the desicator for about 1-2h.
- The speed of filling is slower.

## □ Summary

### ❖ *The package processes for face-up LED device have been completed.*

But the processes still need to be optimized.

- Metal adhesion need to be enhanced for wire-bonding.
- Avoid bubbles when filling silicon gel and curing.

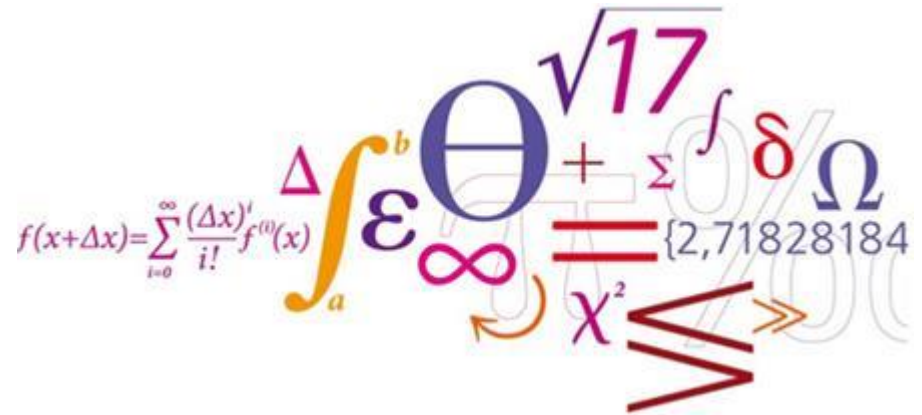
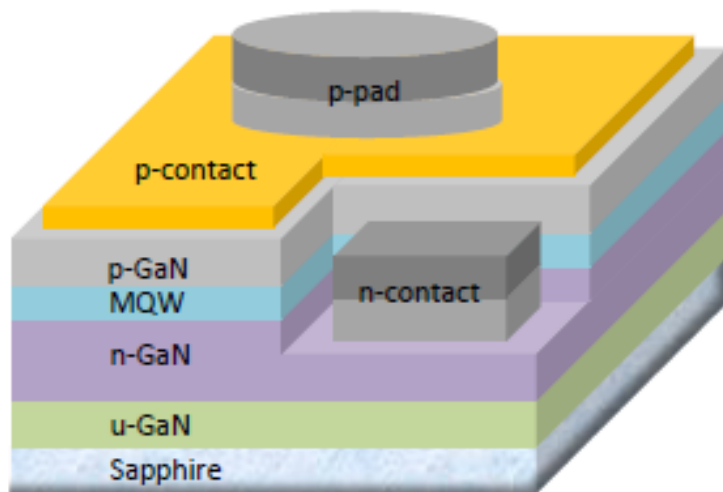
### ❖ *Next work:*

- Optimize the device process for wire-bonding.
- Improve the performance of LED device.
- Optimize the package processes for the nanostructure LED device.



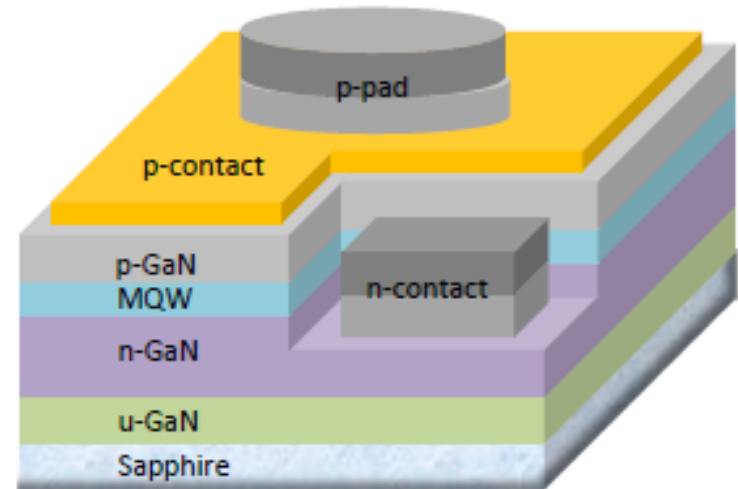
# Fabrication and characterization of GaN LEDs

*Li Lin*



## Outline:

1. Fabrication process of GaN LEDs
2. Characterization of GaN LEDs
3. Adhesion of metal contacts
4. Conclusion
5. Future plans



## Outline:

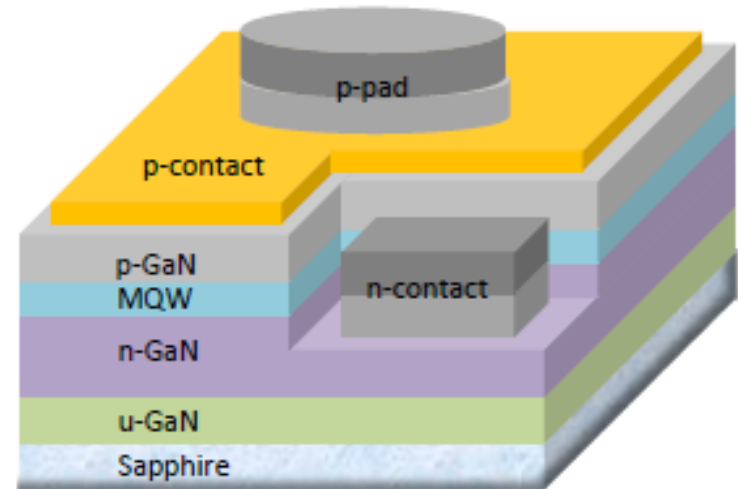
### 1. Fabrication process of GaN LEDs

### 2. Characterization of GaN LEDs

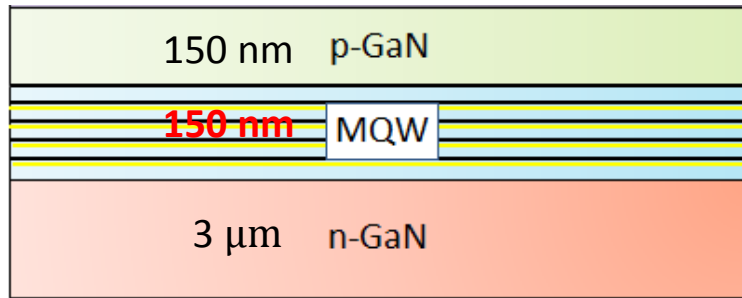
### 3. Adhesion of metal contacts

### 4. Conclusion

### 5. Future plans

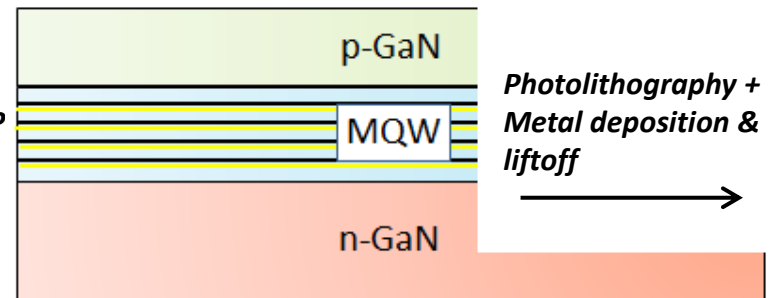


## Fabrication of GaN LEDs

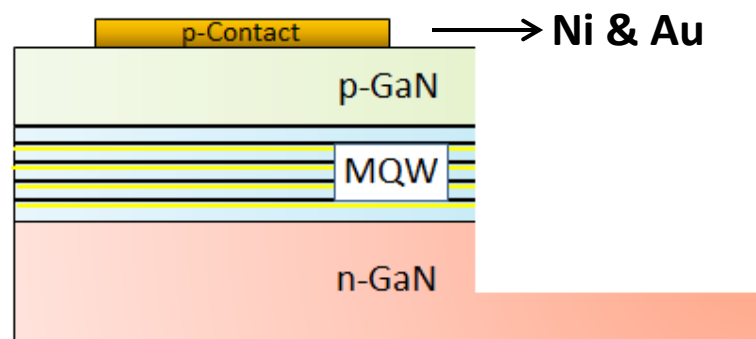


a) The GaN-based Green LED Epitaxial Wafer

*Photolithography + GaN mesa etch by ICP*

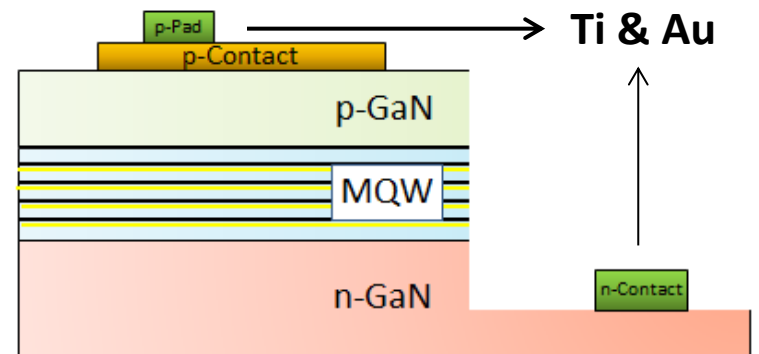


b) Mesa Formation to expose the n-GaN (etch depth of 500-600 nm)



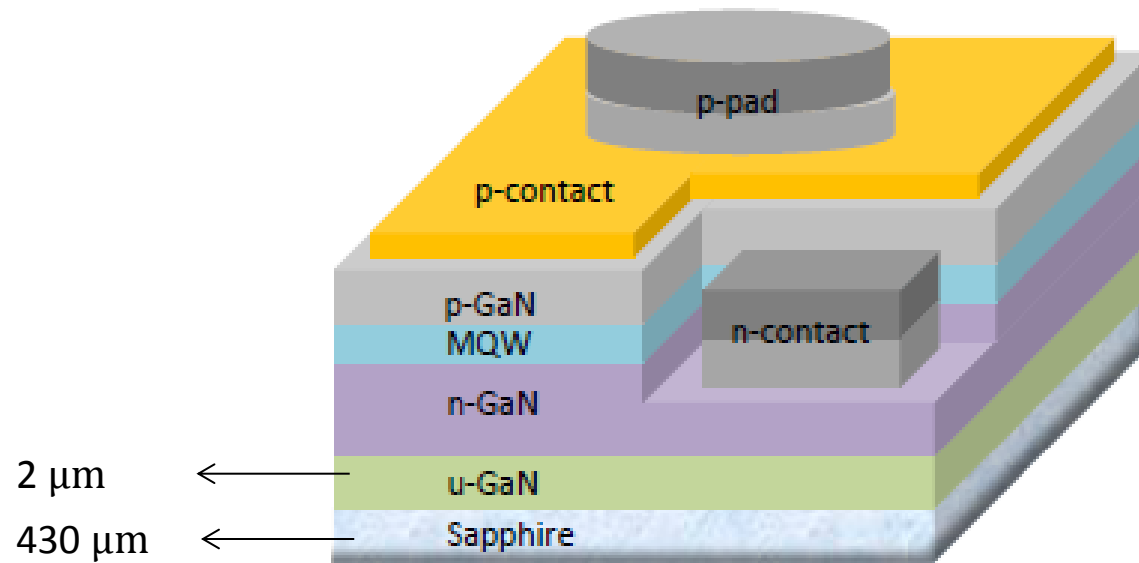
c) Formation of p-contact (current spreading layer) to p-GaN

*Photolithography + Metal deposition & liftoff*



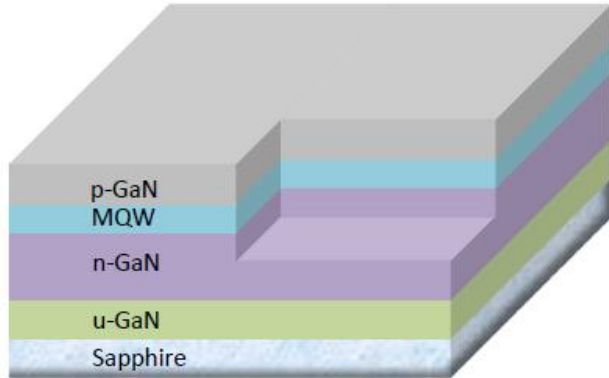
d) Formation of p-pad to p-contact & n-contact to n-GaN (metal deposition at the same time)

## Final structure of a GaN LED

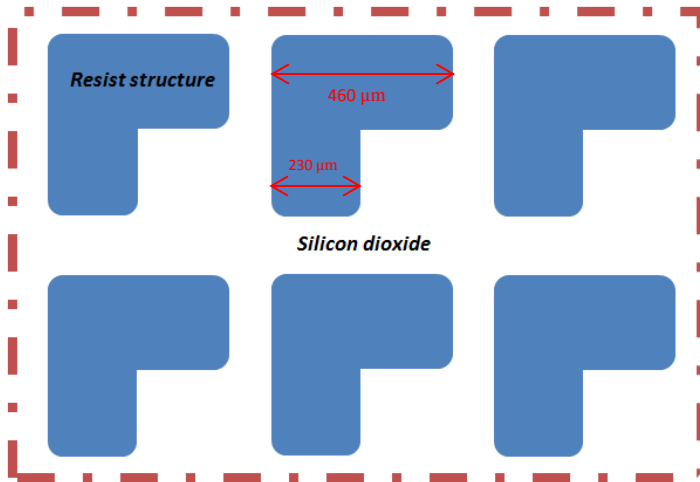




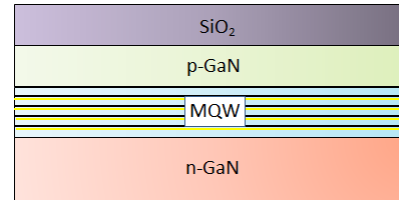
# 1. Formation of GaN mesas



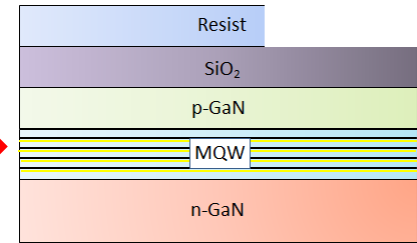
GaN mesa after fabrication



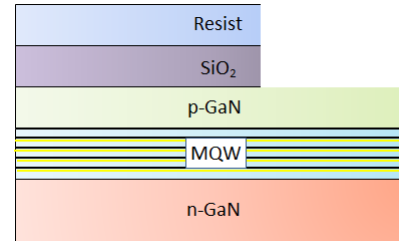
The resist pattern for formation of silicon dioxide mesas



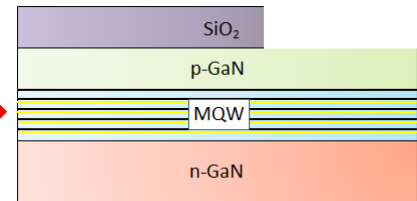
a) Deposition of 200-nm silicon dioxide layer by PECVD



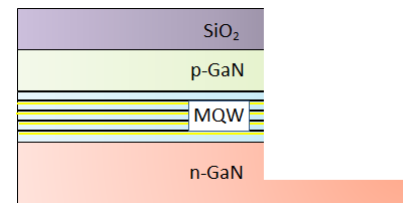
b) Formation of resist pattern by positive photolithography



c) Formation of silicon dioxide mesa by RIE and 5% HF

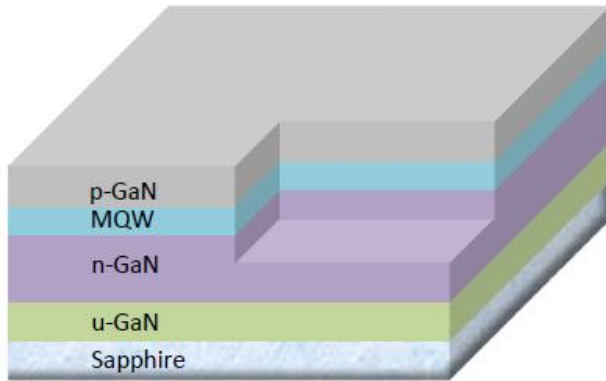


d) Removal of resist layer by acetone and oxygen plasma

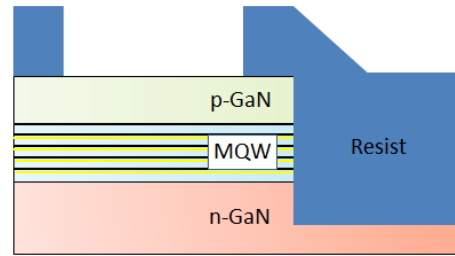


e) Etch of GaN mesa by ICP and then remove silicon dioxide by 5% HF

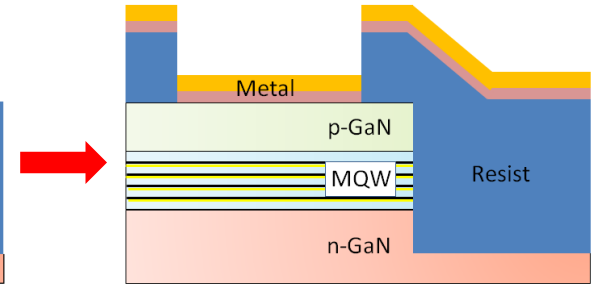
## 2. Formation of p-contacts



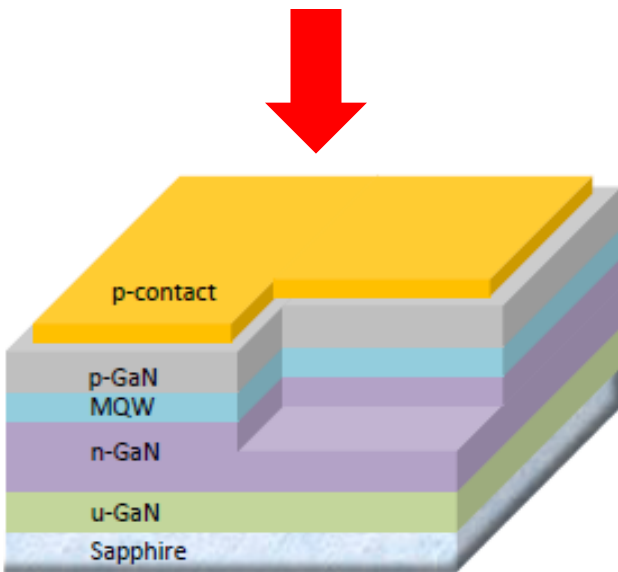
*GaN mesa after fabrication*



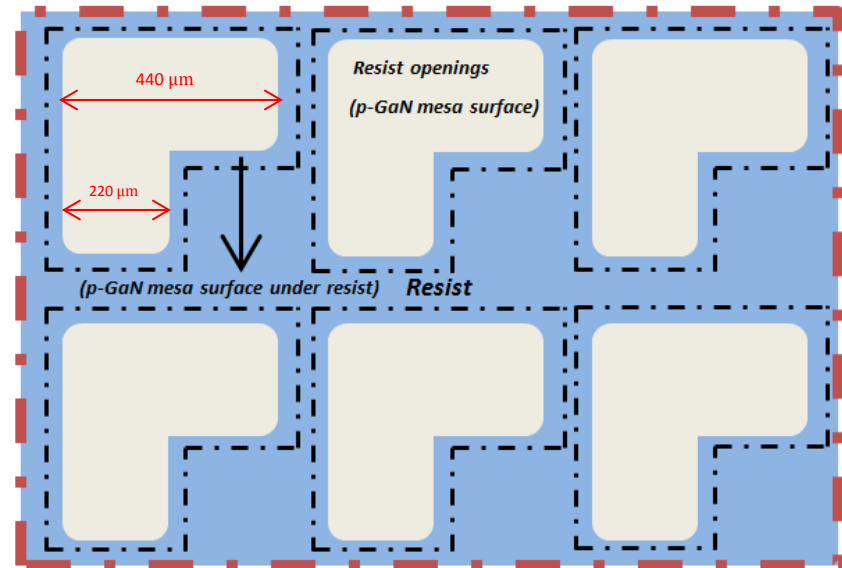
a) Formation of resist pattern by negative photolithography



b) Metal deposition of 10-nm Ni and 40-nm Au by e-beam evaporation and then liftoff

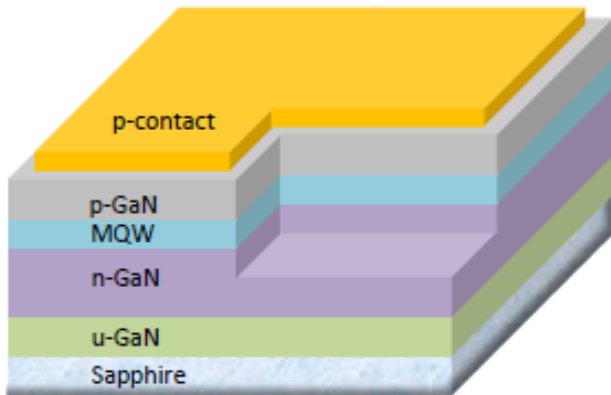


*P-contact after fabrication*

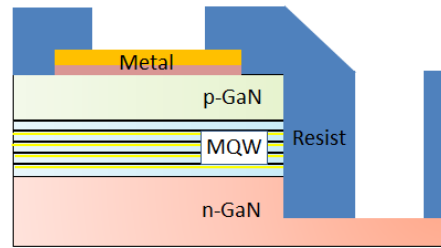


*The resist pattern for p-contact metal deposition*

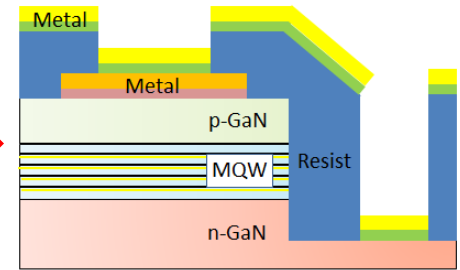
### 3. Formation of p-pads and n-contacts



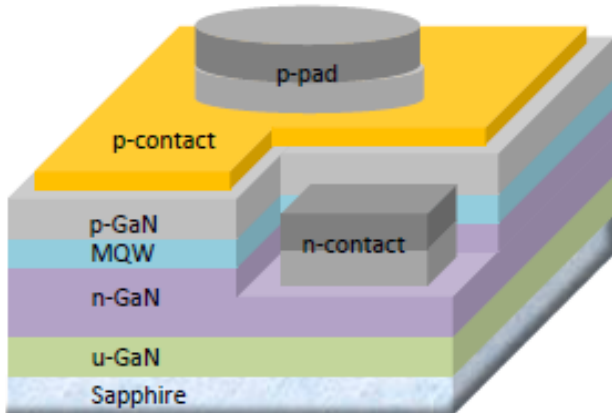
*P-contact after fabrication*



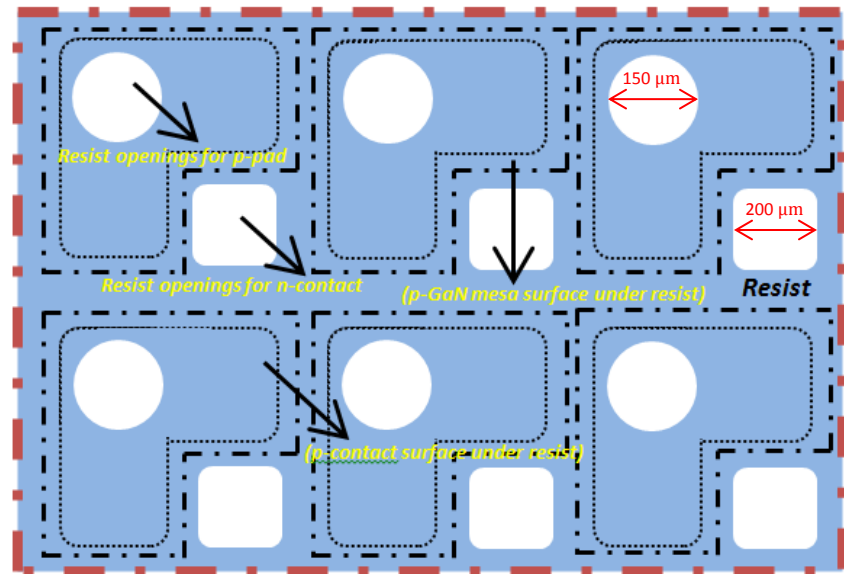
a) Formation of resist pattern by negative photolithography



b) Metal deposition of 30-nm Ti and 200-nm Au by e-beam evaporation and then liftoff by ultrasonic



*n-contact & p-pad after fabrication*



*The resist pattern for n-contact & p-pad metal deposition*

## Outline:

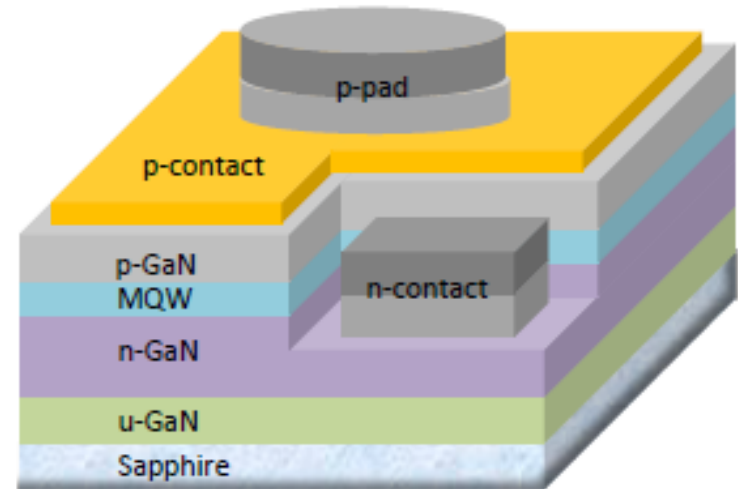
### 1. Fabrication process of GaN LEDs

### 2. Characterization of GaN LEDs

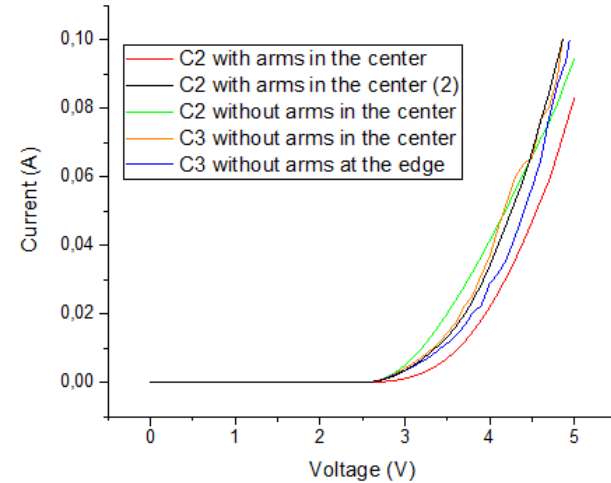
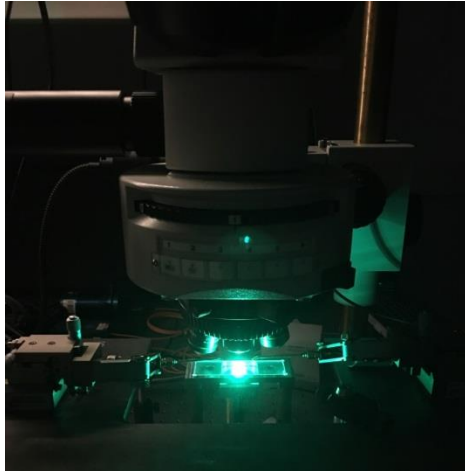
### 3. Adhesion of metal contacts

### 4. Conclusion

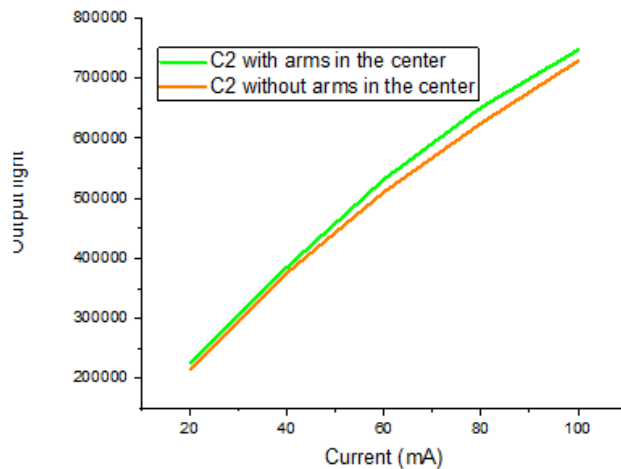
### 5. Future plans



## Characterization of GaN LEDs

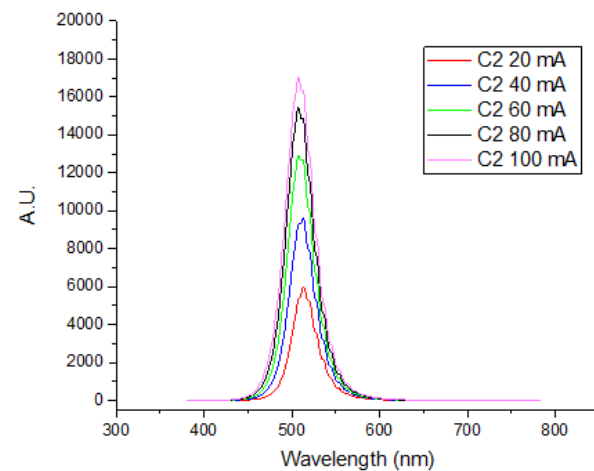


*Electro-luminescence of the LED devices*



**Power vs current**

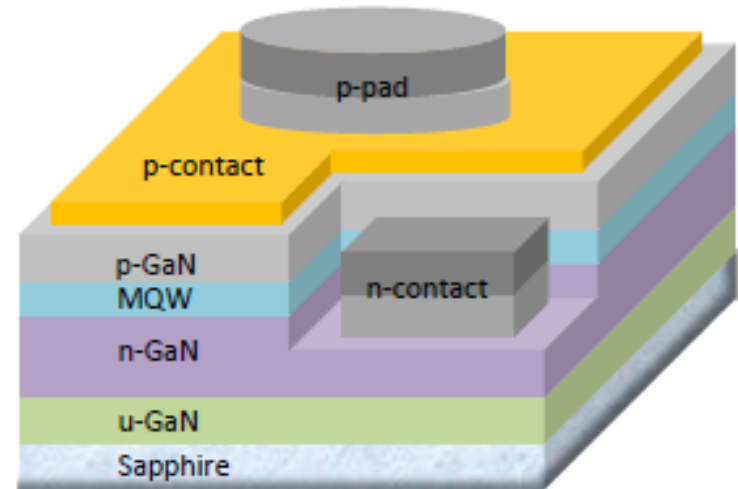
*I-V Curve of GaN Green LED on Sapphire*



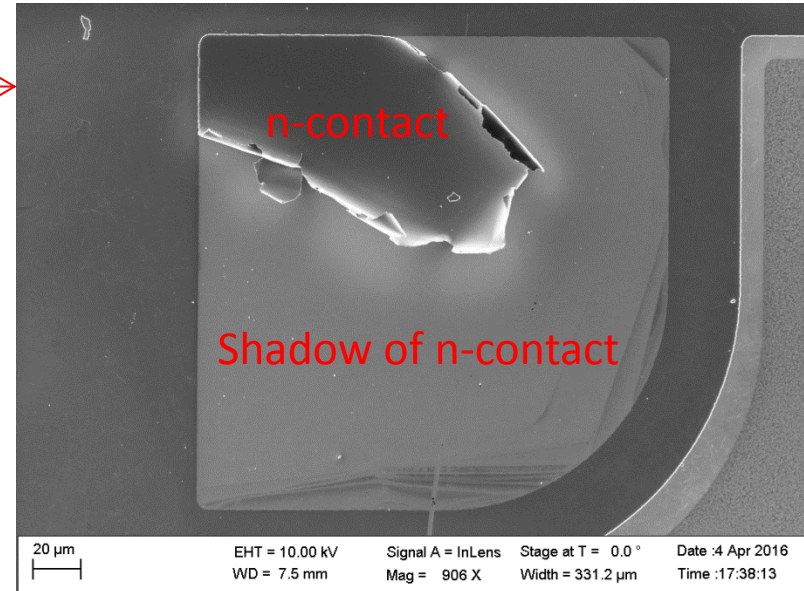
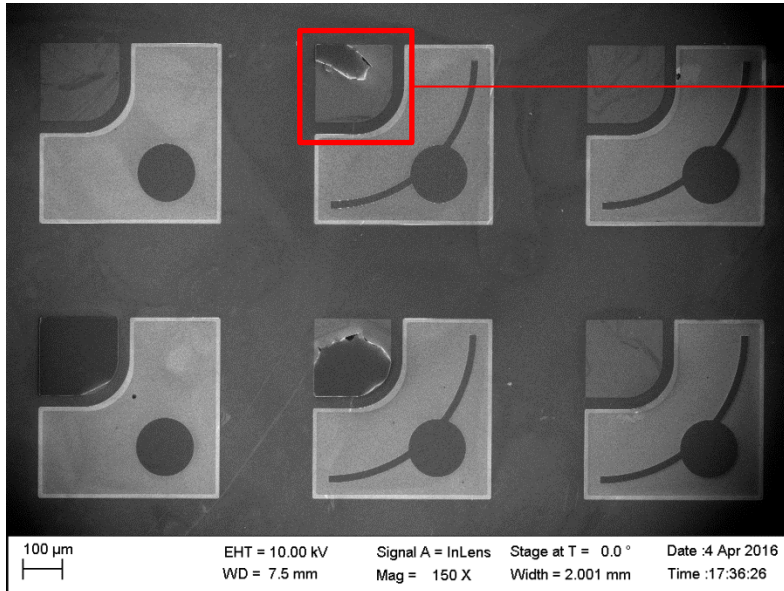
**E-L Curve of GaN Green LED on Sapphire**

## Outline:

1. Fabrication process of GaN LEDs
2. Characterization of GaN LEDs
3. Adhesion of metal contacts
4. Conclusion
5. Future plans



## Bad adhesion of n-contacts



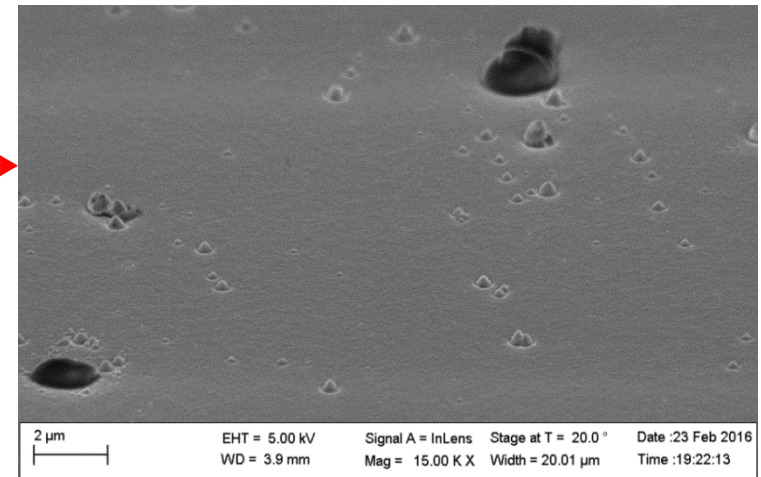
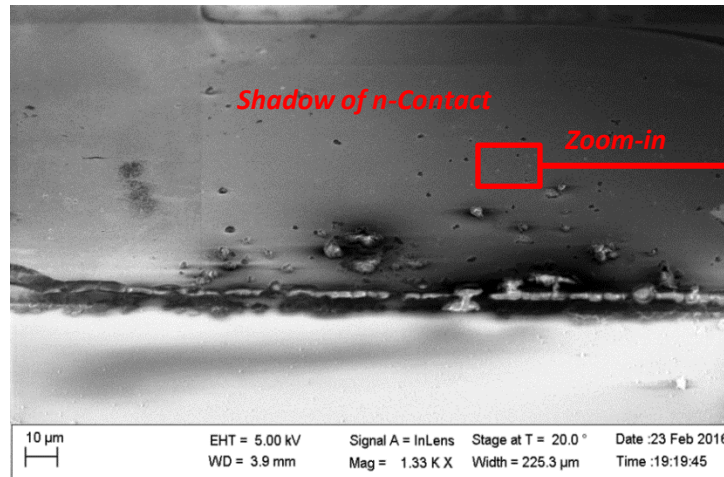
**After liftoff assisted with ultrasonic for removal of 30-nm Ti & 200-nm Au:**

- Most of n-contacts flaked off (about 10 survived in 96 LEDs)
- The shapes of surviving n-contacts are not complete

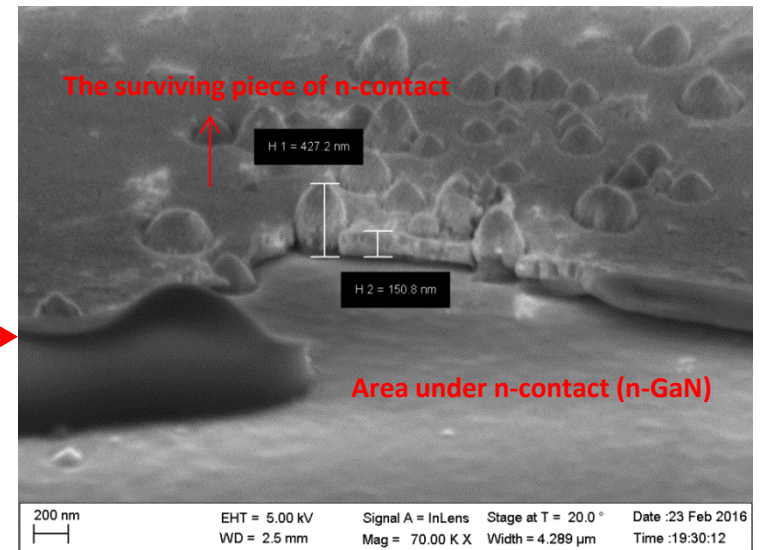
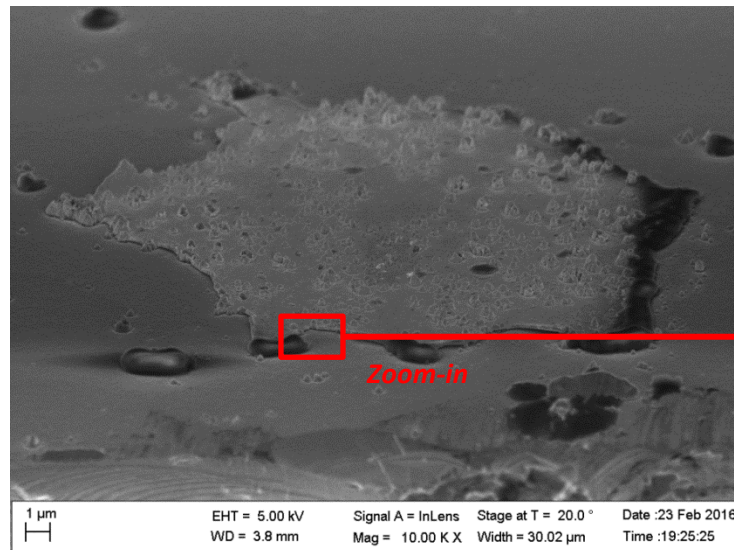
**Reasons behind the bad adhesion:**

- *Surface roughness (especially the etched surface of n-GaN)*
- Residual resist after development in the areas for n-contact metal deposition
- The GaN surface is not clean enough

## *Cones in the etched areas*

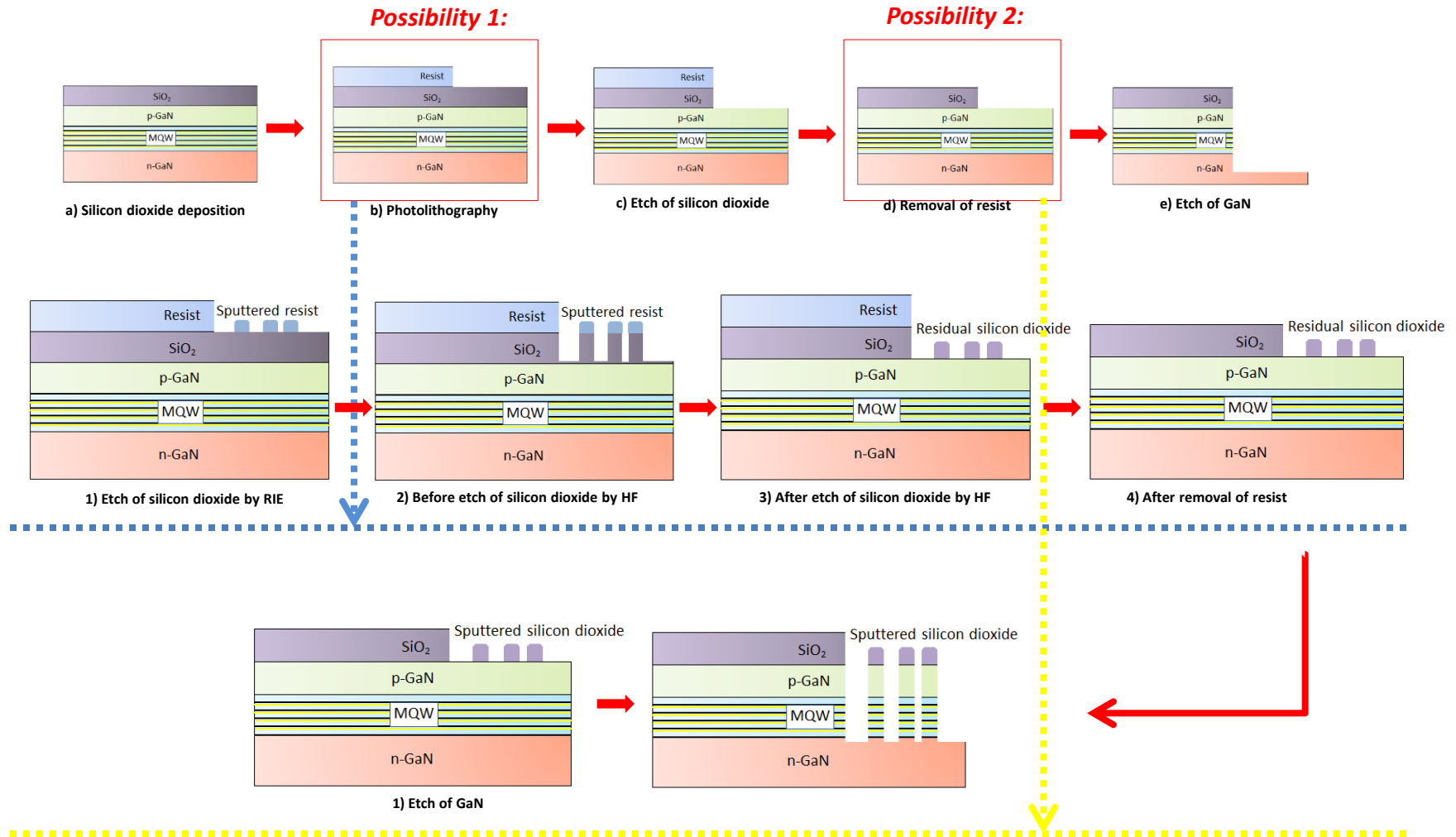


*Many cone structures can be observed by SEM in the etched areas including the n-contact areas*



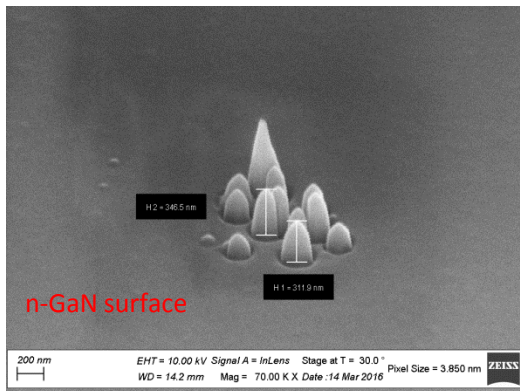
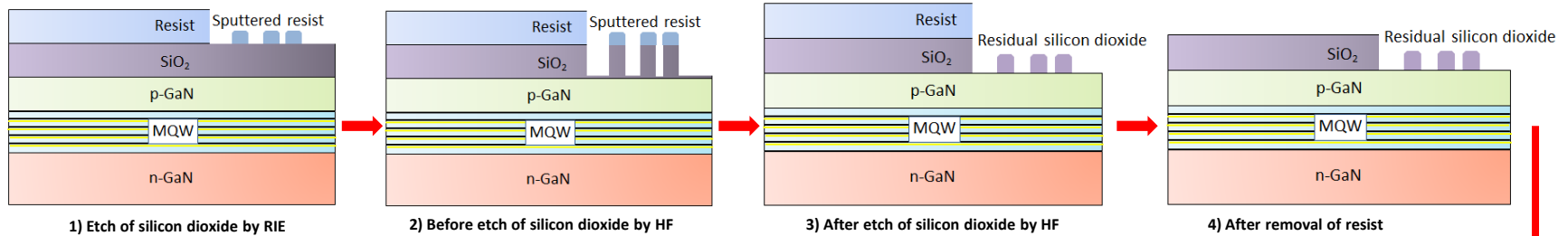


## Reasons behind the cone structures in the etched areas:

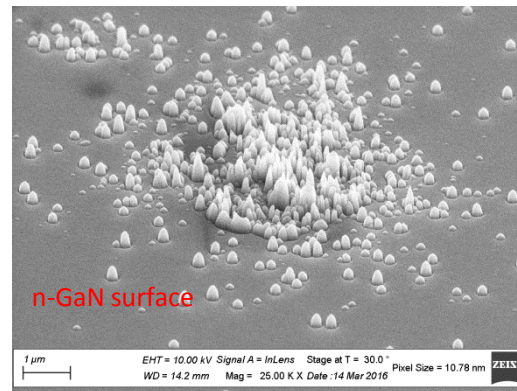


## Reasons behind the cone structures in the etched areas:

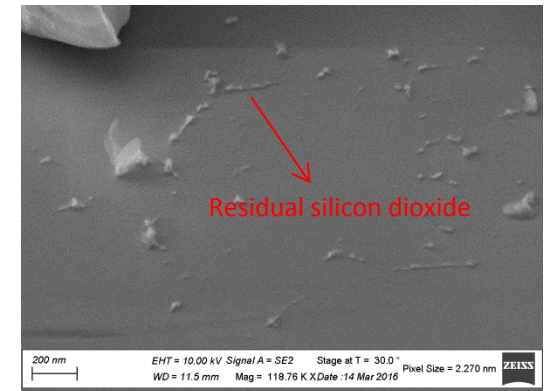
### Possibility 1:



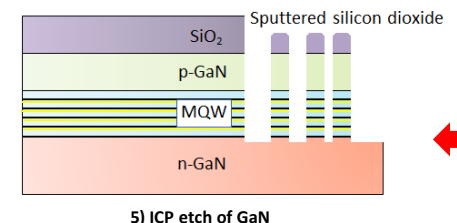
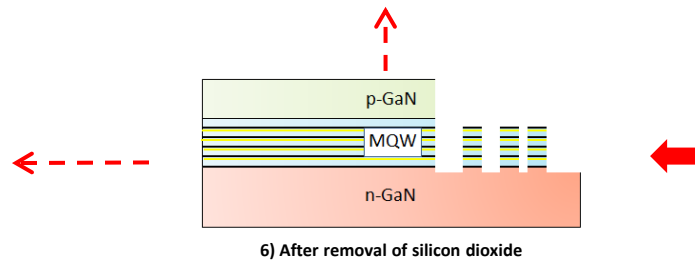
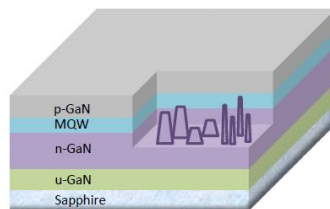
The heights of the cones are roughly several hundred of nanometers, which is comparable to the etch depth of GaN



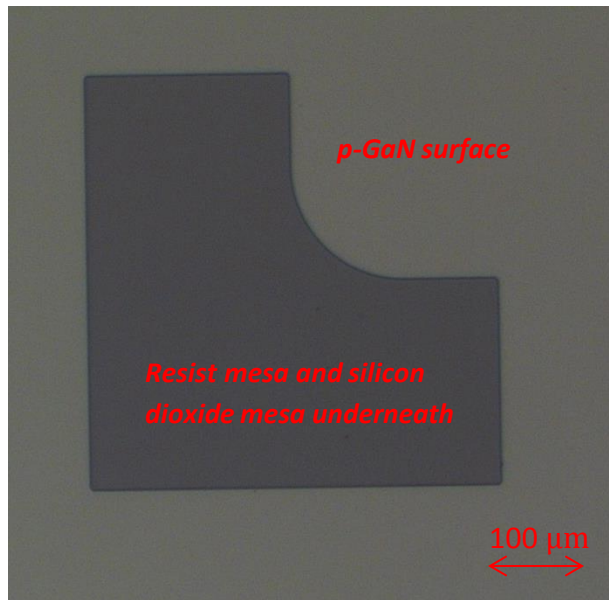
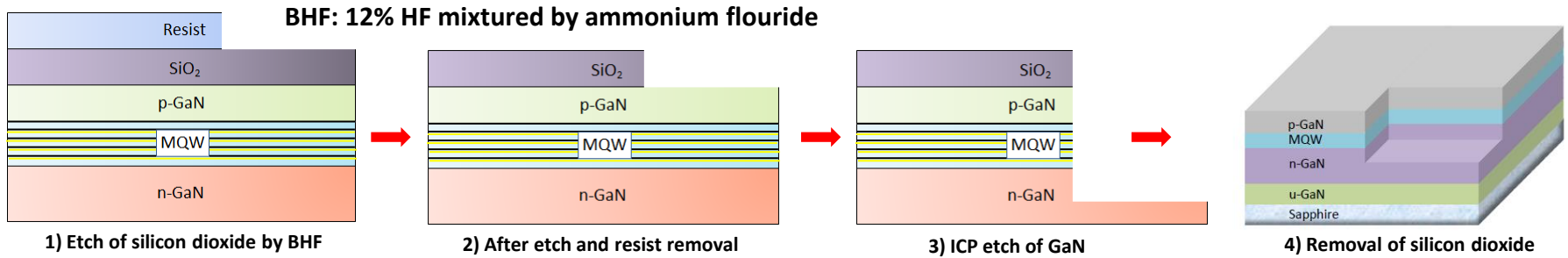
Cone structures after GaN ICP etch and removal of silicon dioxide



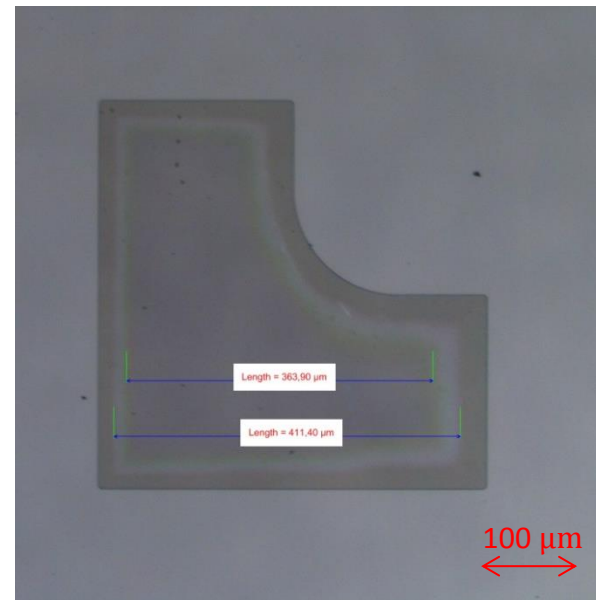
Some structures already existing in the areas to be etched before GaN ICP etch



**Solution: Formation of silicon dioxide mesa by wet etch process (BHF)**



After silicon dioxide mesa formation by BHF (with resist), the surface is quite clean also in the inspection by SEM

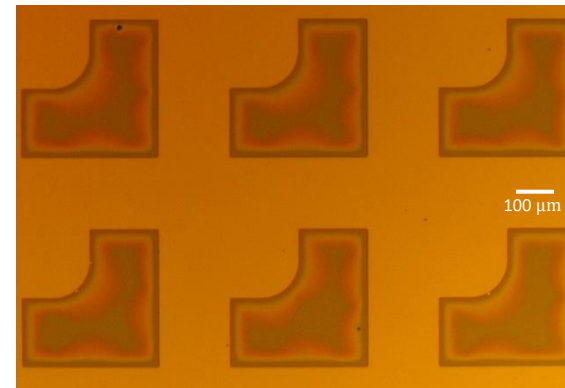


Be careful of the resist structure quality (overdeveloped resist mesa)

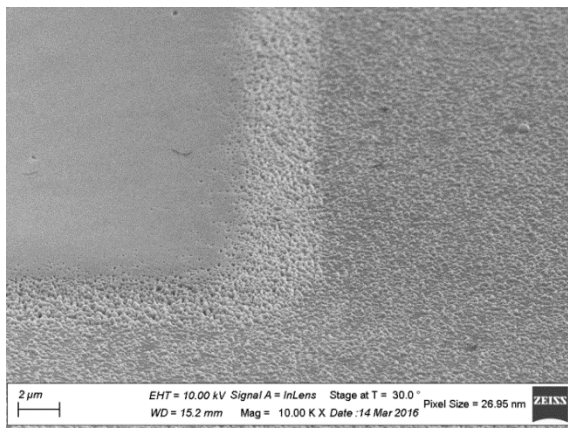
**Solution: Formation of silicon dioxide mesa by wet etch process (5%HF):  
Much more aggressive than BHF**



When etch in 5% HF hits the limit of required mesa size, there are still residual silicon dioxide in the areas to be etched by ICP

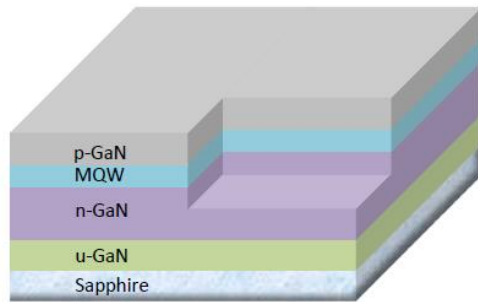


If the oxide in the areas to be etched is completely removed in 5% HF, the mesa is damaged a lot due to penetration of HF into the interface

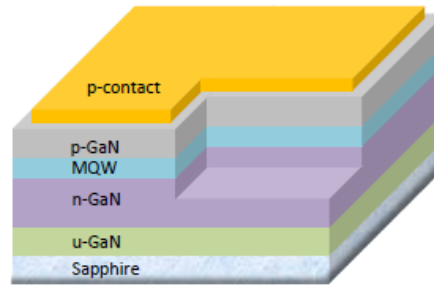


Residual silicon dioxide after incomplete etch in 5% HF introduces a lot of roughness during ICP etch of GaN

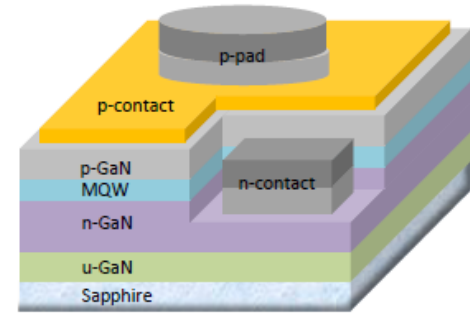
**Solution: Formation of silicon dioxide mesa by wet etch process (BHF)**



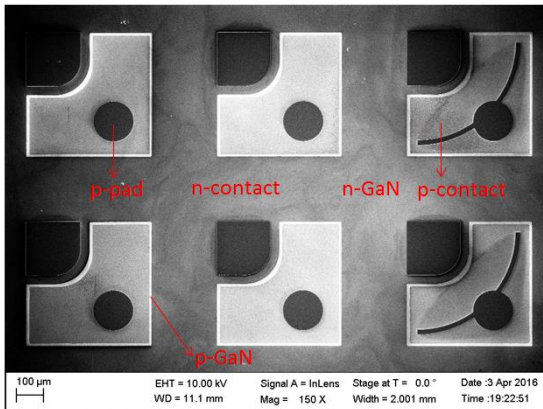
ICP etch of GaN & oxide removal



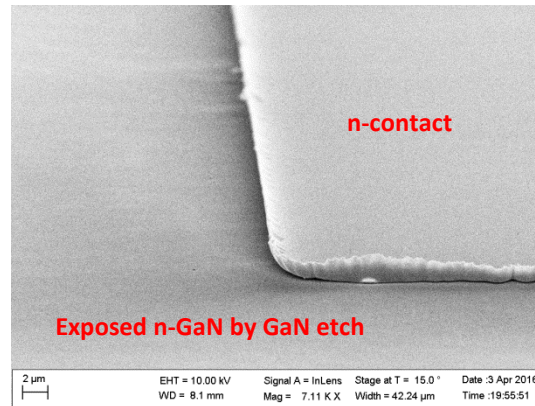
P-contact formation



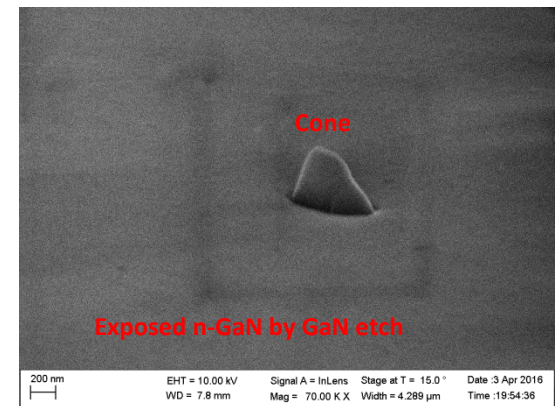
P-pad & n-contact formation



Successful formation of p-contacts, p-pads and especially the n-contacts after lift-off

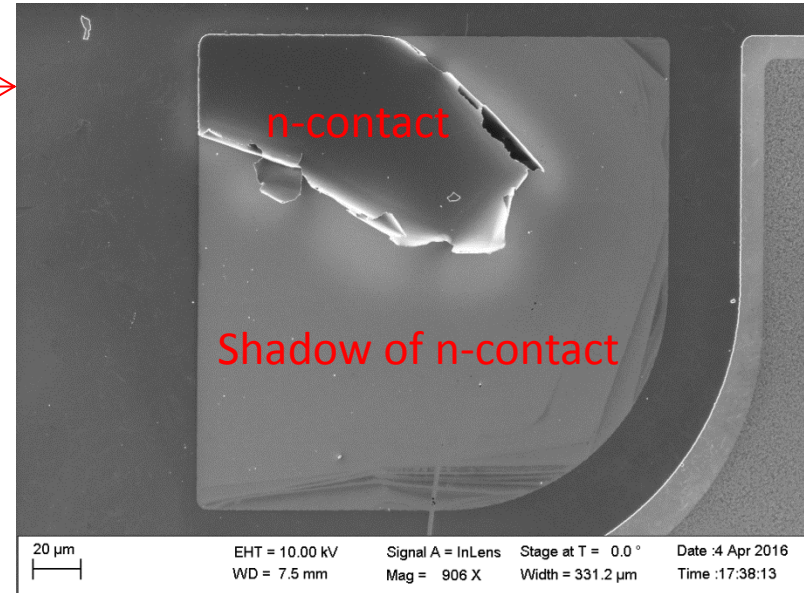
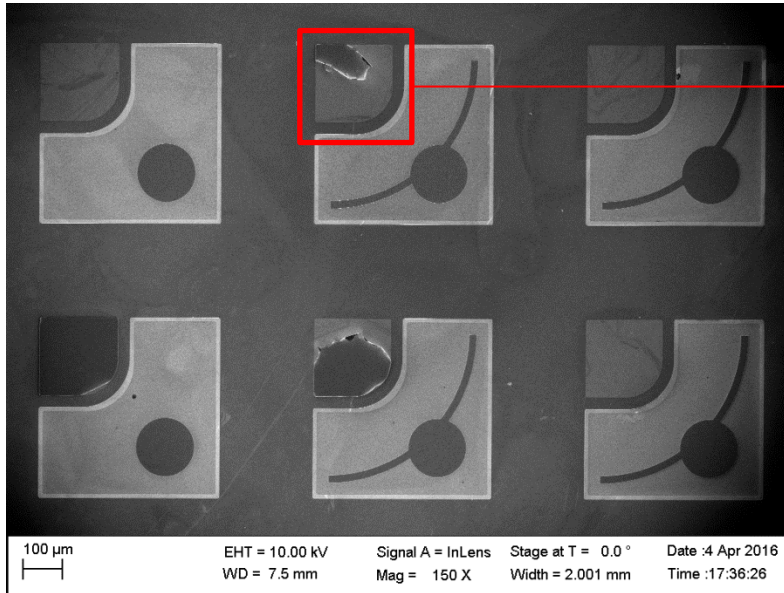


Cone structures barely observed by SEM



Still a few cones observed, might be from silicon dioxide sputtering

## Bad adhesion of n-contacts



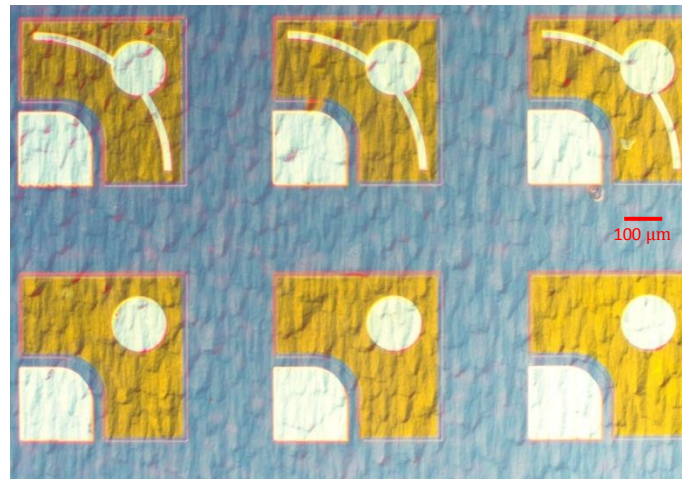
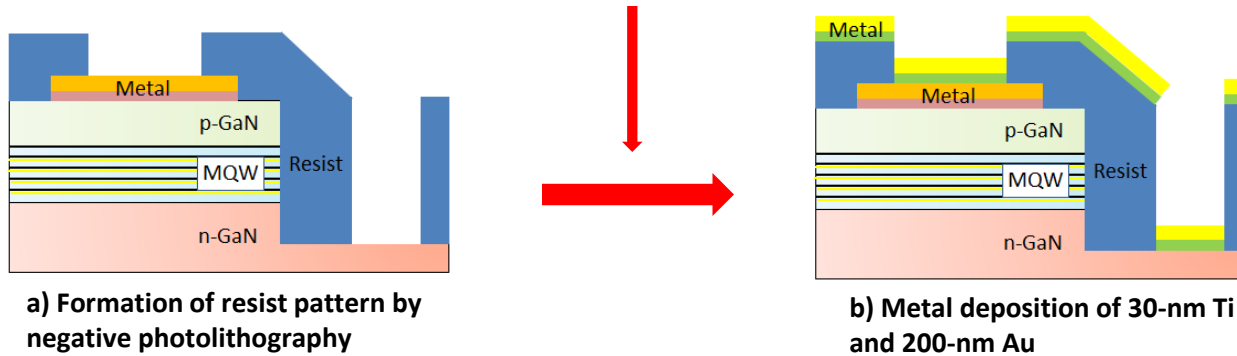
**After liftoff assisted with ultrasonic for removal of 30-nm Ti & 200-nm Au:**

- Most of n-contacts flake off (about 10 survived in 96 LEDs)
- The shapes of surviving n-contacts are not complete

**Reasons behind the bad adhesion:**

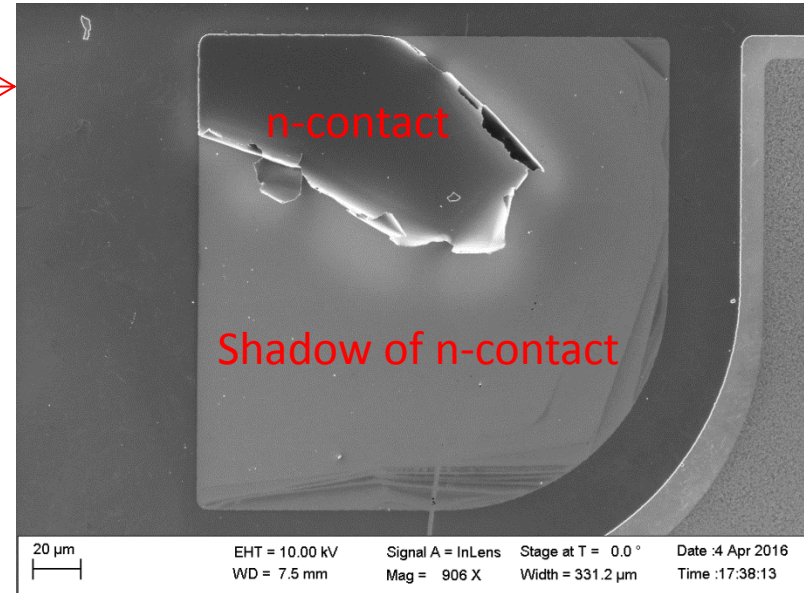
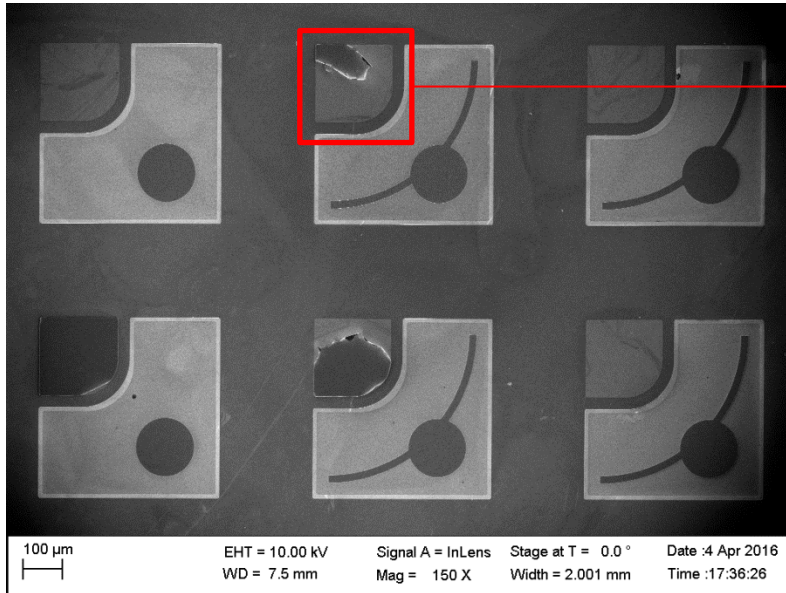
- Surface roughness (especially the etched surface of n-GaN)
- Residual resist after development in the areas for n-contact metal deposition
- The GaN surface is not clean enough

*Solution: A 1-min O<sub>2</sub> plasma process with 40 W before deposition of p-pads and n-contacts*



**Successful formation of n-contacts after liftoff**

## Bad adhesion of n-contacts



**After liftoff assisted with ultrasonic for removal of 30-nm Ti & 200-nm Au:**

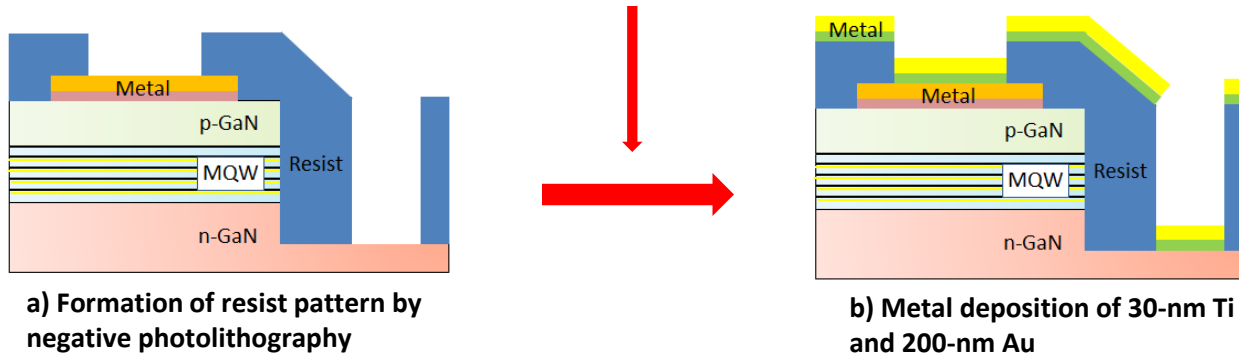
- Most of n-contacts flake off (about 10 survived in 96 LEDs)
- The shapes of surviving n-contacts are not complete

**Reasons behind the bad adhesion:**

- Surface roughness (especially the etched surface of n-GaN)
- Residual resist after development in the areas for n-contact metal deposition
- *The GaN surface is not clean enough*



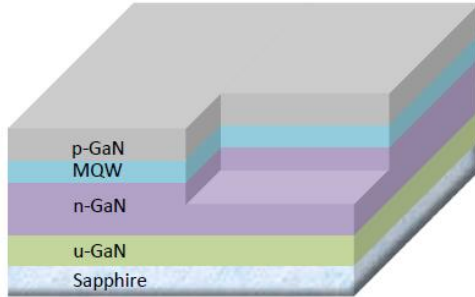
*Solution: Add 30 sec 5% HF to clean the GaN surface for metal deposition*



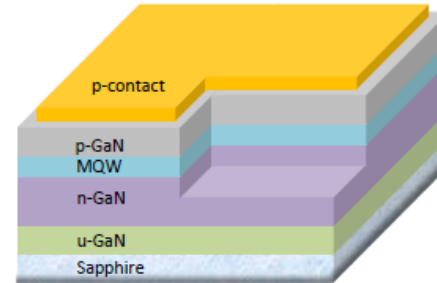
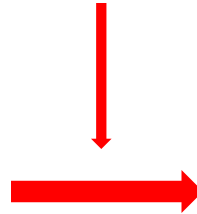
**Successful formation of n-contacts after liftoff**

**Solution: Add Piranha treatment after GaN mesa formation**

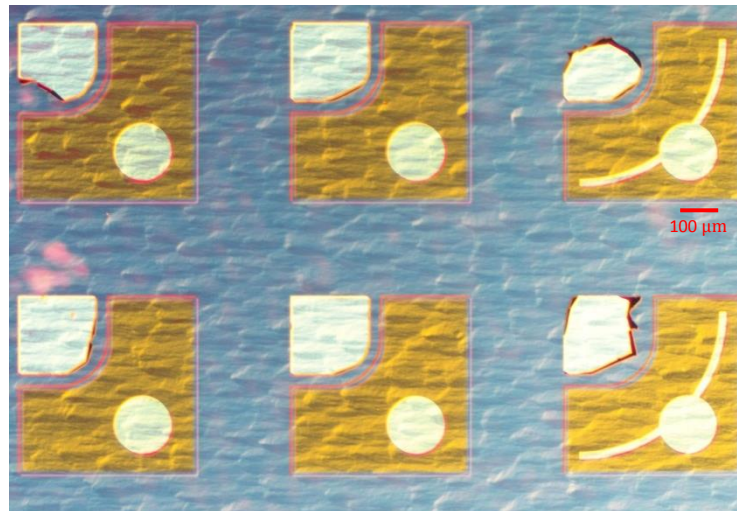
Piranha: 97% H<sub>2</sub>SO<sub>4</sub>: H<sub>2</sub>O<sub>2</sub>=4:1



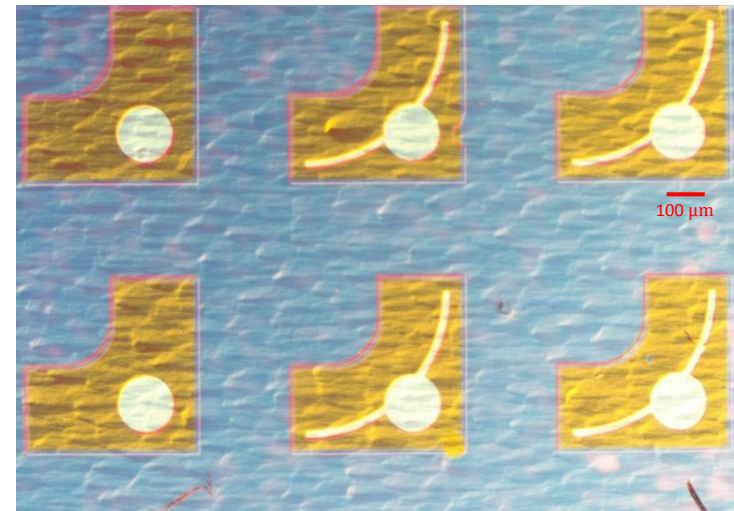
ICP etch of GaN & oxide removal



P-contact formation



Bad quality of n-contacts after liftoff



Missing of n-contacts after liftoff

***Most of n-contacts flaked off (about 8 survived in 96 LEDs)***

## 4. Conclusion:

- *The fabrication process is applicable and some improvements, especially the adhesion of metal contacts, are necessary*
- *Using BHF to form silicon dioxide mesa instead of the RIE can reduce the cone structures in the etched areas, which is good for metal adhesion improvement*
- *Applying a 1-min & 40 W oxygen plasma process or 30-sec 5% HF before n-contact deposition can improve its metal adhesion*
- *Advantage of Piranha treatment after GaN mesa formation is not very obvious*

## 5. Future plans:

*The performance including I-V characteristics, electro-luminescence of these LEDs with good metal contact quality needs to be measured and analyzed*

*In the future, methods providing both good metal adhesion and device performance can be combined to fabricate LED devices*

*After improvement of the process, the modified methods can be applied to the Near-UV LED fabrication*

# Thank you!

April 6, 2016

Yi Wei

**Multi-configuration on <6H> bulk SiC by Ag NPs growing and Al<sub>2</sub>O<sub>3</sub> coating on substrate.****Objective**

Since the photoluminescence (PL) effect of bulk <6H> SiC is contributed by the electron-hole pairs radiative recombination only within several hundreds of nanometers depth range under the surface of the substrate, therefore it is crucial to break through this limitation to extract more electron-hole pairs from deeper part of the bulk material in order to achieve higher internal quantum efficiency. By applying localized surface plasmon (LSP) induced by Ag nanoparticles (NPs) growing on the top of bulk SiC, the enhanced near field around the Ag NPs can be expected to extract more electron-hole pairs from bulk SiC. In addition, an extra Al<sub>2</sub>O<sub>3</sub> thin layer coating on the top of the bulk SiC after Ag NPs growing is expected to tune the  $\lambda_{peak}$  of PL, and it can prevent Ag NPs from being oxidized. Moreover, Ag NPs being sandwiched in two high refractive index materials is believed to scatter more internal emitted light out of the substrate.

In order to fulfill the objective, two experimental investigations have been made:

- How different temperatures of rapid thermal annealing during the formation of Ag NPs will influence the LSP effect & How different Al<sub>2</sub>O<sub>3</sub> deposition method, in which atomic layer deposition (ALD) and radio frequency (RF) sputter methods are applied, will affect LSP.
- The influence of the size of Ag NPs on LSP.

**Conclusions from two experimental investigations**

- For Al<sub>2</sub>O<sub>3</sub> deposition  $\sim 20nm$  range, there is no apparent difference by applying atomic layer deposition or radio frequency sputtering.
- Rapid thermal annealing is preferable at 350°C.
- For samples with Ag NPs which deposited by Ag thin film of 5nm & 10nm,  $\lambda_{peak}$  of PL has  $\sim 10nm$  redshift.
- Photon lifetime can be decreased by adding Ag NPs on bulk SiC, in which can be further decreased by coating extra Al<sub>2</sub>O<sub>3</sub> layer.

- Localized surface plasmon effect can be seen, in which the  $\lambda$  range of LSP depends on the size of Ag NPs.
- LSP couplings cause more absorption than scattering, this is the reason why LSP has decreased PL.

In addition, exponential fitting analysis has been applied to extract the photon lifetime from time-resolved PL measurement, and corresponding confidence range calculation and error analysis have been achieved to ensure the reliability of the fitting analysis. Triple-exponential decay model has been applied on bare SiC and SiC substrate only with Ag NPs, tetra-exponential decay model is found to be applicable on SiC with Ag NPs plus  $\text{Al}_2\text{O}_3$  coating and also the SiC substrate only coated with  $\text{Al}_2\text{O}_3$ . However, the physical explanation behind these two exponential decay model are unknown yet, more literature study is needed in order to give credible interpretation.

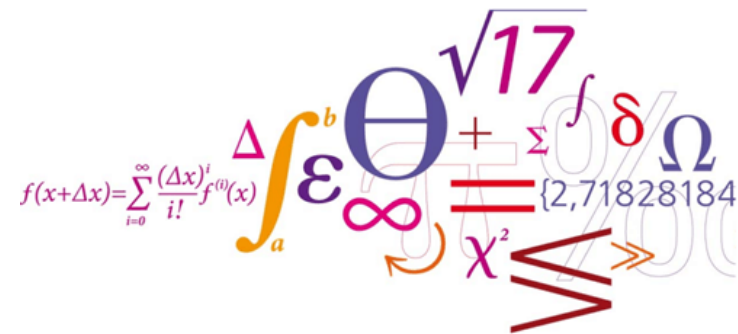
#### **To-do list**

- Application of XPS analysis to investigate whether Ag NPs have been oxidized or not during ALD process.
- Literature study of time-resolved PL curve exponential decay modelling.
- Simulation of LSP of Ag NPs, find out in which size range of Ag NPs, LSP could cause more scattering than absorption of energy.

# *Fabrication and passivation of porous SiC*

Weifang Lu

2016.04.06





# Outline

1

Introduction

2

Fabrication of porous 6H-SiC by **anodic oxidation method**

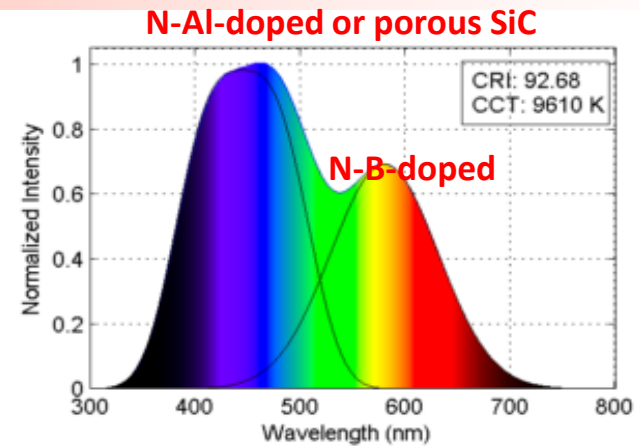
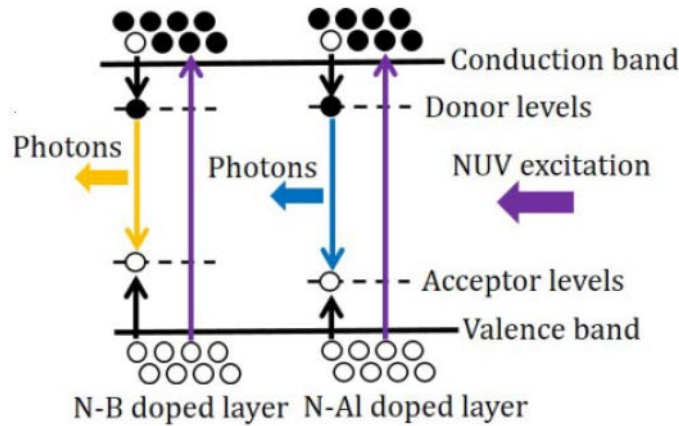
3

Surface passivation of porous 6H-SiC by ALD  $\text{Al}_2\text{O}_3$  and  $\text{TiO}_2$  films

4

Conclusion

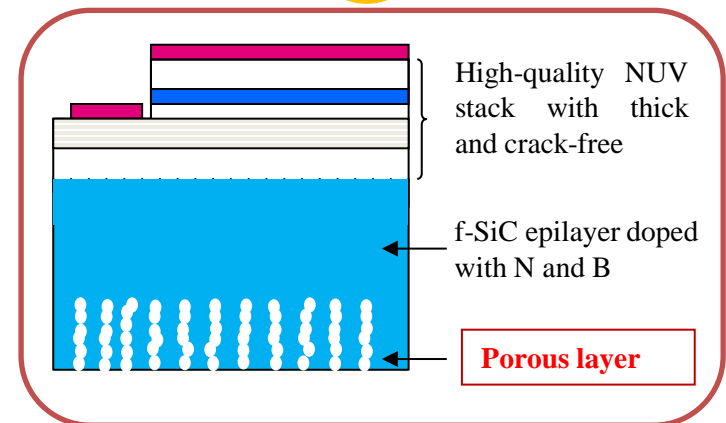
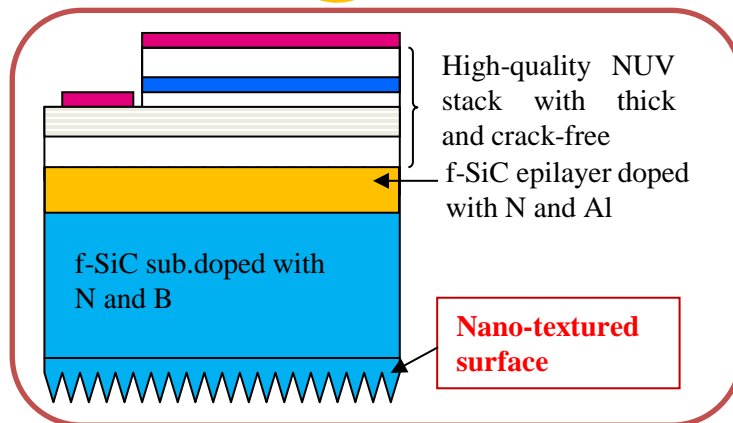
# 1. Introduction



1

## Fluorescent 6H-SiC based white LEDs

2

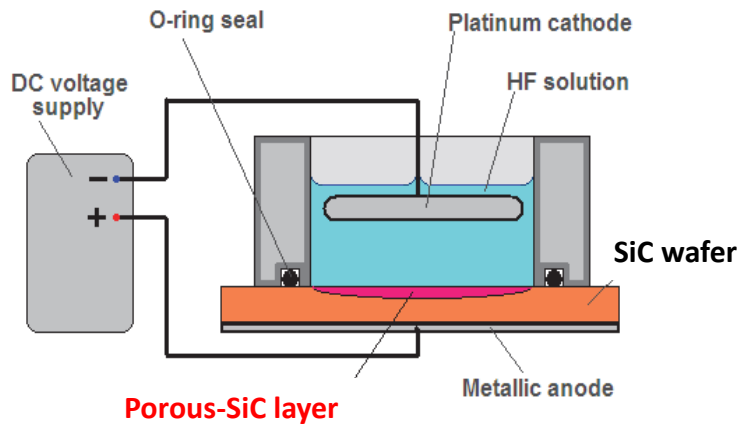
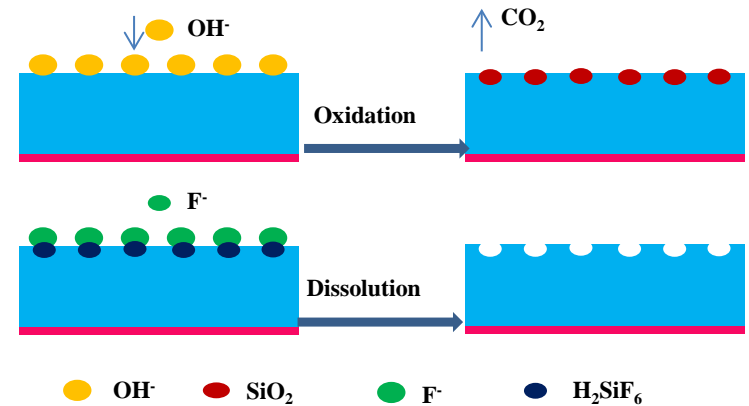
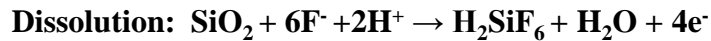
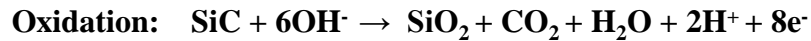


**Passivation** of the large surface area to decrease surface recombination.

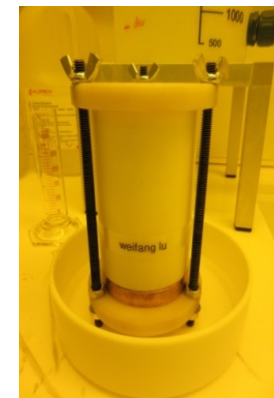
- Ref: 1. S. Kamiyama, et al, Journal of Semiconductors 32 (2011) 013004  
 2. H. Ou, et al, European Physical Journal B: Condensed Matter Physics 87 (2014) 58  
 3. Y. Ou, et al, Optics express 19 (2011) A166  
 4. Y.C. Lee et al. Optical Materials 35 (2013) 1236–1242

# 2.1 Fabrication of porous SiC by anodic oxidation method

The SiC anodic etching in HF solution could be described with two steps, oxidation of SiC and dissolution of the formed SiO<sub>2</sub>:



Schematic diagram of the experimental setup for anodic oxidation



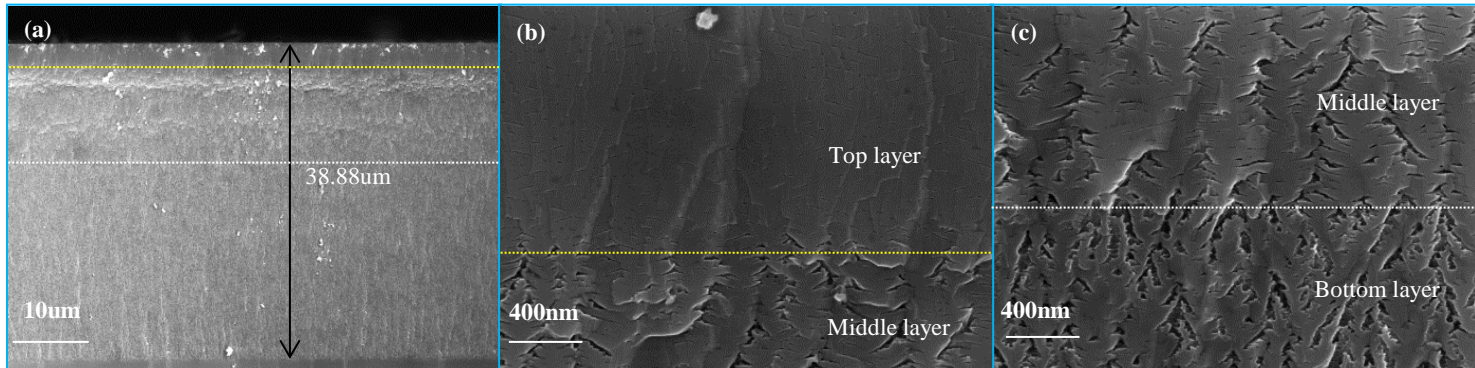
## ① SiCrystal: 6H-SiC

Thickness: 250μm, N-doped, on-axis, Si-Face polished,  
MPD<100/cm<sup>2</sup>

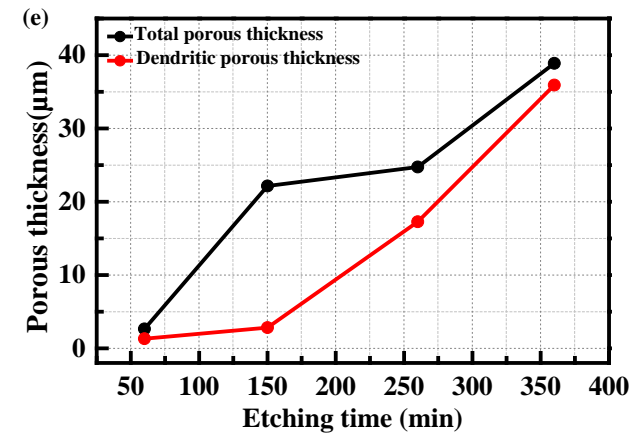
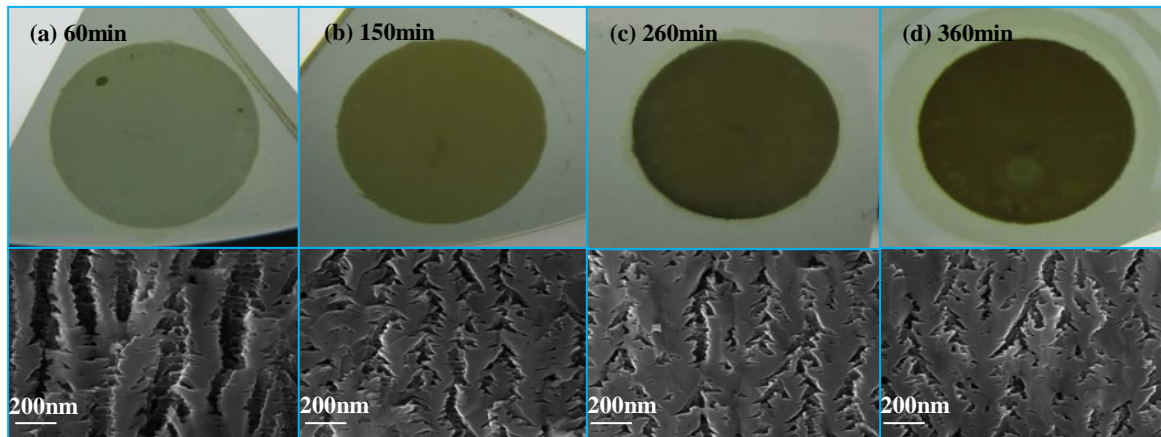
## ② Tankeblue: 6H-SiC

Thickness: 430μm, B-N co-doped  
Double sides polished with Si face CMP

## 2.2 Porous 6H-SiC samples ①---SiCrystal

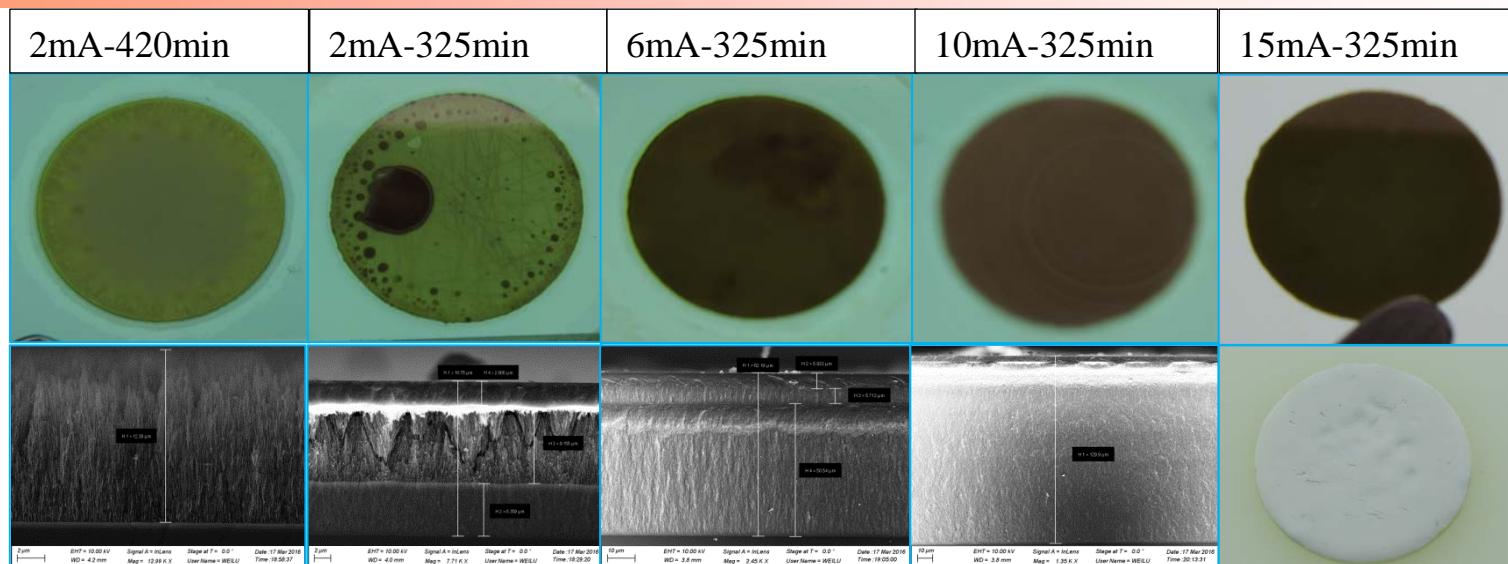


Cross-sectional SEM images of porous sample (360min): (a) the overview of the porous layer consisting of three layers with a total thickness of 38.88 μm, (b) the boundary between top layer and middle layer, and (c) the boundary between the middle and bottom layer.

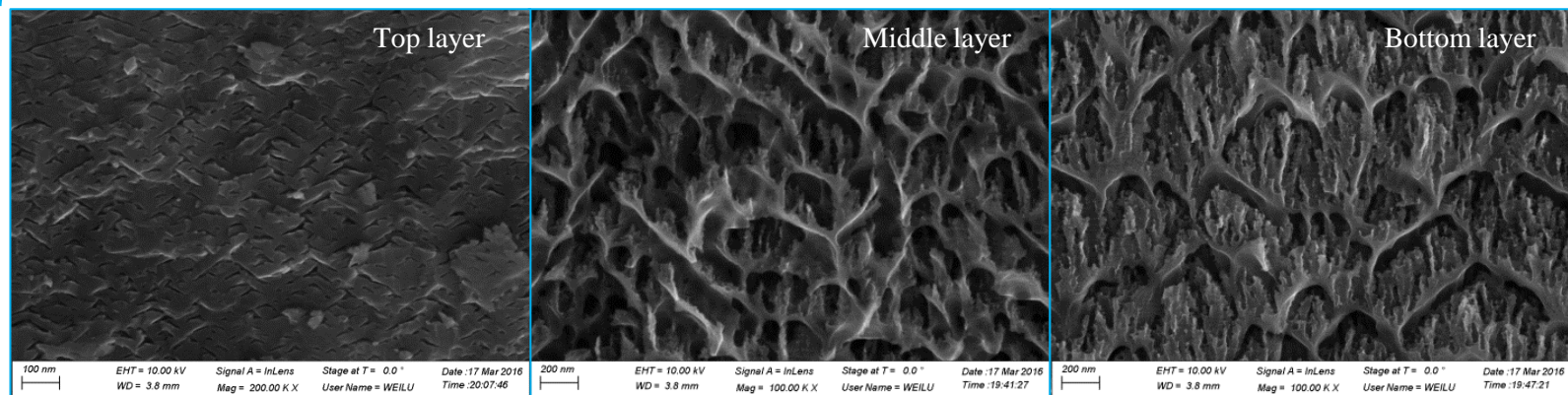


Photographs of porous SiC with different etching time (a) 60 min, (b) 150 min, (c) 260 min and (d) 360 min. (e) The corresponding relationship of porous and dendritic thickness with etching time for 6H-SiC.

## 2.3 Porous 6H-SiC samples ②---Tankeblue

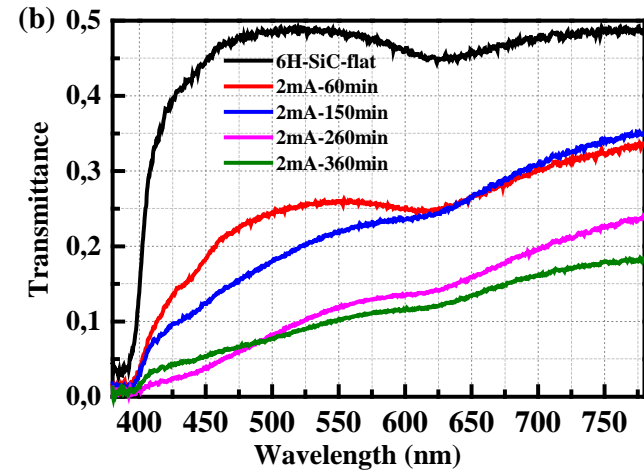
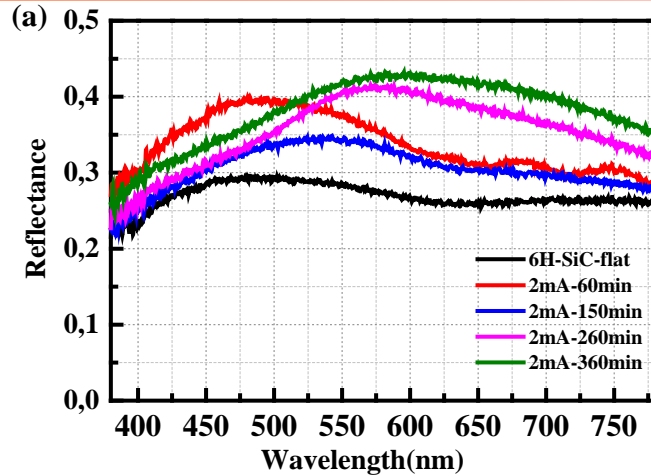


Photographs and SEM images of porous SiC with different etching current: 2mA, 6mA, 10mA and 15mA.

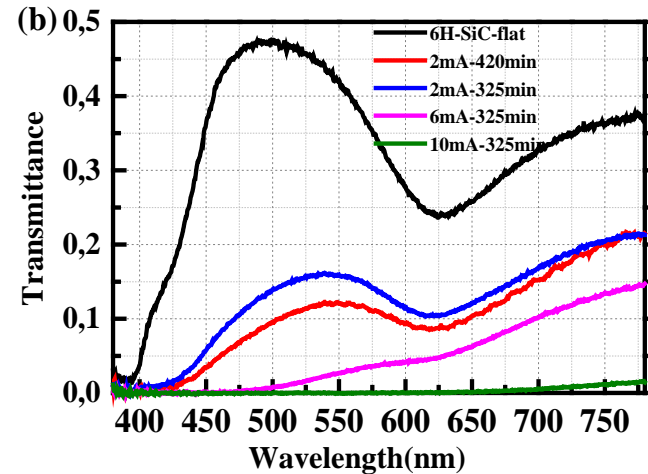
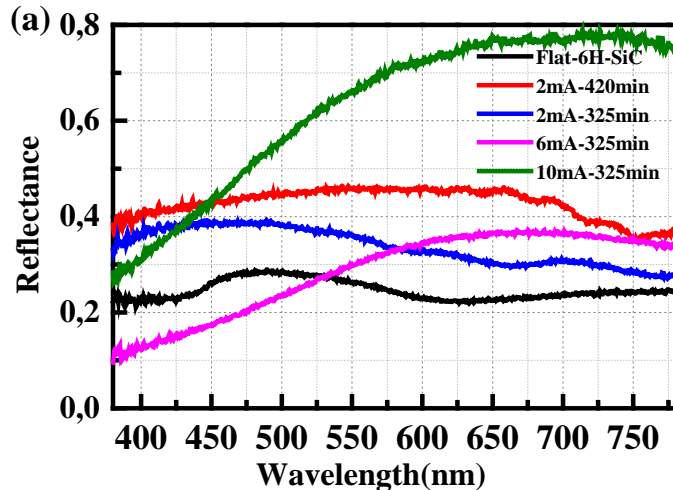


Cross-sectional SEM images of porous sample prepared under 10 mA.

## 2.4 Optical properties—Reflectance and Transmittance



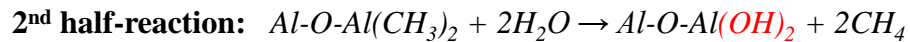
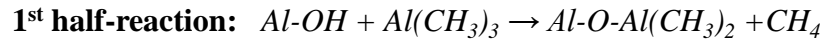
**SiCrystal samples:** (a) reflectance and (b) transmittance spectra for the flat 6H-SiC sample and the porous samples.



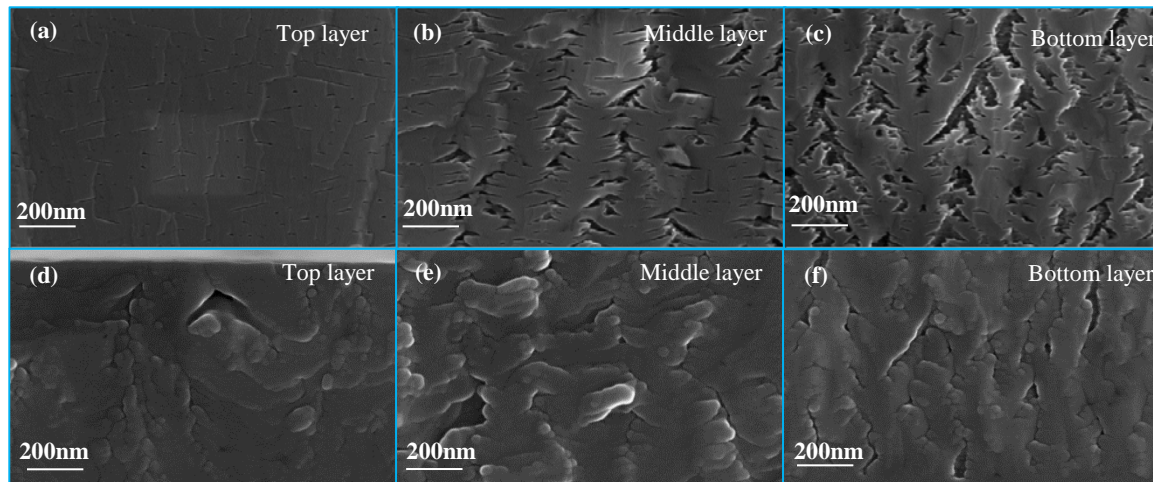
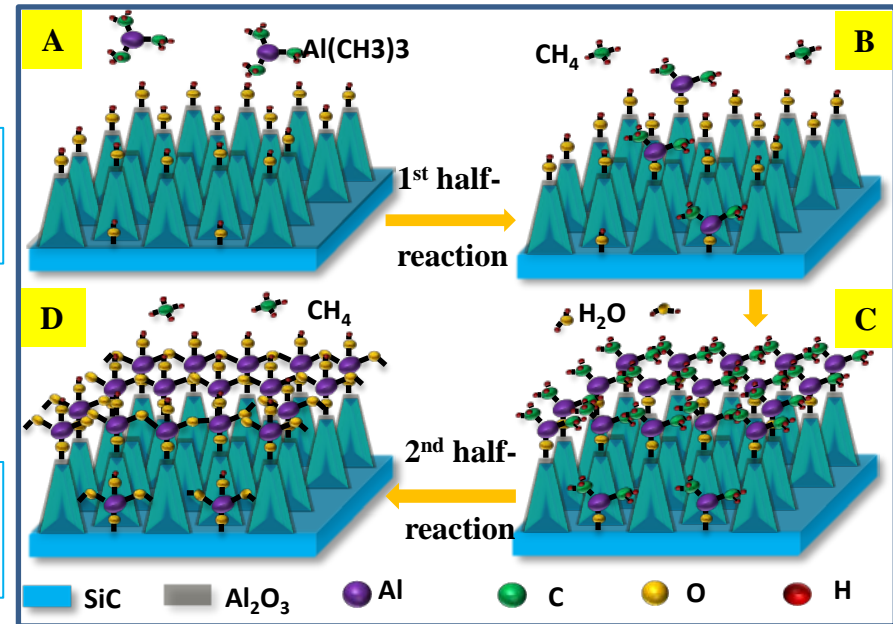
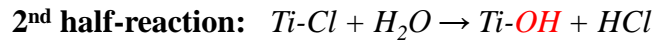
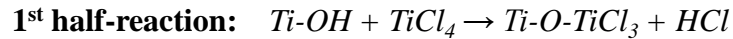
**Tankeblue samples:** (a) reflectance and (b) transmittance spectra for the flat 6H-SiC sample and the porous samples.

# 3.1 Passivation: atomic layer deposited $\text{Al}_2\text{O}_3$ and $\text{TiO}_2$ film

The surface chemistry reaction during  $\text{Al}_2\text{O}_3$  ALD (atomic layer deposition) deposition:



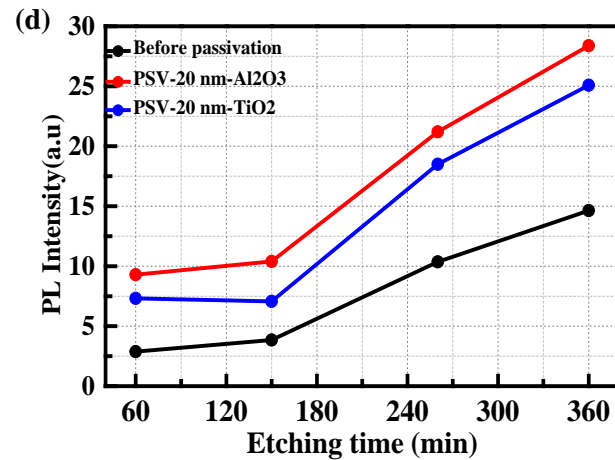
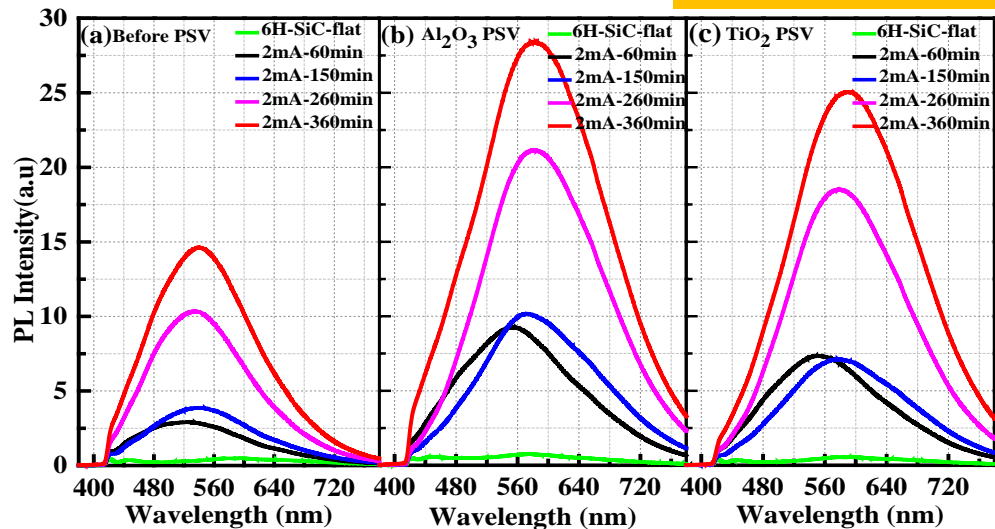
As the similar deposition process for  $\text{TiO}_2$  thin films, the reaction is also divided into the following two half-reactions:



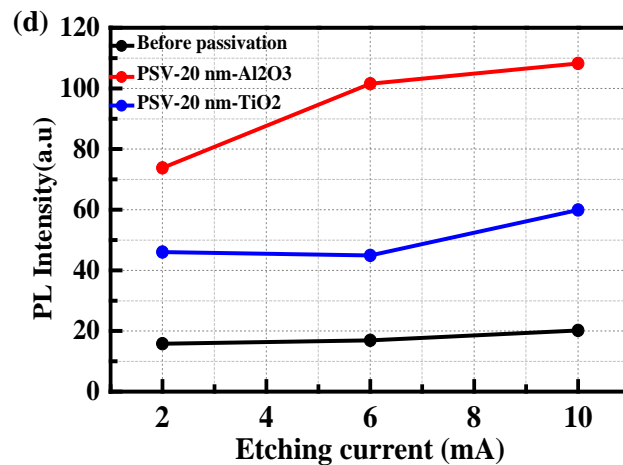
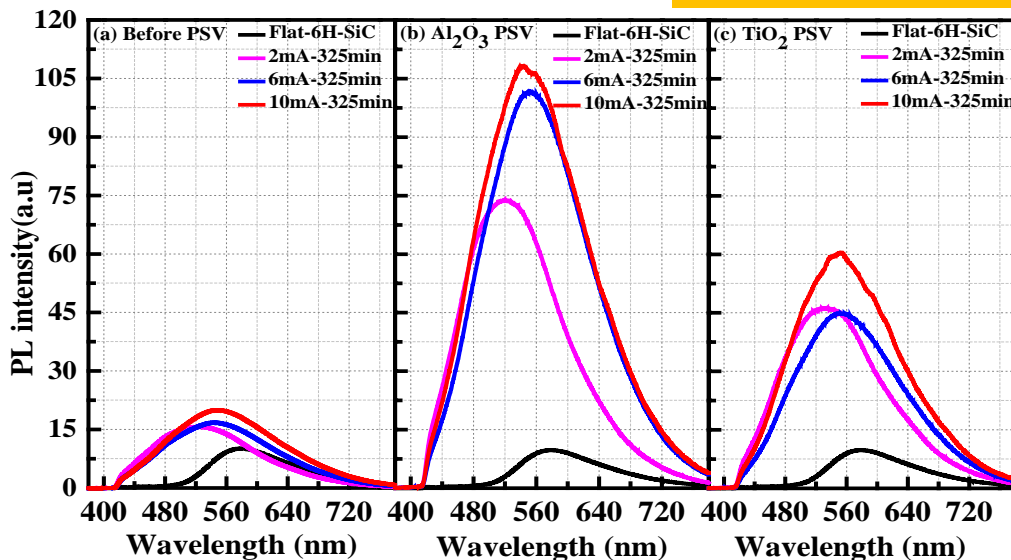
Cross-sectional SEM images of porous sample (360 min): (a) top layer, (b) middle layer and (c) bottom layer covered with 20 nm thick  $\text{TiO}_2$ .

## 3.2 Passivation effect of Al<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub>

SiCrystal samples

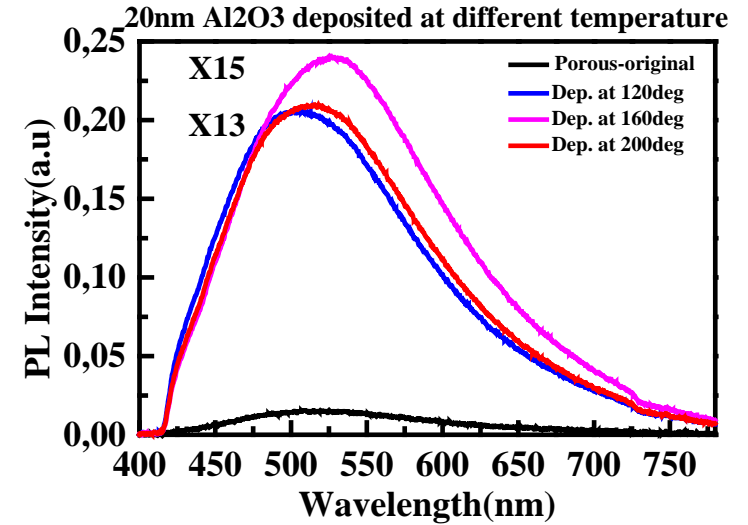
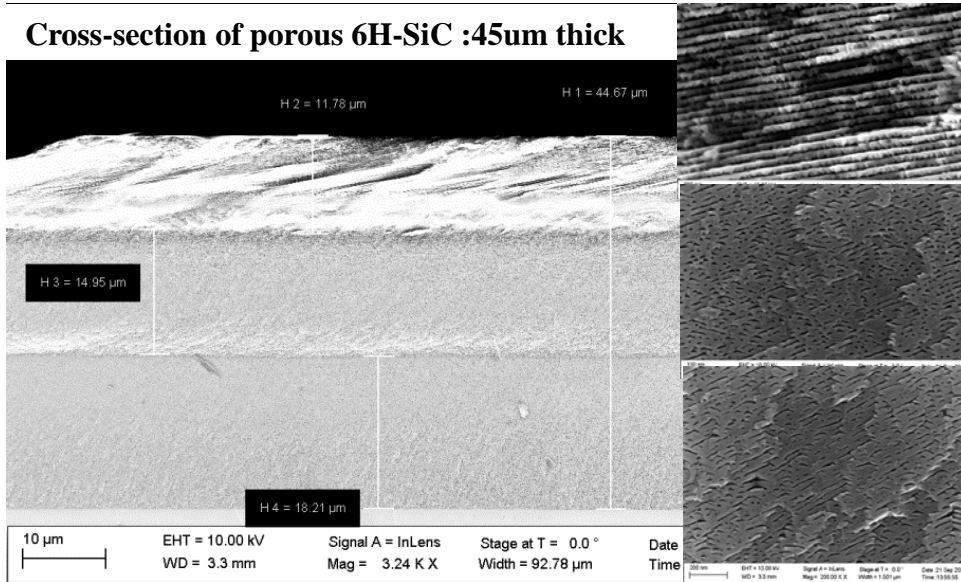


Tankeblue samples

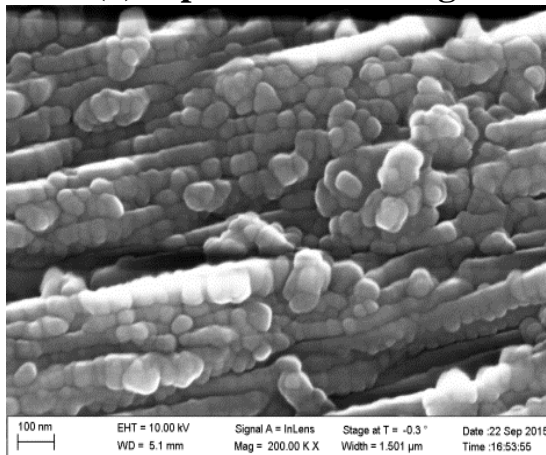




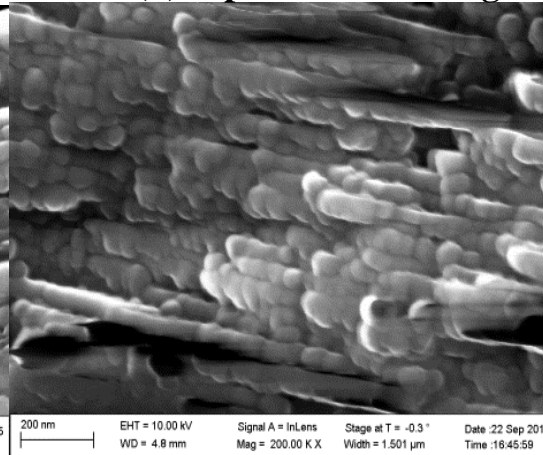
# 3.3 Passivation optimization: deposition temperature



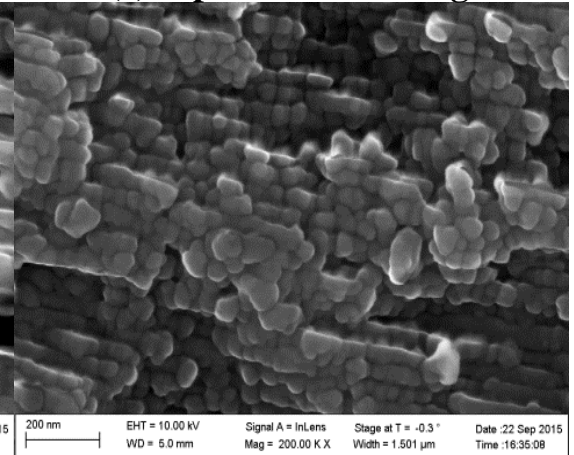
**(a) Deposited at 120deg**



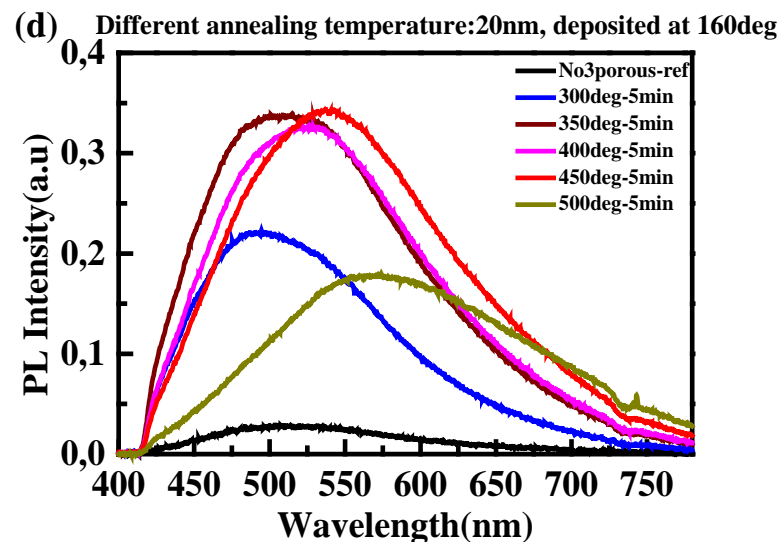
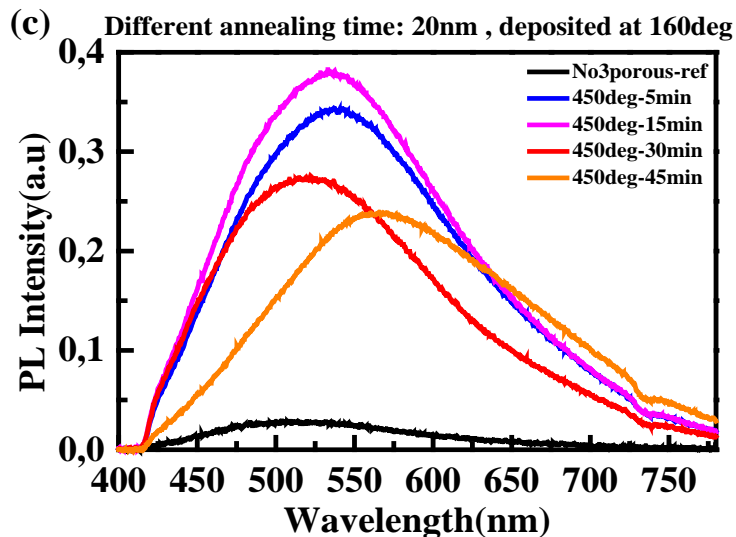
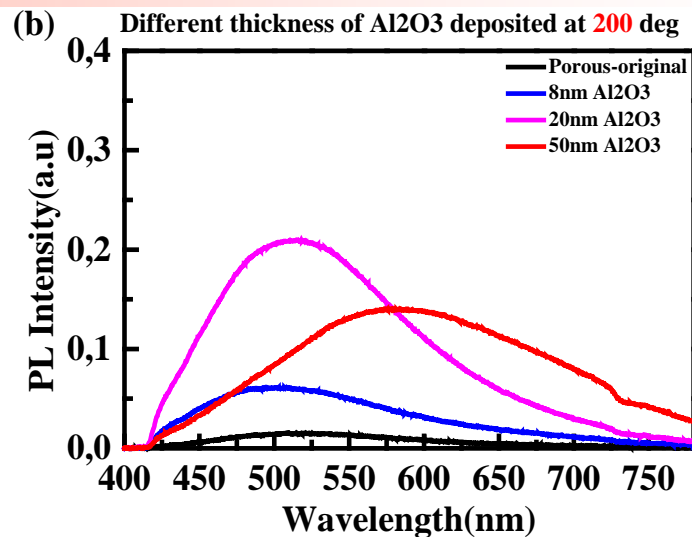
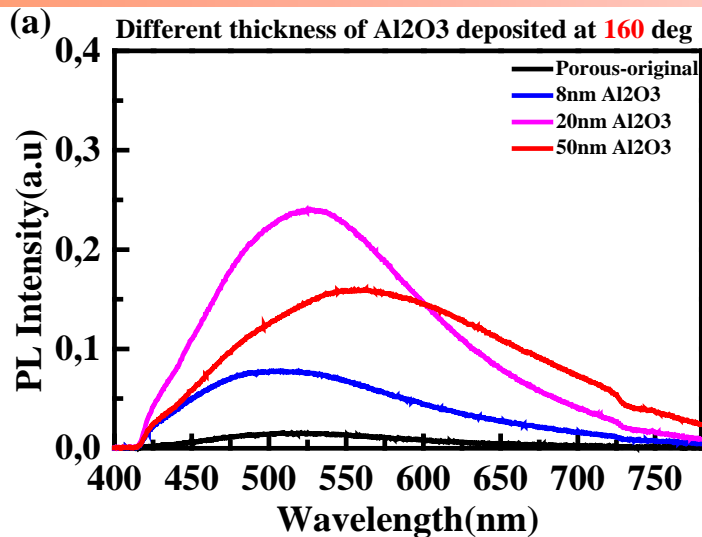
**(b) Deposited at 160deg**



**(c) Deposited at 200deg**



## 3.4 Passivation optimization: thickness, annealing temperature



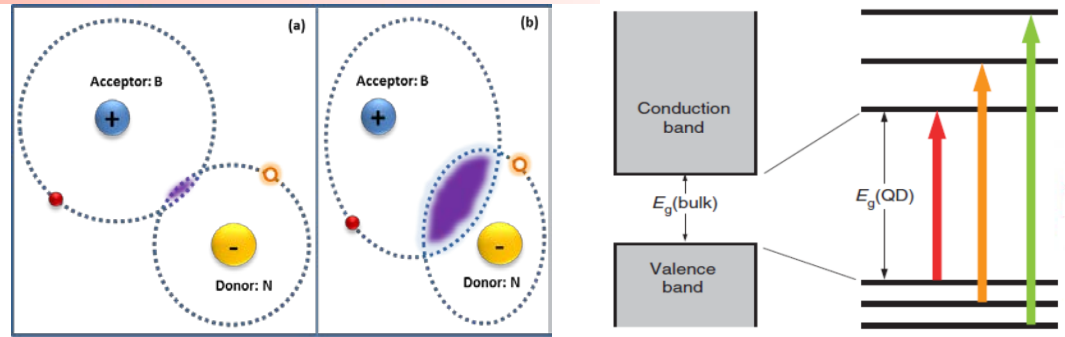
PL spectra of porous 6H-SiC sample: with different Al<sub>2</sub>O<sub>3</sub> thickness and annealing condition. The optimized parameters are 20nm Al<sub>2</sub>O<sub>3</sub>, deposited at 160deg, annealing for 15min, annealing at 350deg.

## 3.5 Discussion

Luminescence mechanisms in porous SiC

Quantum size effect

Surface states



Schematic diagrams showing quantum size effect: blue shift in PL spectra.

- ❖ **Quantum size effect:** The calculated exciton Bohr radius of 6H-SiC and 4H-SiC is **0.7 and 1.2nm**, respectively. The exciton Bohr radius of SiC can be calculated according to the following function :

$$R = \frac{4e\epsilon_0\epsilon_r\hbar^2}{\mu_0 e^2} = \frac{0.053\epsilon_r}{\mu_0 / m_0} \quad \mu_0 = m_e m_h / (m_e + m_h)$$

- ❖ **Surface states:** The PL of porous SiC is very sensitive to the surface, which leads to carrier trapping and/or surface recombination. Passivation of surface states could decrease the non-radiative recombination rates.

- ❖ **Al<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> passivation:** **chemical passivation --- hydrogen saturation of the dangling bonds;**  
field-effect passivation --- high negative fixed charge density on the interface.

- ❖ **Competition between surface passivation and oxidation:** **if the surface area of porous SiC is oxidized by Al<sub>2</sub>O<sub>3</sub> during annealing process, the PL intensity maybe decrease and red shift.**

## 4. Conclusion

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1. Porous structures have been fabricated on Tankeblue and SiCrystal 6H-SiC substrates by using anodic oxidation method.
2. The effects of etching time and current on morphology and optical properties have been investigated.
3. Compared with the uncoated porous SiC, the  $\text{Al}_2\text{O}_3$  or  $\text{TiO}_2$  coated porous samples exhibit a strong enhancement of photoluminescence, which is attributed to the decrease of non-radiative recombination.
4. The passivation conditions have been optimized: 20nm  $\text{Al}_2\text{O}_3$ , deposited at 160deg, annealing for 15min, annealing at 350deg.

Thank you!

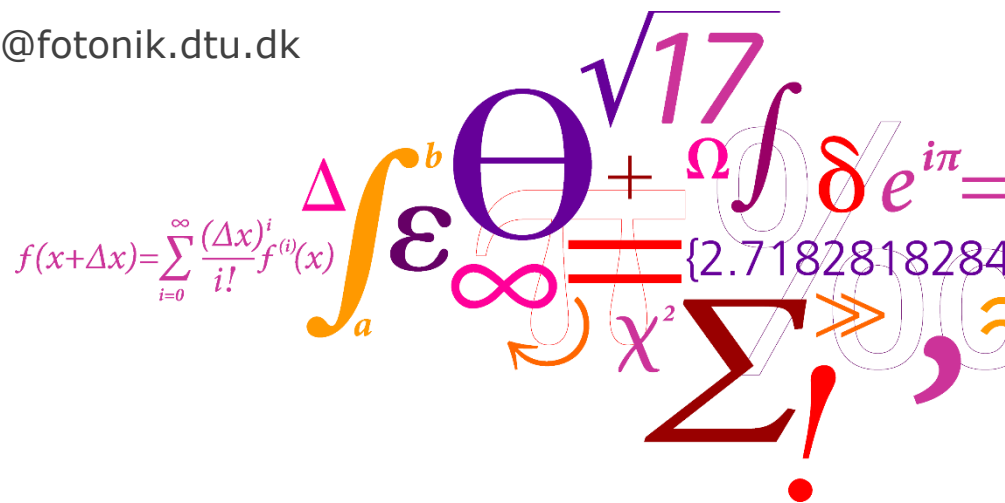


# Efficiency enhancement: fluorescent SiC with hybrid structures and nitride LED with nanopillar structure

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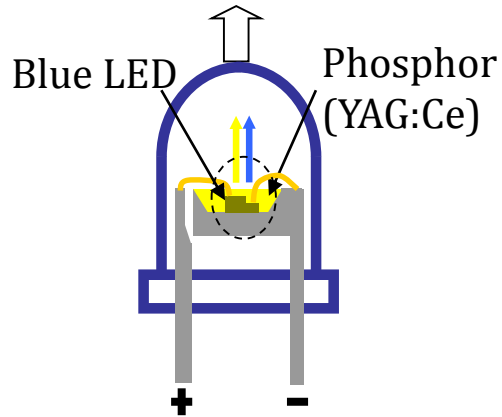


# Outline

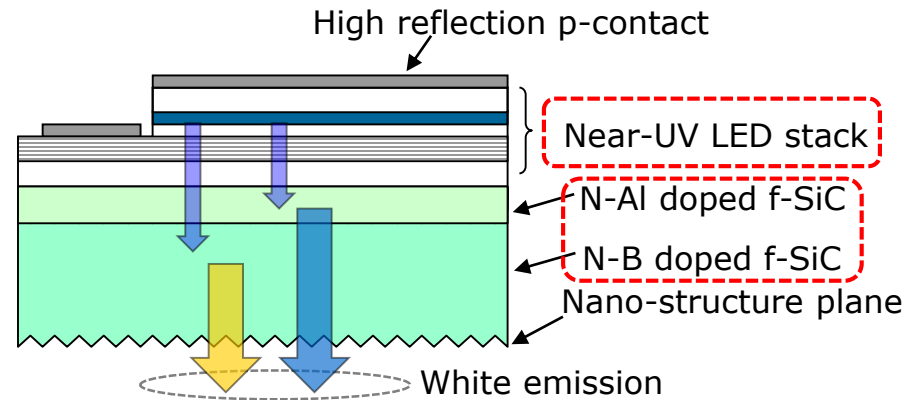
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1. Introduction
2. Fluorescent SiC with hybrid structures
3. InGaN/GaN LED with nanopillar structures
4. Summary

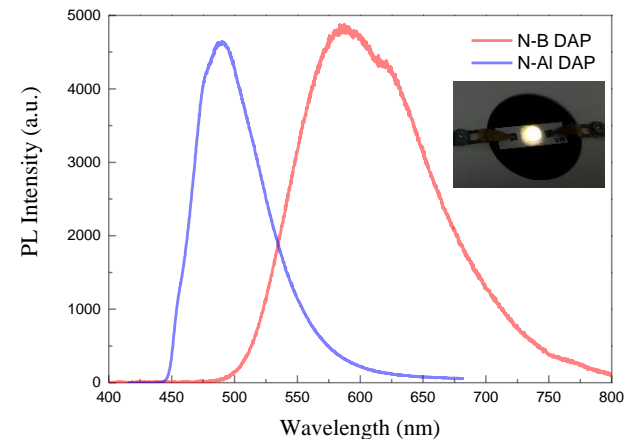
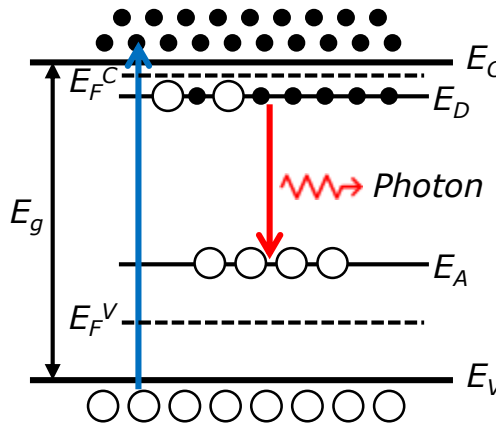
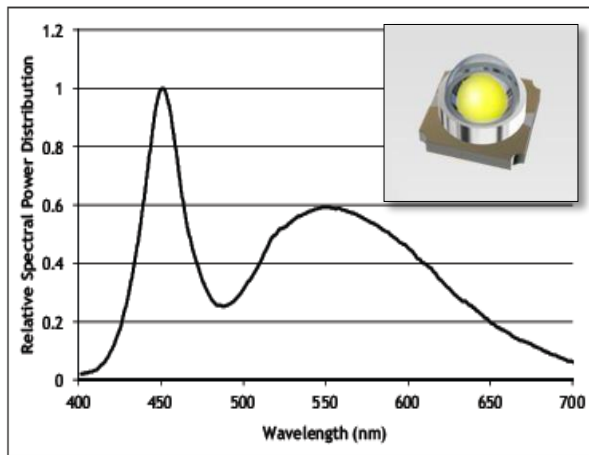
# 1. Fluorescent SiC based white LED



*Typical commercial white LED*



*F-SiC monolithic white LED*

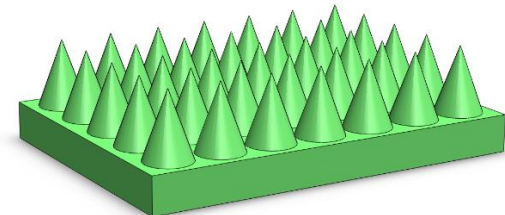
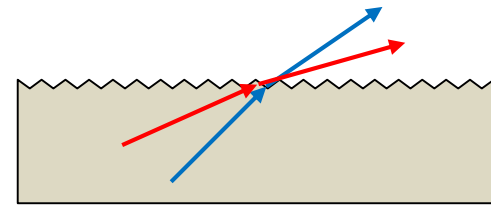
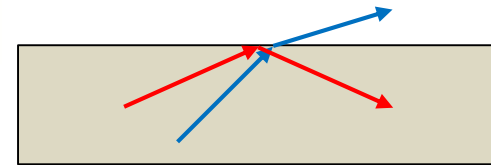
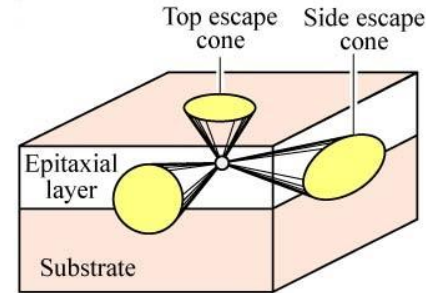




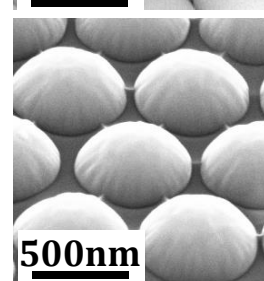
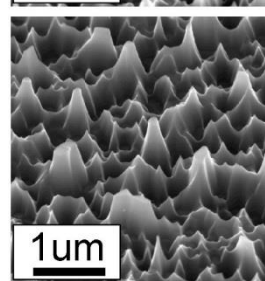
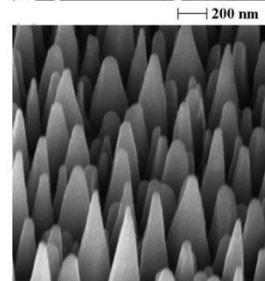
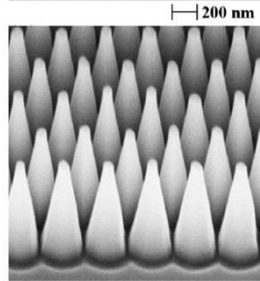
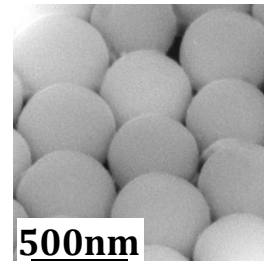
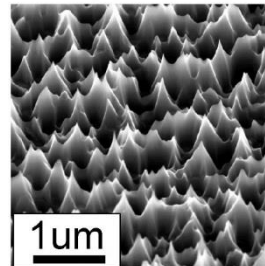
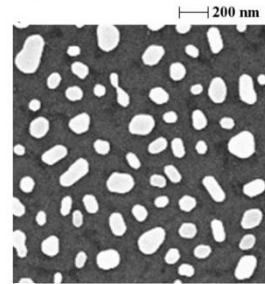
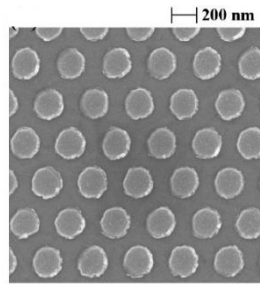
# 1. Fluorescent SiC

6H-SiC:

- Large refractive index 2.7@580nm
- Large internal reflection
- Low light extraction efficiency 3.4%
- Use nanostructure fabrication



Previous work of nanostructures on f-SiC:



E-beam lithography  
(*Opt. Express* 20, 7575, 2012)

Self-assembled nanopatterning  
(*Opt. Lett.* 37, 3816, 2012)  
(*Opt. Mat. Express* 3, 86, 2013)

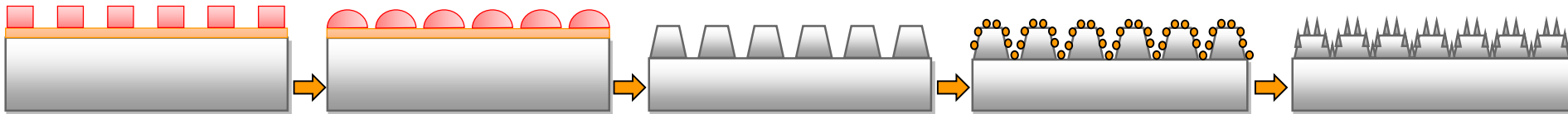
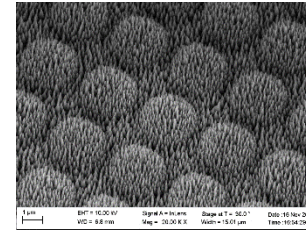
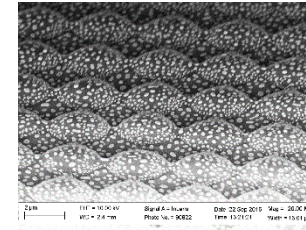
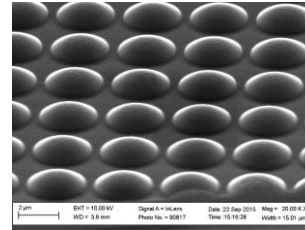
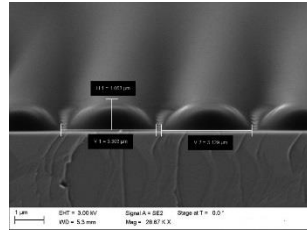
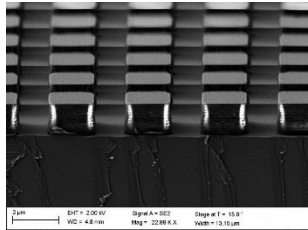
Maskless nanopatterning  
(*Opt. Mat. Express* 3, 1119, 2013)

Nanosphere lithography  
(*Sci. Rep.* 4, 4662, 2014)

## 2. F-SiC with hybrid structures

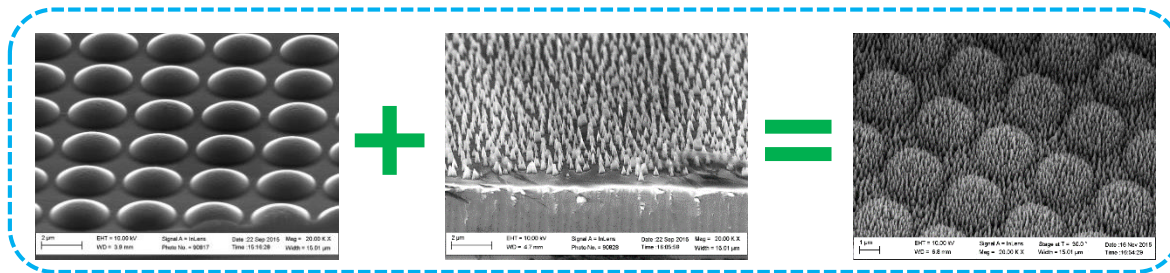
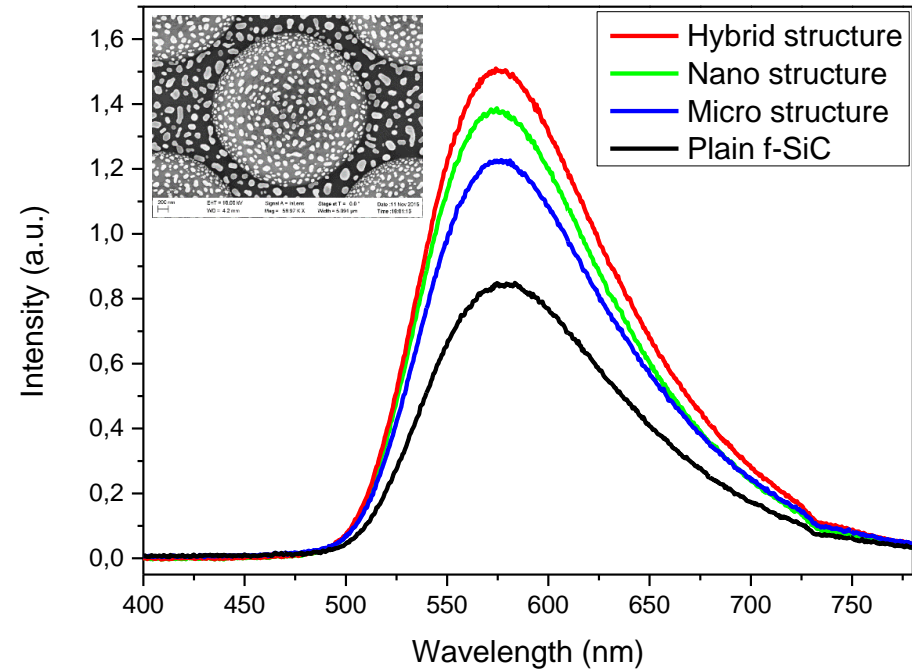
Fabrication of hybrid structures:

- A combination of micro-structure ( $\sim 3\mu\text{m}$ ) and nano-structure (100-200nm)
- Method: special photolithography with nanopatterning



## 2. F-SiC with hybrid structures

- Combined hybrid structure shows larger luminescence enhancement than pure nano- or micro-structure on f-SiC
- Feature size of  $\mu$ -structure:  $3\mu\text{m}$
- Feature size of n-structure: 100-200nm



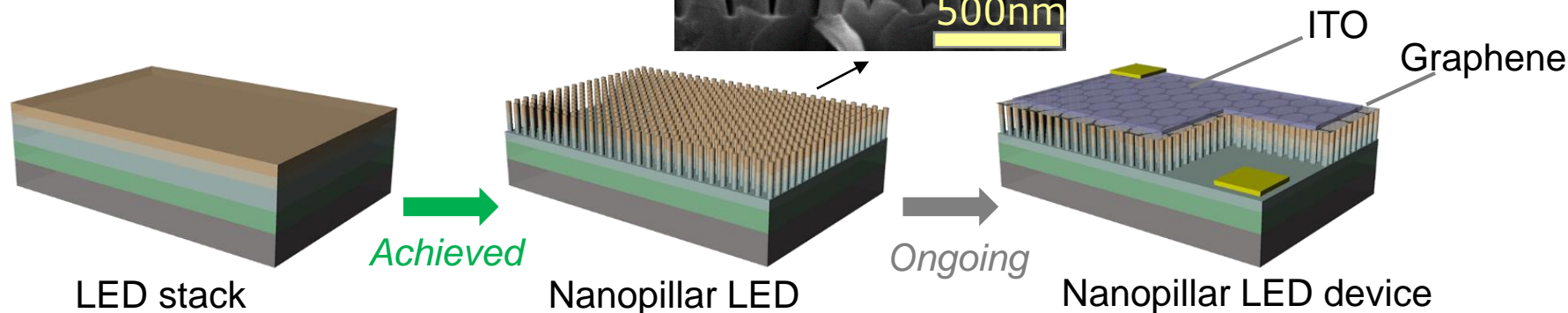
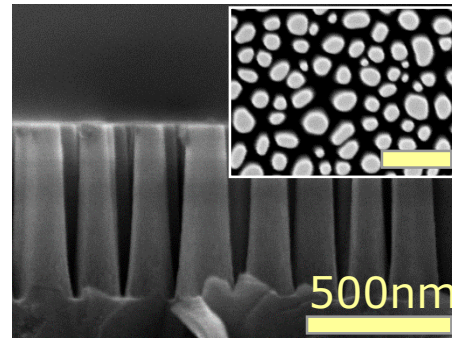
# 3. Nanopillar nitride-based LED

Problem in normal nitride-based LED:

- larger internal strain due the lattice mismatch between InGaN and GaN in QWs
- Limited IQE caused by so-called quantum confined Stark effect (QCSE)

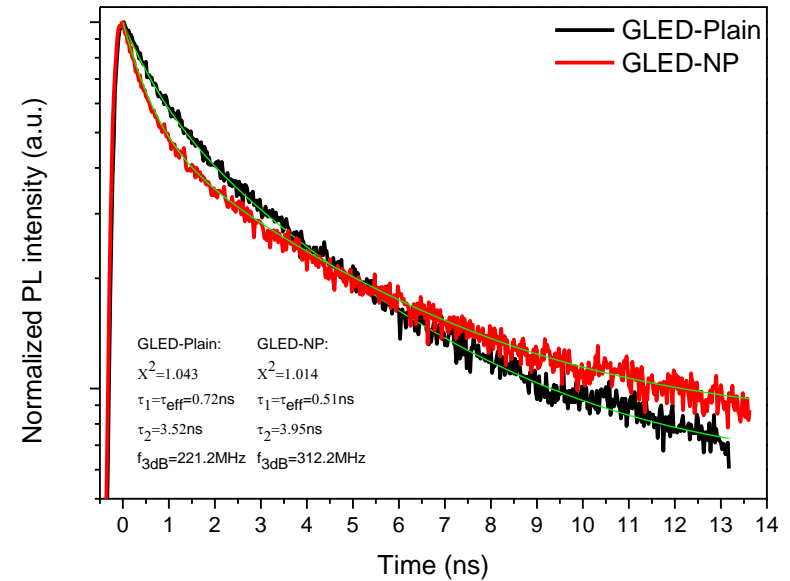
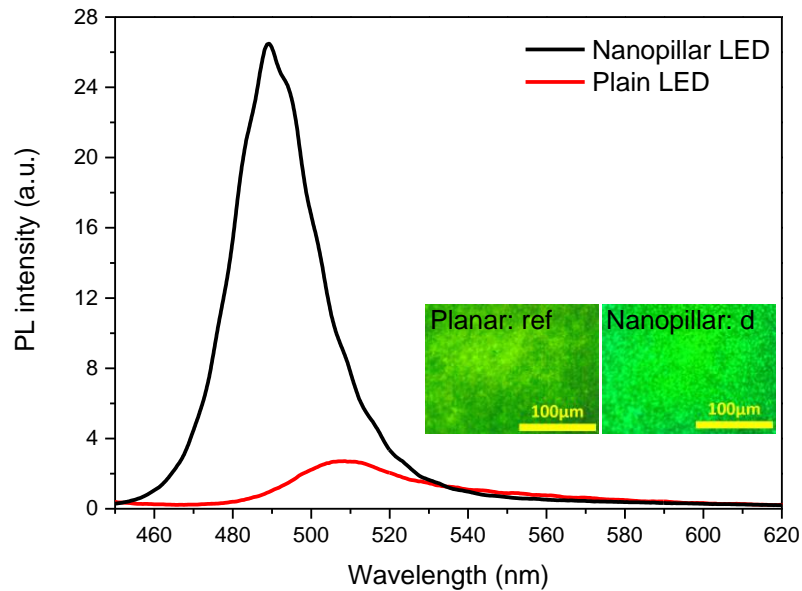
Method to reduce QCSE:

- Fabricate nanopillar structures on LED epi-wafer
- Release internal strain
- Increase IQE
- Enhance LEE



# 3. Nanopillar nitride-based LED

- Significant luminescence enhancement: more than 4 times
- Larfer optical modulation bandwidth: 221.2MHz→312.2MHz



## 4. Summary and outlook

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### Summary:

- Fabrication of hybrid structure on f-SiC → luminescence enhancement of f-SiC
- Nanopillar fabrication on nitride-based LED → luminescence enhancement /increased modulation bandwidth of LED

### Future work:

- Optimize the luminescence performance of nanostructured f-SiC
- Apply nanopillar structure on NUV LED
- To realize a nanopillar LED device with enhanced electroluminescence

# Acknowledgement

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- Innovation Fund Denmark: 'LEDSiC' project (no. 4106-00018B)

# Questions

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*Thank you for your attention!*

